

1	<b>GENERAL</b> The 200 series of austenitic stainless steels was developed during and after World War II to provide inexpensive, low-nickel substitutes for the 300 series stainless steels. Type 201 is a substitute for Type 301; the nickel in Type 301 is partially replaced with alternate austenite-stabilizing elements manganese and nitrogen. Mechanical properties and corrosion behaviors are very similar for the two grades. Type 201 is metastable and forms martensite on cold deformation, increasing the work-hardening rate. Type 201 is used in transit and railway cars, large truck trailers, moderately drawn utensils such as cookware lids, roll-formed and stretch bent automotive trim and wheel covers, household appliances, flat conveyor chain, architectural panels, and welded fire extinguisher tanks.	2.015 2.016  2.02 2.021 2.022 2.0221  2.023 2.0231  2.0232  2.024 2.025  2.03 2.031 2.0311  2.0312  2.032 2.0321    2.033 2.0331  2.034 2.0341    2.0342  2.0343   2.04	Specific heat, 32 to 212 F, 0.12 Btu/lb/F (3). Thermal diffusivity.  <b>Other Physical Properties</b> Density is 0.287 lb/in. <sup>3</sup> , 7.94 g/cc (8). Electrical properties. Electrical resistivity, 69 microhm-cm at room temperature (7). Magnetic properties. Type 201 is nonmagnetic in the annealed condition and magnetic when cold worked (2). Effect of cold reduction on magnetic permeability, Table 2.0232. Emissivity. Damping capacity.  <b>Chemical Environments</b> General corrosion. Type 201 has good corrosion resistance to a wide variety of media and can be used in most applications where Type 301 is used. The corrosion rate of Type 201 in 10 percent formic acid at room temperature is 0.4 mpy (11). Intergranular corrosion. Type 201, like Type 301 and most of the unstabilized austenitic stainless steels, is subject to intergranular corrosion in the sensitized condition. Sensitization occurs when the chromium content of the matrix is reduced by precipitation of chromium-rich carbides in the grain boundary regions. Sensitization is caused by heating in the 800 to 1500 F temperature range. In particular, slow cooling through the critical temperature range after heat treating at a higher temperature or after welding can cause sensitization. Corrosion resistance can be restored by full annealing to redissolve the precipitated carbides. Stress corrosion cracking. Type 201 is subject to stress corrosion cracking in chloride solutions. Cracking can be either transgranular or intergranular depending on the composition and temperature of the solution. Oxidation. The oxidation resistance of Type 201 is good up to about 1550 F, comparable to that of Type 301. Above this temperature, however, Type 201 scales more rapidly due to its higher manganese content. Alloy is resistant to scaling up to 1550 F for continuous service and up to 1400 F for intermittent service (9). Effect of cold work on long time corrosion in steam at 1000 F, Figure 2.0343.
1.01	<b>Commercial Designation</b> Type 201.		
1.02	<b>Alternate Designations</b> AISI 201, UNS S20100, and SAE 30201.		
1.03	<b>Specifications</b> AMS 5762, ASME SA412, ASTM A412, ASTM A581, ASTM A666, QQ-5-766, and SAE J405.		
1.04	<b>Composition</b> Composition, Table 1.04.		
1.05	<b>Heat Treatment</b>		
1.051	Annealing temperature range is 1850 to 2050 F. The alloy must be rapidly cooled or water quenched, depending on section size, in order to prevent carbide precipitation. For maximum corrosion resistance, parts should be full annealed at 1900 F or higher.		
1.052	Stress relief, 800 F, 2 hours (5).		
1.06	<b>Hardness</b>		
1.061	The alloy is hardenable only by cold working (6).		
1.062	Effect of temper on hardness, Figure 1.062.		
1.07	<b>Forms and Conditions Available</b>		
1.071	Sheet, strip, plate, bar, rod (7).		
1.08	<b>Melting and Casting Practice</b>		
1.09	<b>Special Considerations</b>		
1.091	Type 201 is subject to intergranular corrosion when sensitized due to precipitation of complex chromium-rich carbides at the grain boundaries. See paragraph 2.0321.		
2	<b>PHYSICAL PROPERTIES AND ENVIRONMENTAL EFFECTS</b>	3	
2.01	<b>Thermal Properties</b>	3.02	
2.011	Melting range is 2550 to 2650 F (9).	3.021	
2.012	Phase changes.		
2.0121	Time-temperature-transformation diagrams.	3.0211	
2.013	Thermal conductivity, 212 F, 9.4 Btu/ft/hr/ft <sup>2</sup> /F.		
2.014	Thermal expansion, Figure 2.014.		
		<b>MECHANICAL PROPERTIES</b>	
		3.01	
		<b>Specified Mechanical Properties</b>	
		3.02	
		<b>Mechanical Properties at Room Temperature</b>	
		3.021	
		Tension – stress-strain diagrams – tension properties. Type 201 is hardened by the strain-induced transformation of austenite to alpha-prime martensite. As shown in Figure 3.0212, the maximum in rate of work hardening occurs simultaneously with the	

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6.5 Mn
4.5 Ni

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3.0212	maximum rate of formation of alpha-prime martensite. The high degree of metastability of Type 201 with respect to cold deformation is indicated by the very large volume fraction, 85 percent, of martensite forming during tensile straining.	3.052	Axial fatigue behavior of smooth alloy at room temperature, Figure 3.052.
3.0213	Stress-strain curve and martensite content as a function of strain, Figure 3.0212.	3.053	Effect of cyclic straining on martensite content of annealed alloy, Figure 3.053.
3.0214	Effect of temper on mechanical properties at room temperature, Figure 3.0213.	3.054	True stress-true strain curves for monotonic and cyclic deformation, Figure 3.054.
3.0215	Effects of cold working on tensile properties, Figure 3.0214.	3.06	<b>Elastic Properties</b>
3.022	Effects of cold rolling and stress relief on tensile yield strength, Figure 3.0215.	3.061	Poisson's ratio.
3.0221	Compression - stress-strain diagrams - compression properties.	3.062	Modulus of elasticity.
3.0222	Effect of cold work on compressive yield strength, Figure 3.0221.	3.0621	Modulus of elasticity for stress-relieved and cold rolled alloy, Figure 3.0621.
3.0223	Effects of cold rolling and stress relief on compressive yield strength, Figure 3.0222.	3.0622	Modulus of elasticity in compression for stress relieved and cold rolled alloy, Figure 3.0622.
3.023	Impact.	3.063	Modulus of rigidity, 12,500 ksi (8).
3.024	Bending.	3.064	Tangent modulus.
3.025	Torsion and shear.	3.065	Secant modulus.
3.026	Bearing.	4	<b>FABRICATION</b>
3.027	Stress concentration.	4.01	<b>Forming</b>
3.0271	Notch properties.	4.011	Type 201 has good formability and is widely used in press-forming operations. Type 201 was developed as a less costly substitute for Type 301 but does not equal it in press forming characteristics in many applications. The true stress-true strain curves, shown in Figure 4.012, indicate high work-hardening rates for both 201 and 301, which effectively retard localized reduction, or necking, and permit significant stretching during forming operations. By holding nickel on the high side and nitrogen on the low side of the specification, the formability of Type 201 is improved to that of Type 301 (15).
3.0272	Fracture toughness.	4.012	Work hardening characteristics of Types 201 and 301, Figure 4.012.
3.028	Combined properties.	4.013	General. 180-degree bending of 2 T radius of Type 201 (1/2 hard) longitudinal and transverse specimens can be successfully done without cracking (6).
3.03	<b>Mechanical Properties at Various Temperatures</b>	4.014	Forging. Starting temperature 2100 to 2250 F, finishing temperature 1700 F or above (7).
3.031	Tension - stress-strain diagrams - tension properties.	4.02	<b>Machining and Grinding</b>
3.0311	Tensile properties of annealed alloy at elevated temperatures, Figure 3.0311.	4.021	Type 201 has fair to good machinability. The machinability rating is 49 percent, based on 100 percent machinability for an AISI B1112 carbon steel. Maximum cutting speed with high-speed steel tools is 80 surface feet per minute. Cutting speed can be increased with carbide tools.
3.0312	Tensile properties of full hard alloy at elevated temperatures, Figure 3.0312.	4.022	Type 201 work hardens during machining so work piece and tool must be rigid for best results. Tools must be kept sharp.
3.032	Compression - stress-strain diagrams - compression properties.	4.023	Apply good cutting fluids, such as sulfurized-chlorinated petroleum oil. This is particularly important in making heavy cuts at relatively slow feeds. Thinning the fluid with paraffin oil helps in making finishing cuts at higher speeds. The blend improves the cooling effect on the work piece and tool.
3.033	Impact.	4.024	Take positive cuts and avoid dwelling to minimize work hardening and glazing (8, 16).
3.0331	Effect of low and elevated temperatures on Charpy V impact strength of alloy, Figure 3.0331.	4.03	<b>Joining</b>
3.034	Bending.	4.031	Type 201 can be welded by all conventional techniques employed for the standard 18-8 stainless steels. Shielded-arc methods, arc welding, and spot
3.035	Torsion and shear.		
3.036	Bearing.		
3.037	Stress concentration.		
3.0371	Notch properties.		
3.0372	Fracture toughness.		
3.038	Combined properties.		
3.04	<b>Creep and Creep-Rupture Properties</b>		
3.041	Typical stress-rupture properties, Figure 3.041.		
3.05	<b>Fatigue Properties</b>		
3.051	The fatigue behavior of Type 201 in terms of total strain amplitude as a function of fatigue life is shown in Figure 3.052. The stress-strain hysteresis loops for this alloy progressively change with deformation and do not allow separation of total strain into plastic and elastic components. The change in behavior during fatigue life results from the continued formation of martensite. The rate of formation of martensite during cyclic straining, shown in Figure 3.053, is similar to that during monotonic straining, shown earlier in Figure 3.0212. The cyclic stress-strain curve exhibits hardening at much lower strains than the monotonic curve, as shown in Figure 3.054.		

welding produce good welds with excellent mechanical properties. Weld ductility is good. Type 201 is somewhat less sensitive to intergranular corrosion in the weld heat-affected zone than 300-series compositions with comparable carbon content (8).

4.04 **Surface Treating**

4.041 Scale from heat treating operations can be removed by pickling in a bath of 20 percent hydrochloric acid at 120 to 140 F or in a solution of 10 percent sulfuric acid plus 6 to 12 percent rock salt. After pickling, rinse thoroughly, dip in a warm bath of 15 to 30 percent nitric acid, and wash in water (8).

**REFERENCES**

1 Nekervis, R. J., Lund, C. H., and Hall, A. M., "Status of High-Strength Steels for the Aircraft Industry", TML, Battelle Memorial Institute, Report No. 91 (January 3, 1958).  
 2 "Stainless and Heat Resisting Steels", Steel Products Manual, AISI (June 1957).  
 3 Mangone, F. J., Roach, D. B., and Hall, A. M., "Properties of Certain Cold-Rolled Austenitic Stainless Sheet Steels", Battelle Memorial Institute, DMIC Report No. 113 (May 15, 1959).  
 4 "Trends in Metals, Stainless Steels", Electro Metallurgical Company, Division of Union Carbide, Report from STEEL (November 4, 1957).  
 5 Sumsion, H. T., "High-Strength Steels for the Missile Industry", ASM, Proceedings of the Golden Gate Metals Conference (February 4-6, 1960).  
 6 "Mechanical Properties of Some Engineering Materials - Unpublished Data From Company Sponsored

7 Programs", Third Quarter Report, Vol. II, Phase I (September 1 to November 30, 1961).  
 8 Materials in Design Engineering, "Materials Selector Issue", Reinhold Publishing Corp., Vol. 56, No. 5, Mag. (mid-October 1962).  
 9 "AISI Type 201", Alloy Digest, Filing Code: SS-174 (November 1965).  
 10 "Properties of 200 - 300 - 400 Series", Trade Literature, Republic Steel Corporation, Cleveland, Ohio (May 1964).  
 11 "Metals Handbook, Vol. 1, 8th Edition", edited by T. Lyman, American Society for Metals, Metals Park, Ohio (1961).  
 12 Lackey, J. Q. and Degnan, T. F., "Formic Acid Corrosion of Common Materials of Construction", *Materials Performance*, Vol. 13, No. 7 (July 1974) pp 13-18.  
 13 McCoy, H. E. and McNabb, B., "Corrosion of Several Iron- and Nickel-Base Alloys in Supercritical Steam at 1000 F", ORNL-TM-4552 (August 1974).  
 14 Franke, G. and Altstetter, C., "Low-Cycle Behavior of Mn/N Stainless Steels", *Metallurgical Transactions*, Vol. 7A (November 1976) pp 1719-1727.  
 15 "Mechanical and Physical Properties of Austenitic Chromium-Nickel Stainless Steels at Ambient Temperatures", The International Nickel Company, Inc., New York, N.Y., Trade Literature (October 1966).  
 16 Cunningham, E. R., "Cold Forming Stainless Steels and Other Specialty Grades", *Metals Engineering Quarterly*, Vol. 13, No. 2 (May 1973) pp 7-12.  
 17 Blott, D. M., "How to Cut the Standard Grades", *Metal Progress*, Vol. 93, No. 2 (February 1968) pp 77-83.

Fe
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6.5 Mn
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Type 201

Source	AISI (2)		(7)	
	Percent		Percent	
	Min	Max	Min	Max
Carbon	-	0.15	-	0.15
Chromium	16.0	18.0	16.0	19.0
Manganese	5.5	7.5	5.5	10.0
Nickel	3.5	5.5	3.5	6.0
Nitrogen	-	0.25	-	-
Silicon	-	1.0	-	-
Sulfur	-	0.03	-	-
Iron	Balance		Balance	

TABLE 1.04. COMPOSITION (2, 7)

	Fe
17	Cr
6.5	Mn
4.5	Ni

Type 201

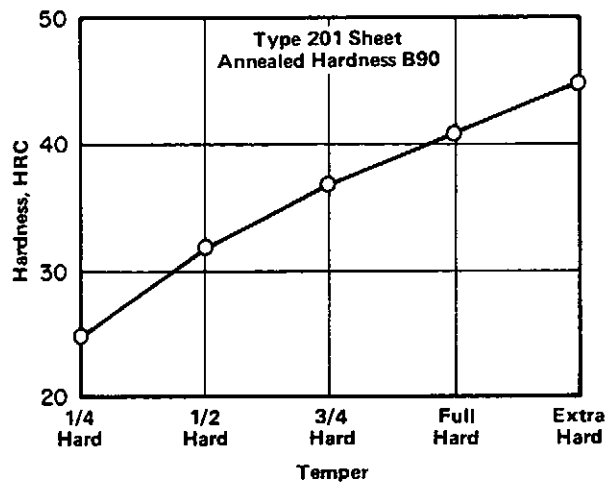


FIGURE 1.062. EFFECT OF TEMPER ON HARDNESS (8)

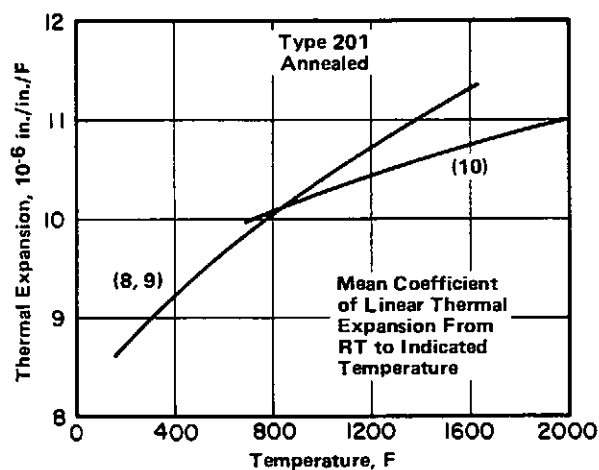


FIGURE 2.014. THERMAL EXPANSION (8, 9, 10)

Alloy	Type 201
Cold Reduction, percent	Normal Magnetic Permeability at 200 Oersteds
Anneal	1.003
5	1.016
10	1.147
20	2.738
30	7.006
50	15.395

TABLE 2.0232. EFFECT OF COLD REDUCTION ON MAGNETIC PERMEABILITY (3)

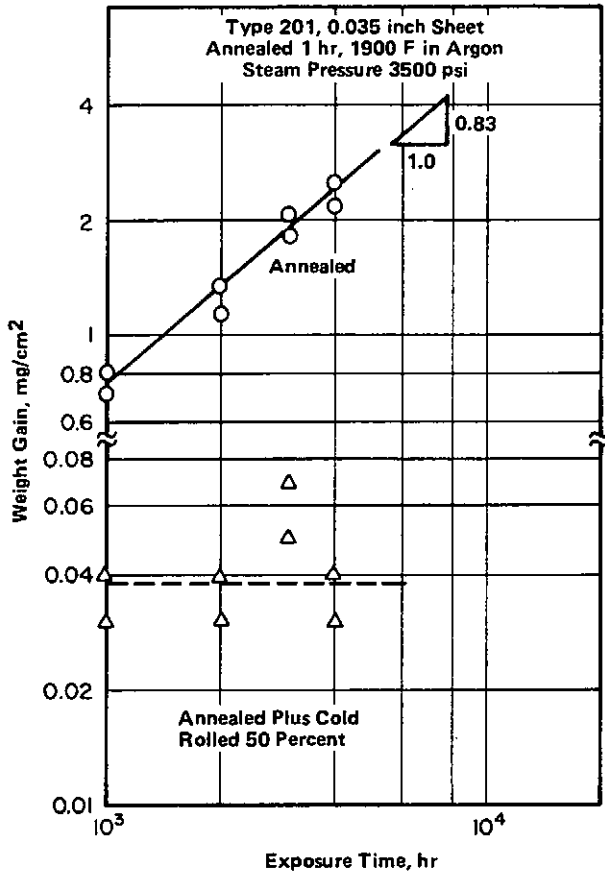
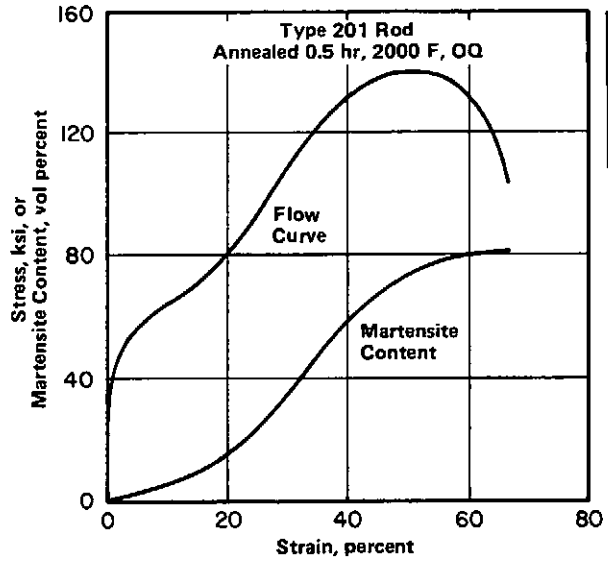


FIGURE 2.0343. EFFECT OF COLD WORK ON LONG-TIME CORROSION IN STEAM AT 1000 F (12)



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FIGURE 3.0212. STRESS-STRAIN CURVE AND MARTENSITE CONTENT AS A FUNCTION OF STRAIN (13)

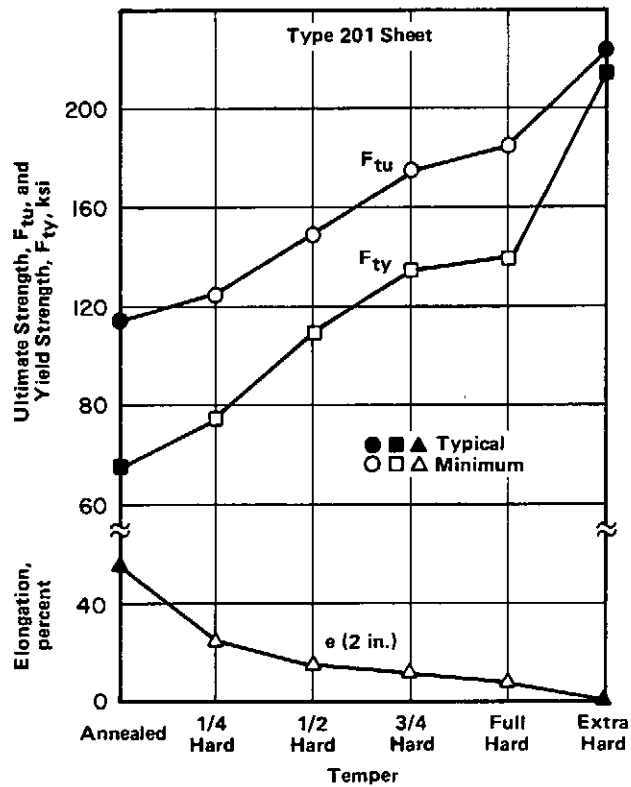


FIGURE 3.0213. EFFECT OF TEMPER ON MECHANICAL PROPERTIES AT ROOM TEMPERATURE (8)

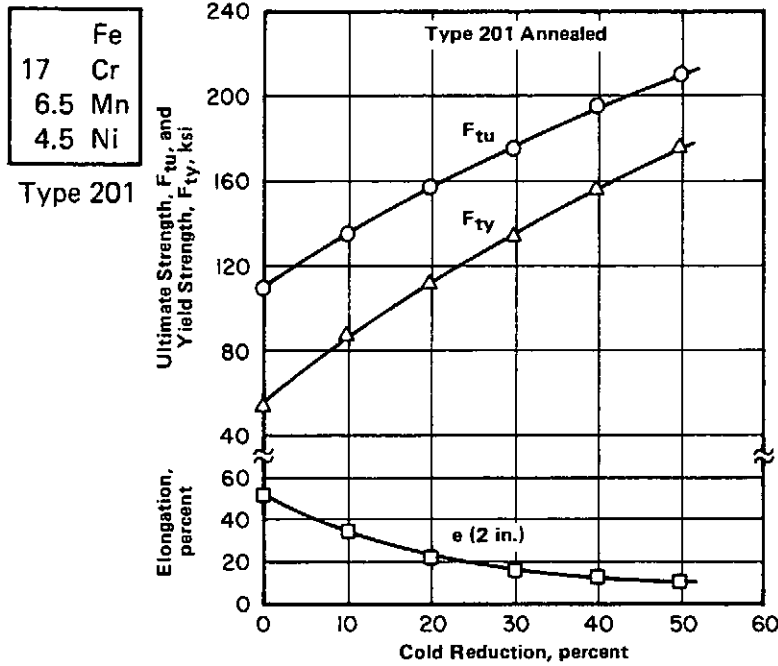


FIGURE 3.0214. EFFECTS OF COLD WORKING ON TENSILE PROPERTIES (8)

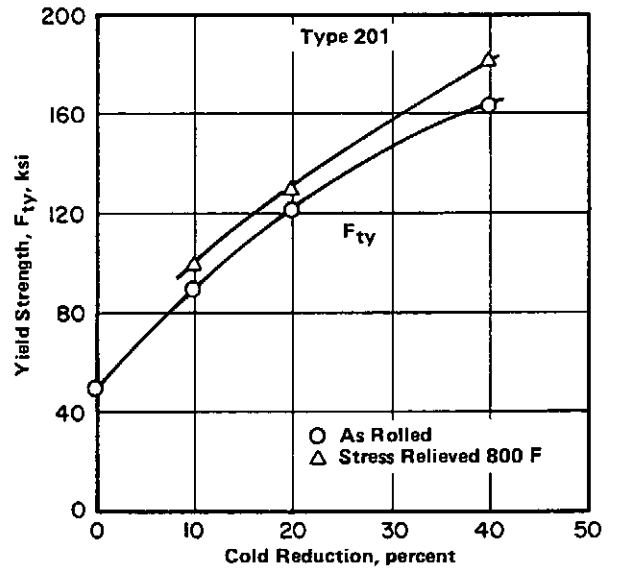


FIGURE 3.0215. EFFECTS OF COLD ROLLING AND STRESS RELIEF ON TENSILE YIELD STRENGTH (8)

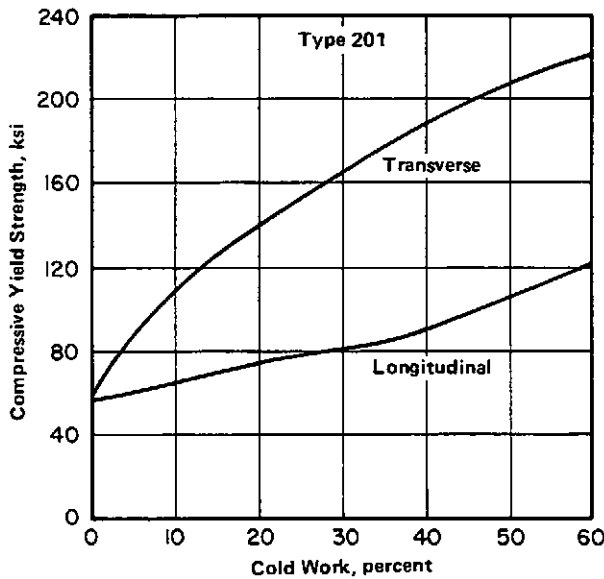


FIGURE 3.0221. EFFECT OF COLD WORK ON COMPRESSIVE YIELD STRENGTH (14)

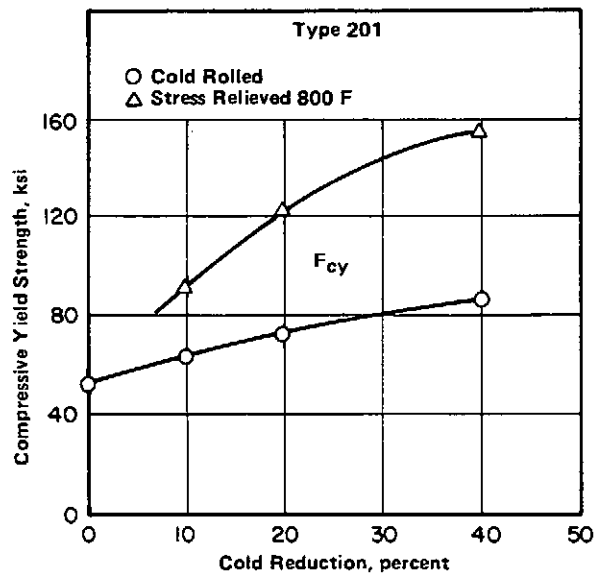
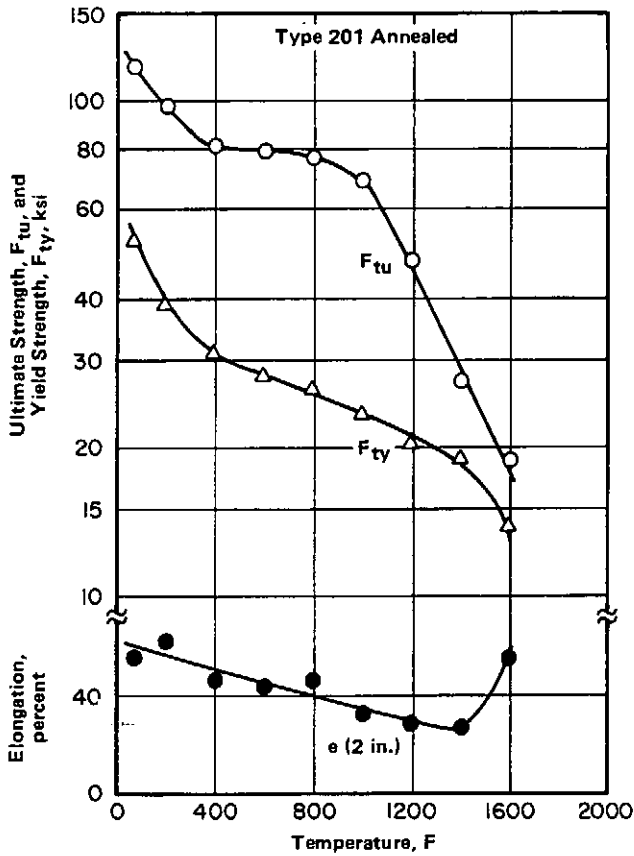


FIGURE 3.0222. EFFECTS OF COLD ROLLING AND STRESS RELIEF ON COMPRESSIVE YIELD STRENGTH (8)



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FIGURE 3.0311. TENSILE PROPERTIES OF ANNEALED ALLOY AT ELEVATED TEMPERATURES (8)

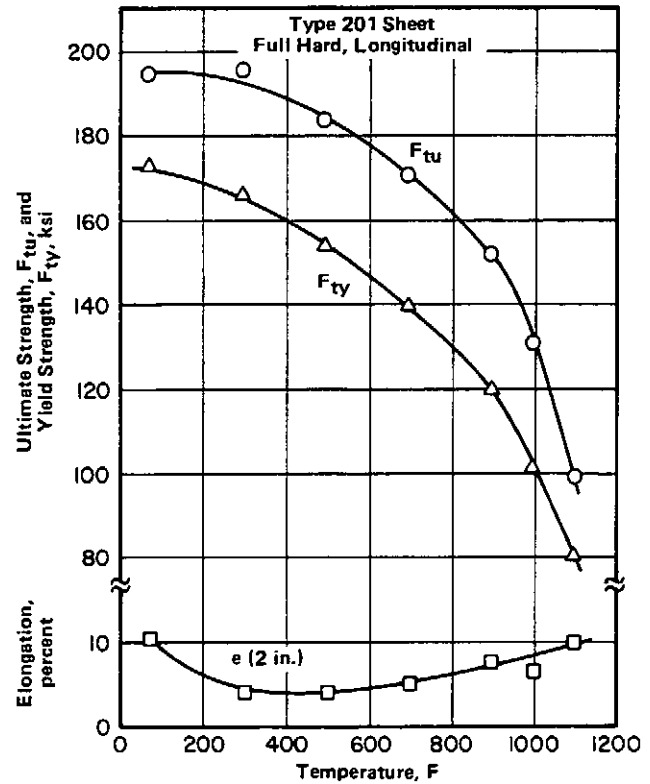


FIGURE 3.0312. TENSILE PROPERTIES OF FULL HARD ALLOY AT ELEVATED TEMPERATURES (8)

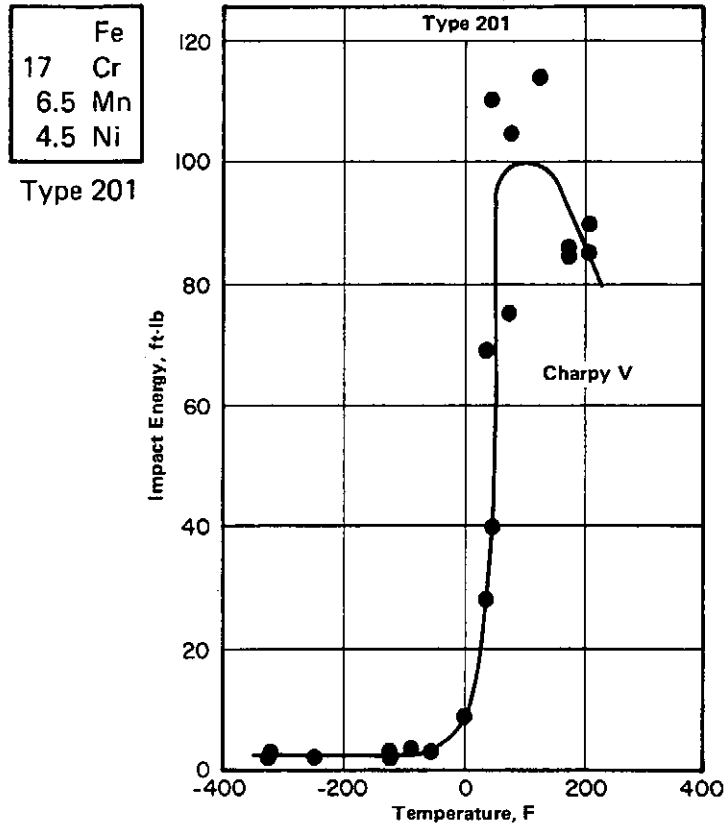


FIGURE 3.0331. EFFECT OF LOW AND ELEVATED TEMPERATURES ON CHARPY V IMPACT STRENGTH OF ALLOY (2)

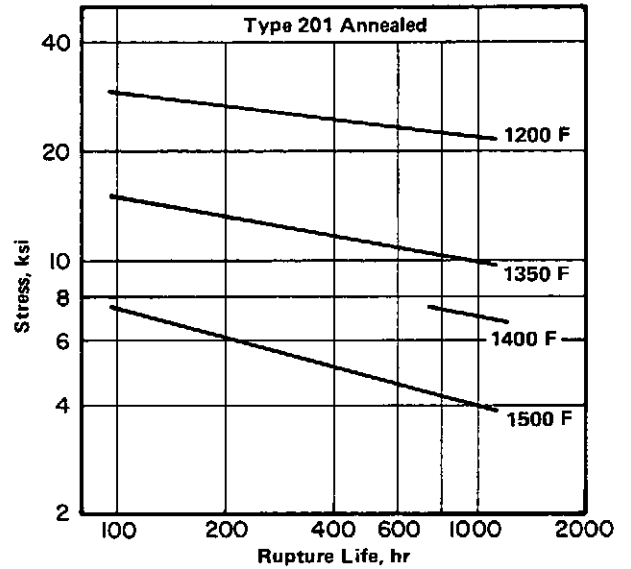


FIGURE 3.041. TYPICAL STRESS-RUPTURE PROPERTIES (8)

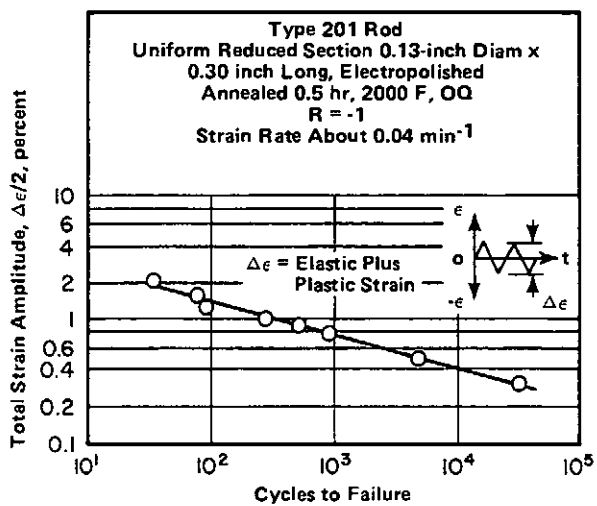


FIGURE 3.052. AXIAL FATIGUE BEHAVIOR OF SMOOTH ALLOY AT ROOM TEMPERATURE (13)

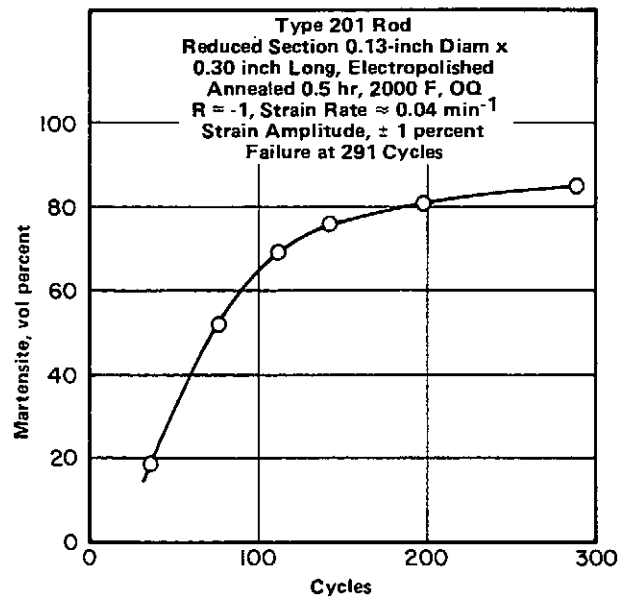


FIGURE 3.053. EFFECT OF CYCLIC STRAINING ON MARTENSITE CONTENT OF ANNEALED ALLOY (13)

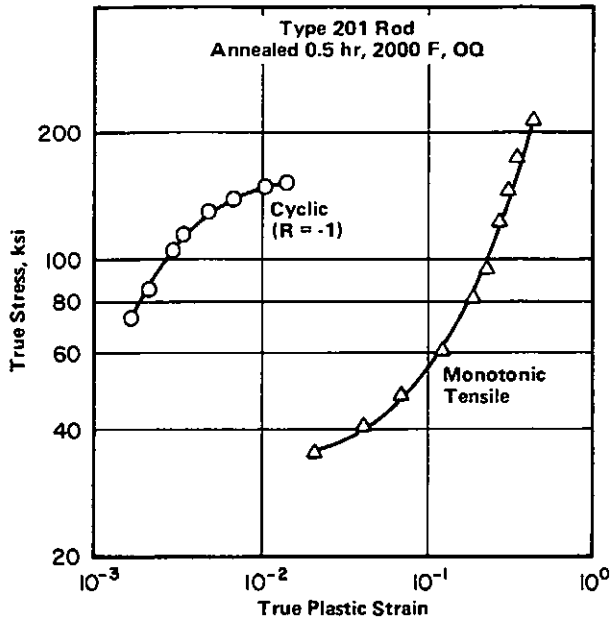
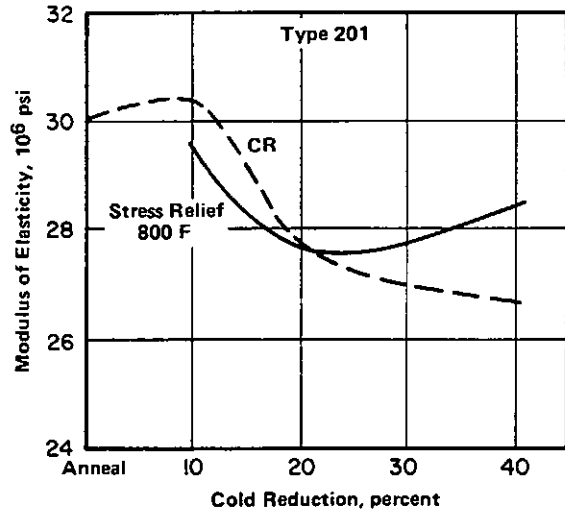


FIGURE 3.054. TRUE STRESS-TRUE STRAIN CURVES FOR MONOTONIC AND CYCLIC DEFORMATION (13)



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FIGURE 3.0621. MODULUS OF ELASTICITY FOR STRESS RELIEVED AND COLD ROLLED ALLOY (3)

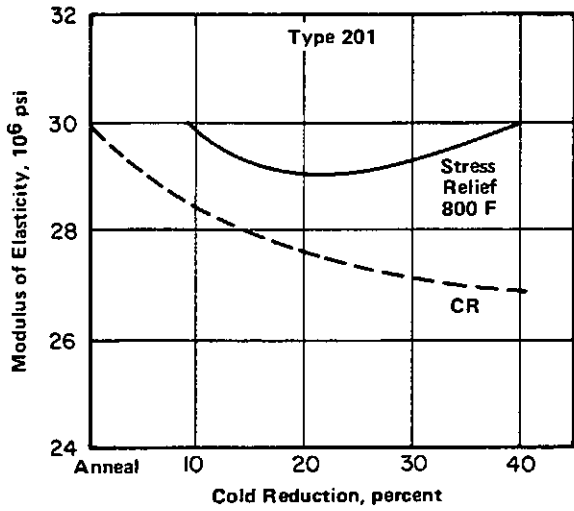


FIGURE 3.0622. MODULUS OF ELASTICITY IN COMPRESSION FOR STRESS RELIEVED AND COLD ROLLED ALLOY (3)

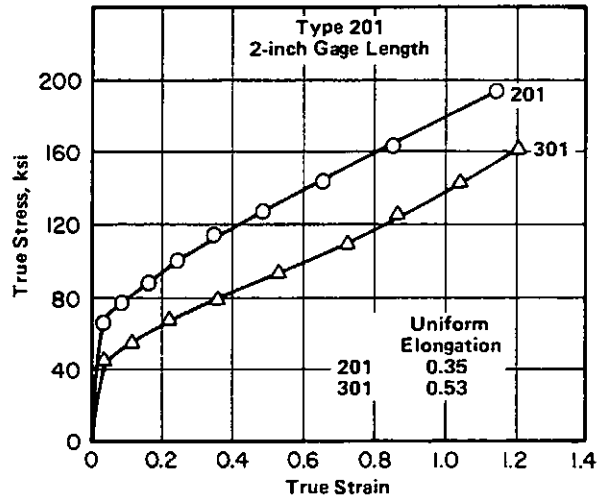


FIGURE 4.012. WORK HARDENING CHARACTERISTICS OF TYPES 201 AND 301 (15)

