

1 GENERAL
 This heat-treatable low-alloy steel has relatively low hardenability; nevertheless, it is one of the most popular alloy steels because of its good formability and weldability along with an excellent combination of mechanical properties. Its optimum combination of properties is developed in limited section thicknesses by quench-and-temper heat treatment, but sufficient strength and toughness for many applications can be obtained by normalizing. It is recommended for use at temperatures up to a maximum of 700 F because its strength decreases markedly with increasing temperatures above that level. At sub-zero temperatures it undergoes a transition from ductile to brittle behavior in a Charpy V impact test, when it has poor impact properties. The transition temperature varies with the heat treatment.

This steel is not subject to temper embrittlement, and can be nitrided. It is usually forged at 2000 F to 2200 F and the finishing temperature should never fall below 1800 F. For applications in moist, marine, and other corrosive environments, paint, electroplate, or other protective coatings should be applied to inhibit rusting and other corrosion. It is used in both cast and wrought form for many applications requiring high strength and toughness.

4130 steel is available as billet, bar, rod, forgings, sheet, plate, tubing, and castings. It is used to make automotive connecting rods, engine mounting lugs, shafts, fittings, bushings, gears, bolts, axles, gas cylinders, airframe components, hydraulic lines, and nitrided machinery parts. (30,74)

- 1.01 **Commercial Designation**
4130.
- 1.02 **Alternate Designations**
AISI 4130, SAE 4130, 4130H, UNS G41300, UNS H41300.
- 1.03 **Specifications**
Table 1.03.
- 1.04 **Composition**
Table 1.04.
- 1.05 **Heat Treatment**
- 1.051 **Normalize:** Heat to 1600 to 1700 F, air cool (tempering at 900 F or above is often carried out after normalizing to raise the relatively low as-normalized yield strength) (30).
- 1.052 **Anneal:** Heat to 1525 to 1575 F, furnace cool (30).
- 1.053 **Harden:** Heat to 1500 to 1600 F, water quench, or heat to 1575 to 1625 F, oil quench. Inert or endothermic atmosphere is desirable to prevent decarburization (30).
- 1.054 **Temper:** Heat to 400 to 1300 F, hold 1/2 hour minimum, air cool or water quench (the alloy is not susceptible to temper embrittlement) (30).
- 1.055 **Spheroidize:** Heat to 1400 to 1425 F, hold at temperature for 6 to 12 hours, cool slowly (30).

- 1.06 **Hardness**
- 1.061 Effect of tempering temperature on hardness of bar, Figure 1.061.
- 1.062 Effect of tempering temperature on hardness of oil-quenched and water-quenched bar, Figure 1.062.
- 1.063 Effect of exposures at elevated temperatures on room-temperature hardness of sheet after three different tempering treatments, Figure 1.063.
- 1.064 Hardness of bar, plate, sheet, and extrusion in various heat-treated conditions, Table 1.064.
- 1.065 Effect of diameter on surface hardness of quenched-and-tempered bars, Figure 1.065.
- 1.066 End-quench hardenability, Figure 1.066.
- 1.067 Hardness profile through nitrided case, Figure 1.067.
- 1.068 Effect of repress (forging) pressure and temperature on hardness of powder metallurgy forging, Figure 1.068.
- 1.069 Effect of fabrication process on hardness properties of powder metallurgy forging, Table 1.069.

- 1.07 **Forms and Conditions Available**
- 1.071 Wrought products are available in the full range of sizes and forms including billets, bars, rods, forgings, sheets, plates, strip, and tubing. The products are furnished in the annealed, normalized and tempered, or hardened and tempered conditions (30). Sand, centrifugal, and investment castings are available in any desired heat-treated condition (30).
- 1.072

- 1.08 **Melting and Casting Practice**
- 1.081 The alloy is generally air melted in basic-electric, basic-open-hearth, or basic-oxygen furnaces. For some applications requiring exceptional quality, it is induction or consumable-electrode remelted in vacuum.

- 1.09 **Special Considerations**
- 1.091 Because of the limited hardenability of Type 4130, optimum and uniform mechanical properties can be obtained only in section thicknesses that are small enough to through-harden when quenched from the austenitizing temperature. For critical applications requiring high strength and toughness, therefore, use should be made of precise hardenability calculations to determine the suitability of specific heat compositions.
- 1.092 Type 4130 is susceptible to hydrogen embrittlement in chrome-, nickel-, and cadmium-plating solutions and in hydrochloric-acid pickling. It is generally not susceptible in alkaline cleaners or anodic acid cleaners. Baking at 375 F for times up to 24 hours, depending upon the severity of the embrittlement, is effective in relieving the embrittlement (19,20).
- 1.093 Effect of hydrogen gas pressure on embrittlement (see Figure 3.02726).

2 PHYSICAL AND CHEMICAL PROPERTIES

- 2.01 **Thermal Properties**
- 2.011 Melting range – about 2795 F.
- 2.012 Phase changes.
- 2.0121 Time-temperature-transformation diagram, Figure 2.0121.

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

2.0122	Upon heating under equilibrium conditions, the transformation from pearlite or martensite to austenite starts at 1395 F (AC ₁) and is complete at 1490 F (AC ₃) (35).	3.01	Specified Mechanical Properties Table 3.01.
2.0123	Upon slow cooling under equilibrium conditions, the transformation from austenite to pearlite starts at 1390 F (Ar ₃) and is complete at 1380 F (Ar ₁) (35).	3.02 3.021	Mechanical Properties at Room Temperature Tension – stress-strain diagrams – tension properties.
2.0124	Upon rapid quenching from the austenitic condition the transformation to martensite starts at 700 F (M _s) and is complete at 550 F (M _f) (30).	3.0211 3.0212	Stress-strain curves (see Figure 3.0311). Tensile properties of round bar in various heat-treated conditions, Table 3.0212.
2.013	Thermal conductivity, Figure 2.013.	3.0213	Tensile properties of plate in various heat-treated conditions, Table 3.0213.
2.014	Thermal expansion, Figure 2.014.	3.0214	Tensile properties of 3/8 inch plate heat treated to various hardnesses, Table 3.0214.
2.015	Specific heat, Figure 2.015.	3.0215	Tensile properties of extruded round bar in various heat-treated conditions, Table 3.0215.
2.016	Thermal diffusivity.	3.0216	Tensile properties of extruded flat bar in various heat-treated conditions, Table 3.0216.
2.02	Other Physical Properties	3.0217	Tensile properties of castings in various heat-treated conditions, Table 3.0217.
2.021	Density, 0.2834 lb/in. ³ , 7.845 gr/cm ³ (30).	3.0218	Effect of tempering temperature on room-temperature tensile properties of 0.530 inch bar, Figure 3.0218.
2.0211	Effect of initial compacting pressure on density of powder metallurgy forging, Figure 2.0211.	3.0219	Effect of tempering temperature on tensile properties of oil-quenched and water-quenched 1 inch bar, Figure 3.0219.
2.0212	Effect of repress (forging) pressure and temperature on density of powder metallurgy forging, Figure 2.0212.	3.02110	Effect of sheet thickness and tempering temperature on tensile properties, Table 3.02110.
2.022	Electrical properties.	3.02111	Effect of austenitizing temperature and quench medium on tensile properties of as-quenched material, Table 3.02111.
2.0221	Electrical resistivity as a function of temperature, Figure 2.0221.	3.02112	Effect of quench medium and tempering temperature on tensile properties of rolled bar, Table 3.02112.
2.023	Magnetic properties, ferromagnetic.	3.02113	Effect of quench medium on tensile properties of sheet, Table 3.02113.
2.024	Emittance.	3.02114	Effect of bar diameter on tensile properties in various heat-treated conditions, Figure 3.02114.
2.025	Damping capacity.	3.02115	Tensile properties of quenched and tempered round bar, Figure 3.02115.
2.0251	Effect of number of cycles in reversed bending and maximum cyclic tensile strain (Em) on damping capacity, Figure 2.0251.	3.02116	Effect of explosive forming on tensile properties of plate, Table 3.02116.
2.03	Chemical Properties	3.02117	Effect of fabrication process on tensile properties of powder metallurgy forging, Table 3.02117.
2.031	Type 4130 is susceptible to general corrosion – rusting and pitting – and to stress corrosion, particularly in industrial and marine environments. For most applications it should be protected by suitable electroplating, paint, or a combination of both (21). Corrosion rate in sea-water decreases at least by 50 percent with exposure from 1 year to 3-1/2 years, and the rate at a depth of 5500 feet is less than half that in surface waters (67).	3.02118	Effect of repress (forging) pressure and temperature on ultimate tensile strength of powder metallurgy forging, Figure 3.02118.
2.032	Stress corrosion of cadmium-plated bar and sheet subjected to alternate immersions (10 minutes in and 50 minutes out) in 3-1/2 percent solution of sodium chloride having pH of 6.8 to 7.2, Table 2.032.	3.02119	Effect of Mn plating and elevated temperature exposure on tensile properties of sheet, Table 3.02119.
2.033	Type 4130 is resistant to corrosion by pure fluorinated hydrocarbons, such as Propellants 113 and 114B2 (Freon-113 and Freon-114B2), which are used as solvents and refrigerants as well as propellants. In the presence of small amounts of moisture, however, these types of hydrocarbons significantly attack the alloy (37).	3.02120	Temperature limits for prolonged unstressed exposure which prevent degradation of tensile strength, Table 3.02120.
2.034	Effect of temperature and chloride concentration on "break point" of sheet in two-point bend test, Table 2.034.	3.02121	Exposure to sulfur hexafluoride (SF ₆) at a temperature of 440 F and a pressure of 525 psi for up to 24 hours does not degrade the tensile ultimate (F _{TU}) and yield (F _{TY}) strengths of 0.094-inch-thick sheet (64).
2.035	Effect of sulfur hexafluoride (SF ₆) on tensile properties (see Section 3.02121).	3.022	Compression – stress-strain diagrams – compression properties.
2.036	Effect of yield strength and flaw depth on threshold stress intensity (K _{I,SCC}) (see Figure 3.02725).	3.0221 3.0222	Stress-strain curves (see Figure 3.0321). Compressive yield strength (see Figure 3.0322).
3	MECHANICAL PROPERTIES	3.023 3.0231 3.0232	Impact. Impact properties of extruded bar, Table 3.0231. Effect of tempering temperature on room-temperature impact strength, Figure 3.0232.

- 3.0233 Impact resistance of normalized and annealed material, Table 3.0233.
- 3.0234 Effect of quench medium and tempering temperature on impact properties of rolled bar, Figure 3.0234.
- 3.0235 Effect of fabrication process and temperature on impact properties of powder metallurgy forging, Table 3.0235.
- 3.0236 Effect of repress (forging) pressure and temperature on impact properties of powder metallurgy forging, Figure 3.0236.
- 3.0237 Effect of carburization on impact properties, Figure 3.0237.
- 3.024 Bending.
- 3.025 Torsion and shear.
- 3.0251 Shear strength (see Figures 3.0351 and 3.0352).
- 3.026 Bearing.
- 3.0261 Bearing strength (see Figures 3.0361 and 3.0362).
- 3.027 Stress concentration.
- 3.0271 Notch properties.
- 3.02711 Effect of initial crack length ($2 a_0$) on residual strength (maximum load divided by gross cross-sectional area) of sheet after two different tempering treatments, Figure 3.02711.
- 3.02712 Effect of temperature on slow-crack-growth rate in gaseous hydrogen, Figure 3.02712.
- 3.02713 Effect of hydrogen pressure on slow-crack-growth rate, Figure 3.02713.
- 3.0272 Fracture toughness.
- 3.02721 Plane-strain fracture toughness of material heat treated to 158 ksi yield (F_{TY}) strength is 100 ksi $\sqrt{\text{in.}}$ when tested in three-point loading (73).
- 3.02722 Effect of sheet thickness on fracture-toughness parameter K_{IC} , Table 3.02722.
- 3.02723 Effect of yield strength on fracture resistance of sheet, Figure 3.02723.
- 3.02724 Effect of tempering temperature, carbon content, and quenching medium on plane-strain fracture toughness of flat rolled bar, Figure 3.02724.
- 3.02725 Effect of yield strength and flaw depth on threshold stress intensity in static loading, Figure 3.02725 (58).
- 3.02726 Effect of hydrogen pressure and displacement rate on initiation of slow crack growth, K_{Iscg} (embrittlement), Figure 3.02726.
- 3.02727 Effect of hydrogen pressure on the arrest of crack propagation, Table 3.02727.
- 3.028 Combined properties.
- 3.03 **Mechanical Properties at Various Temperatures**
- 3.031 Tension – stress-strain diagrams – tension properties.
- 3.0311 Tensile stress-strain curves for sheet at temperatures up to 1000 F after three different tempering treatments; curves apply to specimens held for 0.5 hour up to 100 hours at test temperature, Figure 3.0311.
- 3.0312 Effect of temperature on tensile properties of sheet after three different tempering treatments and two different exposure times, Figure 3.0312.
- 3.0313 Effect of temperature and strain rate on tensile properties of normalized sheet after rapid heating; data pertain to specimens held for 10 seconds up to 1/2 hour at test temperature, Figure 3.0313.
- 3.0314 Effect of temperature and strain rate on tensile properties of heat-treated sheet after rapid heating; data pertain to specimens held for 10 seconds up to 1/2 hour at test temperature, Figure 3.0314.
- 3.0315 Effect of temperature on tensile properties of sheet after 5-minute soak at temperature, Figure 3.0315.
- 3.0316 Effect of short-time exposure to elevated temperature on tensile properties of sheet, Figure 3.0316.
- 3.0317 Effect of prior exposure to test temperature up to 10 hours on tensile properties, Figure 3.0317.
- 3.032 Compression – stress-strain diagrams – compression properties.
- 3.0321 Compressive stress-strain curves for sheet at temperatures up to 1000 F after three different tempering treatments; curves apply to specimens held for 1/2 up to 100 hours at test temperatures, Figure 3.0321.
- 3.0322 Effect of temperature on compressive yield strength of sheet after three different tempering treatments and two different holding times, Figure 3.0322.
- 3.033 Impact.
- 3.0331 Effect of grain size and temperature on impact properties, Figure 3.0331.
- 3.0332 Effect of low temperatures on impact properties in several heat-treated conditions, Figure 3.0332.
- 3.0333 Effect of low temperatures on impact properties of pipe, Figure 3.0333.
- 3.0334 Effect of temperature on impact properties of explosively formed and rolled and annealed plate, Figure 3.0334.
- 3.0335 Effect of temperature on impact properties of explosively formed and cold-rolled heat-treated plate, Figure 3.0335.
- 3.034 Bending.
- 3.035 Torsion and shear.
- 3.0351 Effect of temperature on shear strength of sheet after three different tempering treatments and two different exposure times, Figure 3.0351.
- 3.0352 Effect of temperature on shear strength of normalized and of cold-worked-and-stress-relieved Type 4130, Figure 3.0352.
- 3.0353 Effect of prior test temperature exposure up to 10 hours on shear properties, Figure 3.0353.
- 3.036 Bearing.
- 3.0361 Effect of temperature on bearing strength ($e/D = 2$) of sheet after three different tempering treatments and two different exposure times, Figure 3.0361.
- 3.0362 Effect of temperature on bearing strength ($e/D = 1.5$) of sheet after three different tempering treatments, Figure 3.0362.
- 3.0363 Effect of prior test temperature exposure up to 10 hours on bearing properties, Figure 3.0363.
- 3.037 Stress concentration.
- 3.0371 Notch properties.
- 3.03711 Comparison of tensile ultimate (F_{TU}) and yield (F_{TY}) strengths with crack strength (net fracture stress) and fracture appearance of shear-cracked sheet specimens quenched and tempered at 825 F, Figure 3.03711.
- 3.03712 Comparison of tensile ultimate (F_{TU}) and yield (F_{TY}) strengths with crack strength (net fracture stress) and fracture appearance of shear-cracked sheet specimens quenched and tempered at 400 F, Figure 3.03712.

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

- 3.03713 Effect of temperature and loading rate on crack strength (net fracture stress) and fracture appearance of fatigue cracked sheet specimens quenched and tempered at 400 F, Figure 3.03713.
- 3.03714 Effect of temperature on crack strength (net fracture stress) and fracture appearance of both fatigue-cracked and shear-cracked sheet specimens quenched and tempered at 825 F, Figure 3.03714.
- 3.0372 Fracture toughness.
- 3.038 Combined properties.
- 3.04 **Creep and Creep Rupture Properties**
- 3.041 Creep-rupture curves from 700 to 1100 F, Figure 3.041.
- 3.042 Creep-rupture and minimum-creep-rate characteristics of normalized sheet at 1000 and 1200 F, Figure 3.042.
- 3.05 **Fatigue Properties**
- 3.051 Fatigue life as a function of cyclic strain for Type 4130 tempered at 1200 F, Figure 3.051.
- 3.052 Fatigue life as a function of cyclic strain for Type 4130 tempered at 750 F, Figure 3.052.
- 3.053 Effect of maximum cyclic tensile strain on fatigue life in reversed bending, Figure 3.053.
- 3.054 Fatigue-crack growth rate as a function of maximum cyclic stress intensity factor (K) for center cracked specimens, Figure 3.054.
- 3.055 Fatigue-crack-growth rate in air and vacuum as a function of stress intensity amplitude (ΔK) for compact tension specimen, Figure 3.055.
- 3.056 Effect of prestressing and removal of surface-layer stress by thermal relaxation on fatigue crack growth rate of plate, Figure 3.056.
- 3.057 Effect of prestressing and chem-milling on fatigue crack growth rate of sheet, Figure 3.057. -
- 3.058 S-N curve at room temperature in rotating bending, Figure 3.058.
- 3.059 Comparison of axial-loading and rotating-beam-bending fatigue data, Figure 3.059.
- 3.0510 Axial-stress S-N curves for edge-notched specimens, Figure 3.0510.
- 3.0511 Effect of temper treatment (measured by Rockwell hardness) on repeated-bending fatigue properties, Figure 3.0511.
- 3.0512 Effect of temper treatment (measured by Rockwell hardness) on rotating-beam fatigue properties, Figure 3.0512.
- 3.0513 Effect of chromium coating on axial-stress fatigue properties of sheet, Figure 3.0513.
- 3.06 **Elastic Properties**
- 3.061 Poisson's ratio, 0.25 to 0.29 (22,36).
- 3.062 Modulus of elasticity.
- 3.0621 Effect of temperature on modulus of elasticity in tension and compression of sheet after three different tempering treatments, Figure 3.0621.
- 3.0622 Compressive tangent modulus curves for sheet after three different tempering treatments, Figure 3.0622.
- 3.063 Modulus of rigidity.
- 4 **FABRICATION**
- 4.01 **Formability**
- 4.011 In the annealed condition, Type 4130 has good cold formability for a heat-treatable alloy steel; but its cold formability, of course, does not approach that of a low-carbon steel.
- 4.012 Its hot formability is excellent. All forging should be done within the temperature range 2250 to 1800 F; the optimum forging range is 2200 to 2000 F. Parts should be slow cooled in air or other appropriate environment after hot forming (30,35).
- 4.013 Effect of temperature on formability of sheet, Figure 4.013.
- 4.02 **Machining and Grinding**
- 4.021 Type 4130 can be readily machined in the annealed, normalized, and cold-drawn conditions. In the cold-drawn condition, which provides optimum machinability for this alloy, machinability is 60 to 70 percent of that of AISI B1112 screw stock.
- 4.03 **Welding**
- 4.031 Type 4130 has good weldability by oxyacetylene techniques and by any of the automatic or manual electric-arc processes. In oxyacetylene welding, the filler metal should be the same alloy as the base metal. In arc-welding, low-hydrogen low-alloy electrodes should be used. Although good welded properties can be obtained without thermal treatment, preheating to about 400 F and post-weld tempering are recommended when possible for improved weld properties. If complete quench and temper heat treatment can be carried out after welding, joint efficiencies approaching 100 percent can be obtained.
- 4.032 Tensile properties of parent metal and of 0.5 inch thick plate with butt weld as the center of the specimen gage length, Table 4.032.
- 4.033 Charpy V-notch impact properties at -65 F of butt-welded 0.50-inch-thick plate, Table 4.033.
- 4.034 Tensile properties of sheet butt welded by various arc processes, Table 4.034.
- 4.035 Tensile properties of butt-welded sheet, Table 4.035.
- 4.036 Effect of electrode alloy and sheet or plate thickness on tensile ultimate strength of bead-off butt welds, Table 4.036.
- 4.037 Effect of electrode alloy and sheet or plate thickness on shear ultimate strength of longitudinal-weld lap joints, Table 4.037.
- 4.038 Tensile and shear ultimate-strength design curves for sheet and plate welds heat treated to either 150 or 175 ksi, Figure 4.038.
- 4.039 Constant-life diagram for an axial-loaded tube splice - stresses calculated in the large tube, Figure 4.039.
- 4.0310 Constant-life diagram for an axial-loaded tube splice - stresses calculated in the small tube, Figure 4.0310.
- 4.04 **Surface Treatments**
- 4.041 With proper surface preparation, protective coatings of paint and electroplating can be applied with excellent results.

REFERENCES

1 AMS 5336D (July 1, 1984).
 2 AMS 6348 (October 15, 1979).
 3 AMS 6350F (October 16, 1978).
 4 AMS 6351C (July 16, 1979).
 5 AMS 6356C (April 1, 1983).
 6 AMS 6360G (January 15, 1977).
 7 AMS 6361B (October 1, 1984).
 8 AMS 6362C (January 1, 1985).
 9 AMS 6370J (October 1, 1983).
 10 AMS 6371F (October 15, 1979).
 11 AMS 6373B (July 15, 1978).
 12 AMS 6374 (July 1, 1985).
 13 AMS 6457 (October 15, 1979).
 14 Melonas, J. V., and Kattus, J. R., "Determination of Tensile, Compressive, Bearing, and Shear Properties of Ferrous and Non-Ferrous Structural Sheet Metals at Elevated Temperatures", Southern Research Institute, WADC TR56-340, ASTIA Document No. AD 131 069 (September 1957).
 15 Morrison, J. D., and Kattus, J. R., "Tensile Properties of Aircraft-Structural Metals at Various Rates of Loading After Rapid Heating", Southern Research Institute, WADC TR 55-199, Part 2, ASTIA Document No. AD 110 540 (November 1956).
 16 "Modern Steels and Their Properties", Seventh Edition, Handbook 2757, Bethlehem Steel Corp. (1972).
 17 Sullivan, A. M., and Freed, C. N., "Plane Stress Fracture Resistance of One Sheet Steel and Two Titanium Sheet Alloys", Naval Research Laboratory Report 7332 (October 27, 1971).
 18 Simmons, W. F., and Cross, H. C., "Elevated Temperature Properties of Wrought Medium-Carbon Alloy Steels", ASTM STP No. 199 (1957).
 19 Cataldo, C. E., "Compatibility of Metals With Hydrogen", NASA TM X-53807, George C. Marshall Space Flight Center (December 26, 1968).
 20 Groeneveld, T. P., Fletcher, E. E., Elsea, A. R., "A Review of the Literature on Cleaning, Pickling, and Electroplating Processes and Relief Treatments to Minimize Hydrogen Embrittlement of Ultrahigh-Strength Steels", Battelle Memorial Institute (October 15, 1966).
 21 Nelson, E. E., "Stress Corrosion Cracking of Several High Strength Ferrous and Nickel Alloys", NASA TMX-64626, George C. Marshall Space Flight Center (November 11, 1971).
 22 Smith, R. W., Hirschberg, M. H., and Manson, S. S., "Fatigue Behavior of Materials Under Strain Cycling in Low and Intermediate Life Range", NASA TN D-1574, Lewis Research Center (October 10, 1962).
 23 Horsley, J. J., "Stress Intensity Factors for Selected Alloys", the Boeing Company, No. D3-7326 (April 26, 1968).
 24 Blatherwick, A. A., and Mowbray, D. F., "Cyclic Plasticity Effects in Intermediate and Low Cycle Fatigue", AFML ML-TDR-64-120, University of Minnesota (May 1964).
 25 Kramer, I. R., and Kumar, A., "Effect of Vacuum Environment on Mechanical Behavior", AFOSR-TR-72-0734, Martin Marietta Corporation (February 1972).

26 Goldman, M., "An Evaluation of Four Heat Treated Conditions of 4130 Steel for Low Temperature Application", Goodyear Aircraft Corp., Serial No. 8753 (June 16, 1958).
 27 Haugen, E. B., "Statistical Strength Properties of Common Metal Alloys", North American Aviation, Inc., SID 65-1274 (October 30, 1965).
 28 "Cross-Index of Chemically Similar Specifications and Identification Code", MIL-HDBK-H1D (June 22, 1970).
 29 "Summary of Steel Castings Specifications", Steel Founders' Society of America (January 1, 1971).
 30 "AISI 4130", Alloy Digest, Filing Code: SA-23 (November 1954).
 31 Data supplied by C. J. Cooley, American Iron and Steel Institute (June 18, 1973).
 32 Rolfe, S. T., and Novak, S. R., "Slow-Bend K_{Ic} Testing of Medium-Strength High-Toughness Steels", ASTM STP 463 (1970).
 33 Metals Handbook, Vol 1, 8th Edition, ASM (1961).
 34 "Atlas of Isothermal Transformation Diagrams", U.S. Steel Corporation (1951).
 35 "Mechanical Properties of Alloy Steel", Republic Steel Corporation (1961).
 36 North American Aviation Data Sheet on Alloy Steel - AISI 4130, AI-2604.
 37 Jackson, J. D., and Boyd, W. R., "Compatibility of Propellants 113 and 114B2 With Aerospace Structural Materials", DMIC Memorandum 151 (April 27, 1962).
 38 Morrison, J. D., and Kattus, J. R., "An Investigation of Methods for Determining the Crack-Propagation Resistance of High-Strength Alloys", Southern Research Institute, Bureau Naval Weapons Contract NOas61-0392-d (February 1962).
 39 Morrison, J. D., Jenkins, P. C., and Kattus, J. R., "An Investigation of the Crack-Propagation Resistance of High-Strength Alloys and Heat-Resistant Alloys", Southern Research Institute, Bureau Naval Weapons Contract NOW 61-0392-d (October 23, 1962).
 40 Babcock & Wilcox Company, Data Sheet on 4140, 4130, and 410 Steels (1962).
 41 Manson, S. S., Nachtigall, A. J., and Freche, J. C., "A Proposed New Relation for Cumulative Fatigue Damage in Bending", Proc. ASTM, Vol 61 (1961).
 42 Lange, E. A., and Cooley, L. A., "Fracture Control Plans for Critical Structural Materials Used in Deep-Hole Experiments", Naval Research Laboratory Memorandum Report 2497 (September 1972).
 43 Ashauer, R. C., Goodman, S., Rienks, F., and McMaster, J. A., "A Study of Pulsed Arc Welding Applied to Aerospace Alloys", AFML TR-69-332 (December 1960).
 44 Sinnott, M. T., "Fatigue Properties of Chromium-Plated Heat-Treated SAE 4130 Steel", Engineering Research Institute, University of Michigan, Ann Arbor (September 1, 1951).
 45 "Mechanical Test Results on Various Extruded Materials", Allegheny Ludlum Steel Corporation (January 1, 1956).
 46 Westerman, R. E., and Sump, K. R., "Properties of Prealloyed Steel Powder Metallurgy Products - Final Report", Battelle Pacific Northwest

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

	Fe
0.3	C
0.95	Cr
0.2	Mo

4130

- 47 Laboratories for U.S. Army Weapons Command Contract MIPR A1-1-50248-M1-Ws (December 1972).
- 48 Troiano, A. R., and Hehemann, R. F., "Hydrogen Sulfide Stress Corrosion Cracking in Materials for Geothermal Power", *Materials Performance* (January 1979).
- 49 Paine, R. M., and Stonehouse, A. J., "A Corrosion Protection System for Beryllium in Aircraft Brake Applications", *Materials Performance* (August 1977).
- 50 Oberg, T. T., and Rooney, R. J., "Reversed Bending Fatigue Characteristics of Steel and High Strength Aluminum Alloys as Affected by Type of Specimen", U.S. Air Force Air Materiel Command Technical Report No. 5775 (July 1949).
- 51 Parker, E. R., and Zackay, V. F., "Microstructural Features Affecting Fracture Toughness of High Strength Steels", *Engineering Fracture Mechanics*, Vol 7 (1975).
- 52 Wood, W. E., Parker, E. R., and Zackay, V. F., "An Investigation of Metallurgical Factors Which Affect Fracture Toughness of Ultra-High Strength Steels", Army Materials and Mechanics Research Center Report No. CTR 73-24, Lawrence Berkeley Laboratory, University of California (May 1973).
- 53 McCall, J. L., "Metallography of Vacuum Brazed and Vacuum Heat Treated and Gas Quenched Materials", *Welding Journal, Welding Research Supplement* (February 1972).
- 54 "Mechanical Properties at Elevated Temperatures of As-Received Materials", Boeing Airplane Company, Manufacturing Report No. MDR 2-28014 (April 4, 1964).
- 55 Otto, H. E., and Mikesell, R., "Thermal Properties of Explosively Formed High Strength Low Alloy Steels", presented at the Third International Conference of the Center for High Energy Forming, Vail, Colorado (July 12-16, 1971).
- 56 Cambell, C. M., "Short Time Tensile Properties of Low Alloy Steels at Elevated Temperatures", North American Aviation Inc., Report No. NA-47-825 (October 15, 1947).
- 57 Bertke, R. S., "Impact Damage on Various Metal Leading Edges From Small Hard Objects", University of Dayton for Air Force Wright Aeronautical Laboratories, Report No. AFWAL-TR-81-44066 (August 1981).
- 58 Smith, C. R., "Tests Suggest Coating May Extend Fatigue Life of Mechanical Joints", *Assembly Engineering* (February 1976).
- 59 Brown, B. F., "AFPA Coupling Program on Stress-Corrosion Cracking", Naval Research Laboratory, Report No. 7329 (October 27, 1971).
- 60 Kramer, I. R., and Kumar, A., "The Effects of Surface Layer on Plastic Deformation and Crack Propagation", Martin Marietta Corporation for Army Materials and Mechanics Research Center, Report No. AMMRC CR 71-2/2 (August 1971).
- 61 Soderberg, E. W., and Roth, R. E., "Low Current Plasma Welding", General Motors Allison Division for Air Force Materials Laboratory, Report No. AFML-TR-66-177 (July 1967).
- 62 Harris, F. G., and Rowan, R. W., "Axial Load Fatigue Tests on Chromium Molybdenum Steel Tubing With Welded Fish-Mouth Type Splices", Australian Defence Scientific Service Aeronautical Research Laboratories, Report No. ARL/SM 323 (March 1969).
- 63 Degnan, W. G., Dripchak, P. D., and Matusovich, "Fatigue Crack Propagation in Aircraft Materials", Sikorsky Aircraft for U.S. Army Aviation Materiel Laboratories, USAAVLABS Technical Report No. 66-9 (March 1966).
- 64 Moon, D. P., "Effect of Temperature on the Short-Time Strength Properties of AISI 4130 and AISI 4340 Steels", Battelle Memorial Institute, Defense Metals Information Center, Technical Note (February 4, 1970).
- 65 Kirby, C. E., and Ransone, P. O., "Preliminary Investigation of Effects of Exposure to Sulfur Hexafluoride on Tensile and Yield Strengths of Aluminum and Steel", Langley Research Center, NASA Technical Memorandum X-2333 (September 1971).
- 66 Manson, S. S., and Muralidharan, V., "Fatigue Life Prediction in Bending From Axial Fatigue Information", Case Western Reserve University, NASA CR-165563 (February 1982).
- 67 Williams, D. P., and Nelson, H. G., "Embrittlement of 4130 Steel by Low-Pressure Gaseous Hydrogen", *Metallurgical Transactions*, Vol 1 (January 1970).
- 68 Ryniewicz, J., "Corrosion in the Deep Oceans", *Ocean Engineering Petroleum Engineer Special Edition* (November 15, 1974).
- 69 Sullivan, A. M., and Stoop, J., "Effect of Sheet Thickness on the Fracture-Resistance Parameter K_{IC} of Steels", Naval Research Laboratory, Report 7601 (August 8, 1973).
- 70 Loginow, A. W., and Phelps, E. H., "Steel for Seamless Hydrogen Pressure Vessels", *Corrosion-NACE* (November 1975).
- 71 Convair Division, General Dynamics Corp., "Heat Treatable Welds in Steel Plate", Report No. ZS-194 (September 30, 1953).
- 72 Freed, C. N., Sullivan, A. M., and Stoop, J., "Influence of Dimensions of the Center-Cracked Tension Specimen on K_{IC} ", ASTM STP 514 (1972).
- 73 Nguyen-Duy, P., "Effects of Loading Rate and Temperature on Fracture Toughness of 4130 Quenched and Tempered Steel Using Pre-cracked Charpy V-Notch Specimens", *Fracture Mechanics: Fourteenth Symposium, Volume II: Testing and Applications*, ASTM STP 791, J. C. Lewis and G. Sines, Eds., American Society for Testing and Materials (1983).
- 74 Campbell, J. E., "Plane-Strain Fracture-Toughness Data for Selected Metals and Alloys", Defense Metals Information Center, Report S-28, Battelle Memorial Institute (June 1969).
- 75 "Metals Handbook", 9th Edition, Vol 1, American Society for Metals (1978).

Alloy	4130				
Forms					
Sheet, Strip, Plate	Bars, Billets, Forgings	Pipe, Tube	Castings	Welding Wire	General
AMS 6350F	AMS 6348	AMS 6360G	AMS 5336D	AMS 6457	QQ-M-105b
AMS 6351C	AMS 6370J	AMS 6361B	MIL-S-22141		SAE 4130
AMS 6350C	QQ-S-621b	AMS 6362C	ASTM A487		SAE 4130H
QQ-S-626a	MIL-S-6758A	AMS 6371F			AISI 4130
QQ-S-627b	MIL-S-16974B	AMS 6373B			AISI 4130H
MIL-S-18729C	MIL-S-81242	AMS 6374			AISI TS4130
ASTM A505	ASTM A274	QQ-T-00825			AISI TS4130H
ASTM A507	ASTM A304	MIL-T-6736B			
ASTM A410	ASTM A322	ASTM A519			
	ASTM A372	ASTM A513			
	ASTM A322				
	ASTM A331				
	ASTM A274				
	ASTM A372				

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

TABLE 1.03. SPECIFICATIONS (1-13,28,29)

Alloy Form	4130							
	Wrought		Investment Castings (1)		Castings (29)		Welding Wire	
	Min	Max	Min	Max	Min	Max	Min	Max
Carbon	0.28(a)	0.33(a)	0.05	0.35	-	0.33	0.28	0.33
Manganese	0.40	0.60	0.40	0.80	0.60	1.00	0.40	0.60
Silicon	0.15(b)	0.35	-	1.00	-	0.80	0.15	0.35
Phosphorus	-	0.025	-	0.04	-	0.04	-	0.008(c)
Sulphur	-	0.025	-	0.04	-	0.045	-	0.008(c)
Chromium	0.80	1.10	0.80	1.10	0.75	1.10	0.80	1.10
Molybdenum	0.15	0.25	0.15	0.25	0.15	0.30	0.15	0.25
Nickel	-	0.25	-	0.25	-	0.50	-	0.25
Copper	-	0.35	-	0.25	-	-	-	0.10
Vanadium	-	-	-	-	-	-	-	0.06
Oxygen	-	-	-	-	-	-	-	0.0025
Nitrogen	-	-	-	-	-	-	-	0.005
Hydrogen	-	-	-	-	-	-	-	0.0010
Iron	Balance	Balance	Balance	Balance	Balance	Balance	Balance	Balance

(a) AMS 6336C gives 0.30 min and 0.35 max.

(b) AMS 6360G gives 0.20.

(c) Phosphorus + sulphur 0.012.

TABLE 1.04. COMPOSITION (1-13,29)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

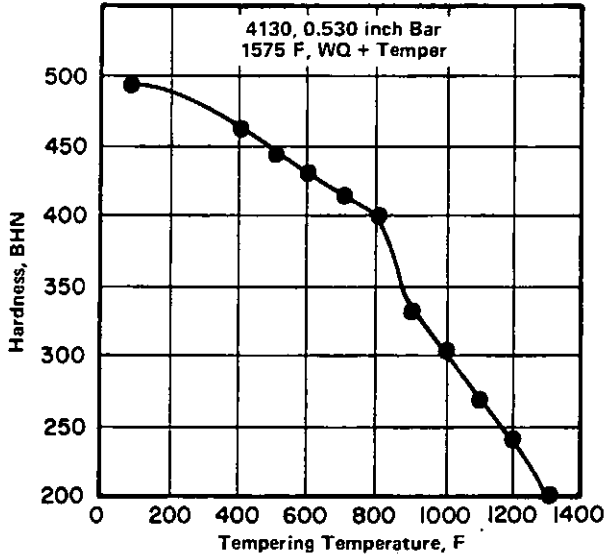


FIGURE 1.061. EFFECT OF TEMPERING TEMPERATURE ON HARDNESS OF BAR (16)

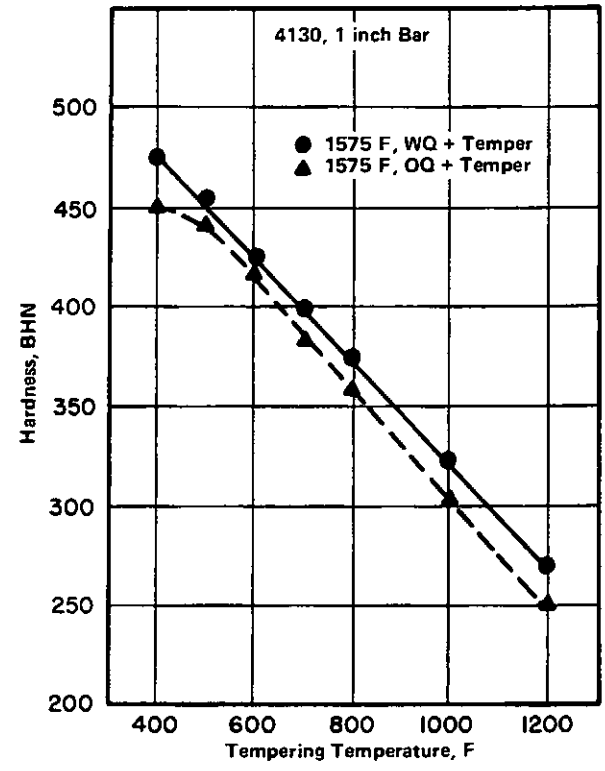


FIGURE 1.062. EFFECT OF TEMPERING TEMPERATURE ON HARDNESS OF OIL-QUENCHED AND WATER-QUENCHED BAR (30)

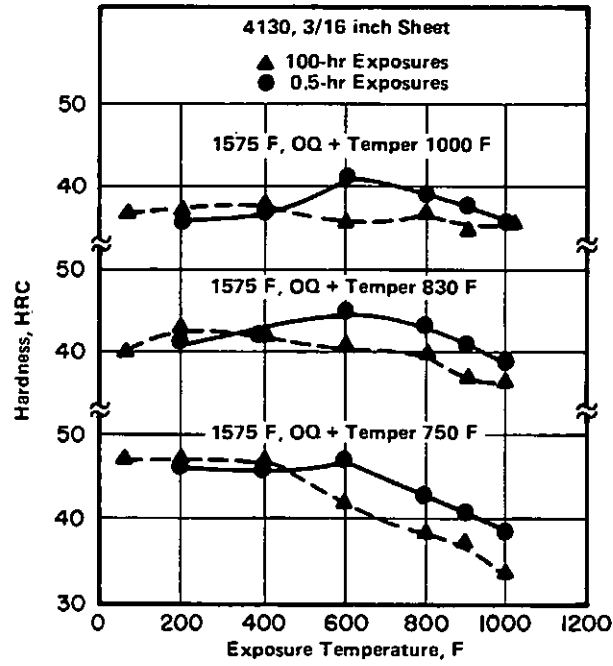


FIGURE 1.063. EFFECT OF EXPOSURES AT ELEVATED TEMPERATURES ON ROOM-TEMPERATURE HARDNESS OF SHEET AFTER THREE DIFFERENT TEMPERING TREATMENTS (14)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Alloy Form	Condition	4130 Hardness		
		BHN	HRB	HRC
5/8 in. Diam Bar	Normalized 1550 F, WQ, tempered 800 F, 1 hr	-	-	40
1 in. Bar	Hot rolled:			
	As hot rolled	229	-	-
	Normalized 1600 F	207	-	-
	Annealed 1550 F	179	-	-
	Cold drawn:			
As cold drawn	248	-	-	
Annealed 1550 F	201	-	-	
0.50 in. Plate	Normalized 1600 F	-	-	14
	Normalized + tempered 1100 F	-	-	12
	Oil quenched + tempered 1000 F	-	-	34
	Oil quenched + tempered 1200 F	-	-	22
1 in. Diam Extruded Bar, Allegheny Ludlum Steel	Annealed 1550 F, 1 hr + slow cool	-	80	-
	Normalized 1575 F, 1 hr, OQ and Temper 850 F, 2 hr	-	-	32
	Temper 1200 F, 2 hr	-	88	-
3/4 x 3-1/8 in. Extruded Flat Allegheny Ludlum Steel	Normalized 1650 F, 1 hr	-	92	-
	Annealed 1550 F, 1 hr	-	76	-
	Normalized 1575 F, 1 hr, OQ + Tempered 850 F, 2 hr	-	-	29
	Tempered 1200 F, 2 hr	-	93	-
0.063 in. Sheet	Austenitized 1575 F, 1 hr, WQ + temper 700 F, 1/2 hr, air cool	-	-	41
	Austenitized 1575 F, 1/2 hr, WQ + temper 500 F, 1/2 hr, air cool	-	-	46

TABLE 1.064. HARDNESS OF BAR, PLATE, SHEET AND EXTRUSION IN VARIOUS HEAT-TREATED CONDITIONS (11,26,30,44,45)

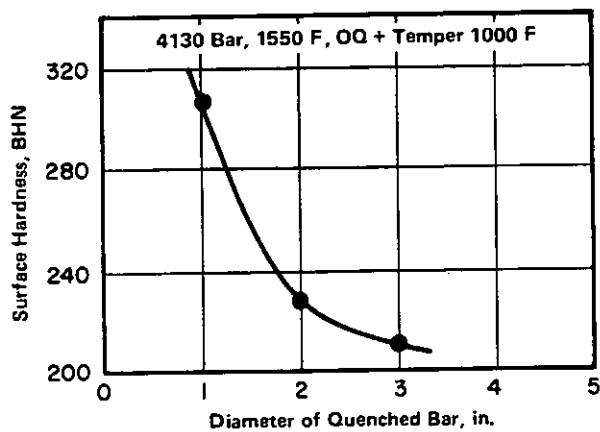


FIGURE 1.065. EFFECT OF DIAMETER ON SURFACE HARDNESS OF QUENCHED-AND-TEMPERED BARS (30)

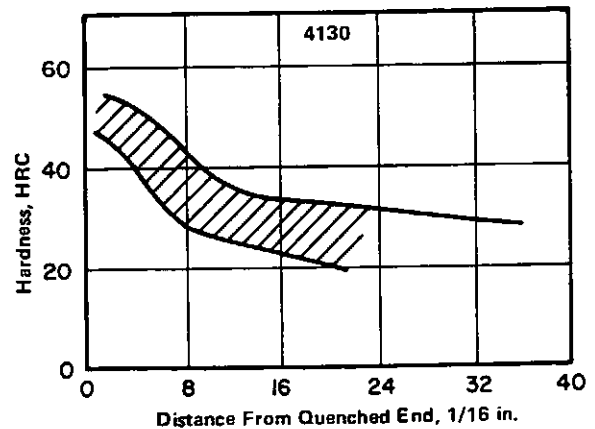


FIGURE 1.066. END-QUENCH HARDENABILITY (33)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

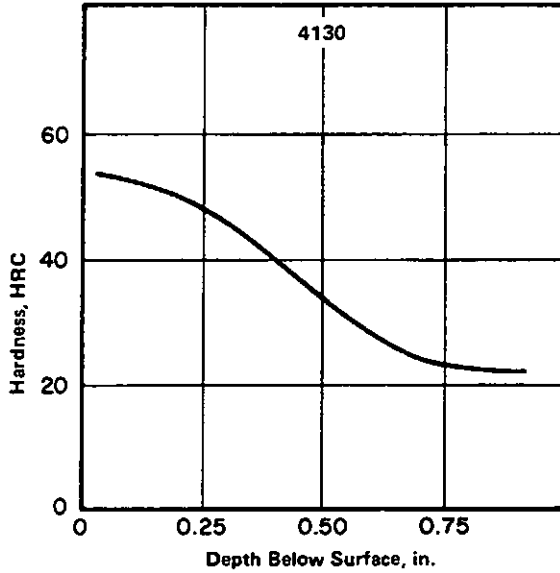


FIGURE 1.067. HARDNESS PROFILE THROUGH NITRIDED CASE (74)

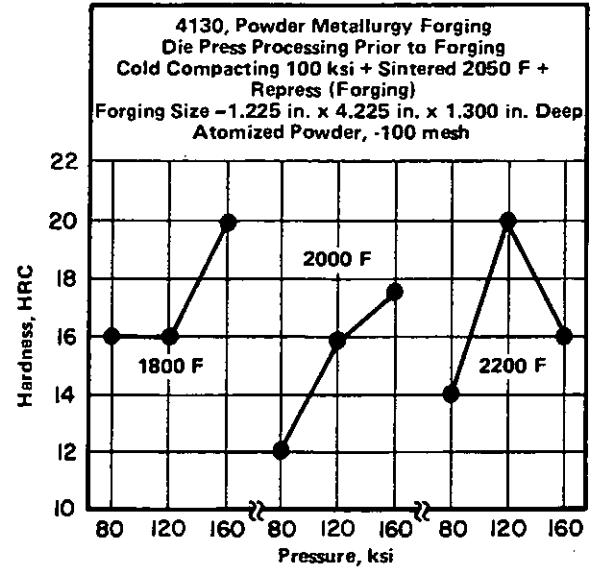


FIGURE 1.068. EFFECT OF REPRESS (FORGING) PRESSURE AND TEMPERATURE ON HARDNESS OF POWDER METALLURGY FORGING (46)

4130						
Alloy	4130					
Form	Powder Metallurgy Forgings, Die Press Process – 1.125 x 4.225 x 3.000 inch Deep, Isostatic Press Process – 0.565 x 4.000 x 1.300 inch Deep, High Energy Rate Forming (HERF), 2.5 inch Diameter x 5.5 inch Long Water Atomized Powder, -100 Mesh					
Condition	Austenitized 1575 F, RT Water Quench, Tempered 800 F, 1 hr					
Process	Compacting		Sinter	Repress (Forging)		Hardness, HRC
	Temperature, F	Pressure, ksi		Temperature, F(a)	Temperature, F	
Die Press	RT	100	2050	2200	150	31
Isostatic	RT	50	1650	2200	150	39
HERF	–	–	–	2200	275	35

(a) For 1 hour.

TABLE 1.069. EFFECT OF FABRICATION PROCESS ON HARDNESS PROPERTIES OF POWDER METALLURGY FORGING (46)

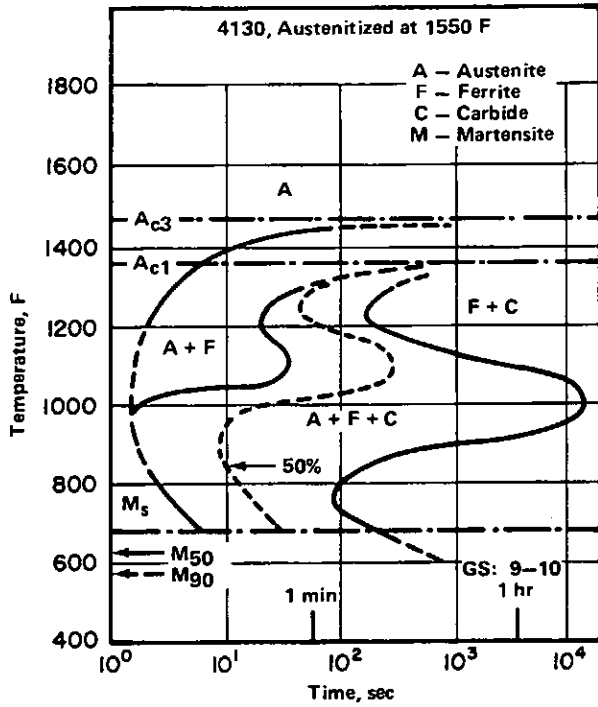


FIGURE 2.0121. TIME-TEMPERATURE-TRANSFORMATION DIAGRAM (34)

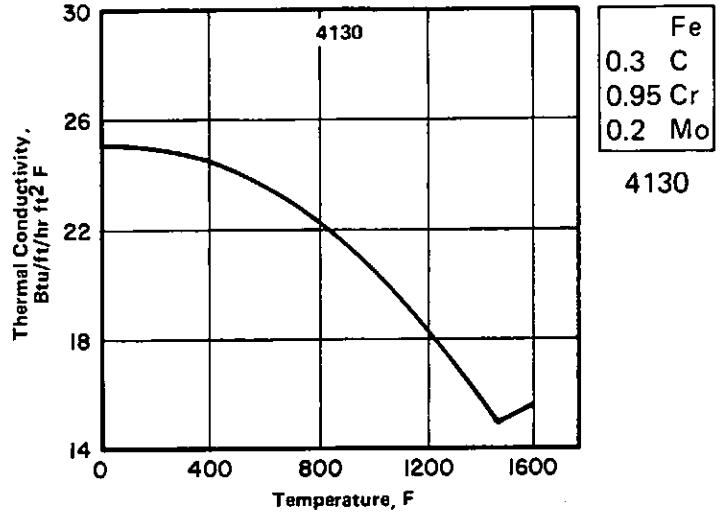


FIGURE 2.013. THERMAL CONDUCTIVITY (36)

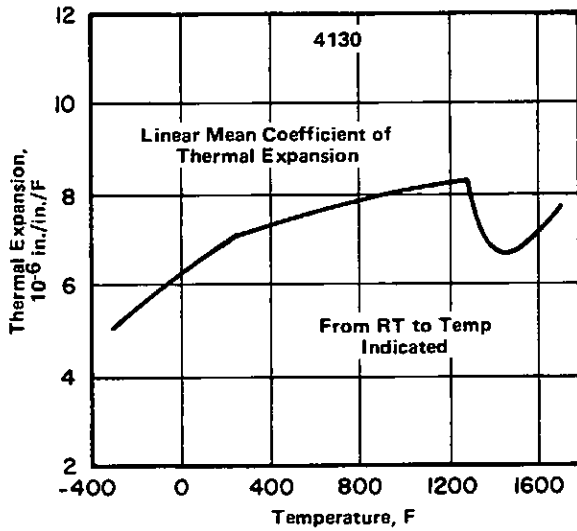


FIGURE 2.014. THERMAL EXPANSION (36)

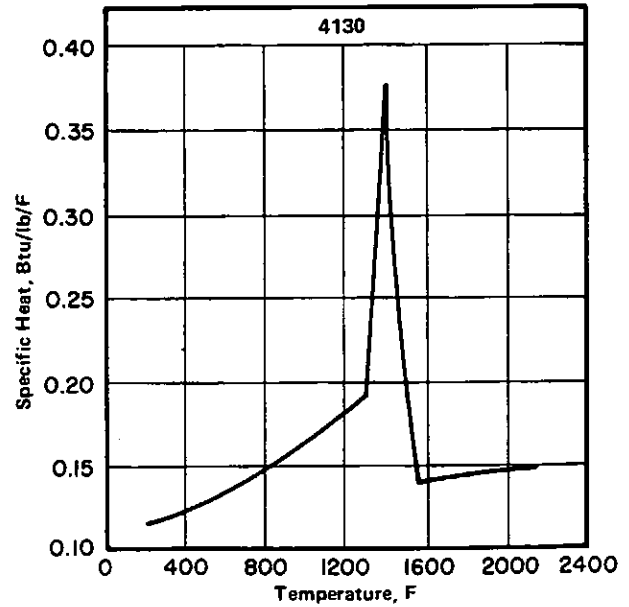


FIGURE 2.015. SPECIFIC HEAT (36)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

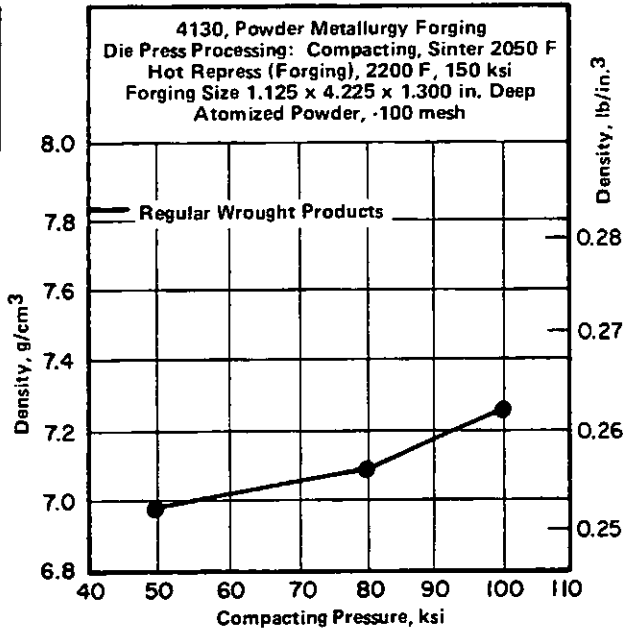


FIGURE 2.0211. EFFECT OF INITIAL COMPACTING PRESSURE ON DENSITY OF POWDER METALLURGY FORGING (46)

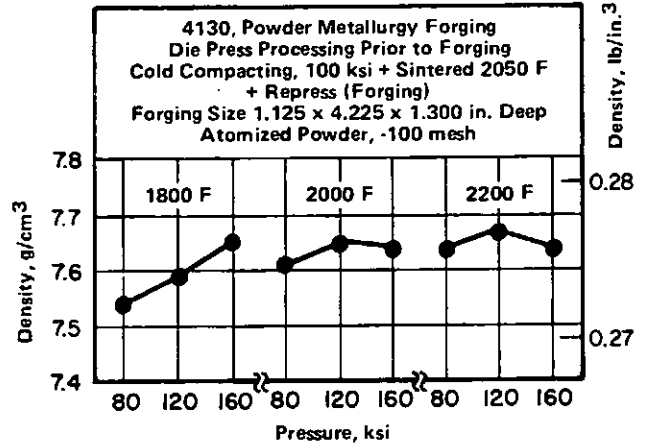


FIGURE 2.0212. EFFECT OF REPRESS (FORGING) PRESSURE AND TEMPERATURE ON DENSITY OF POWDER METALLURGY FORGING (46)

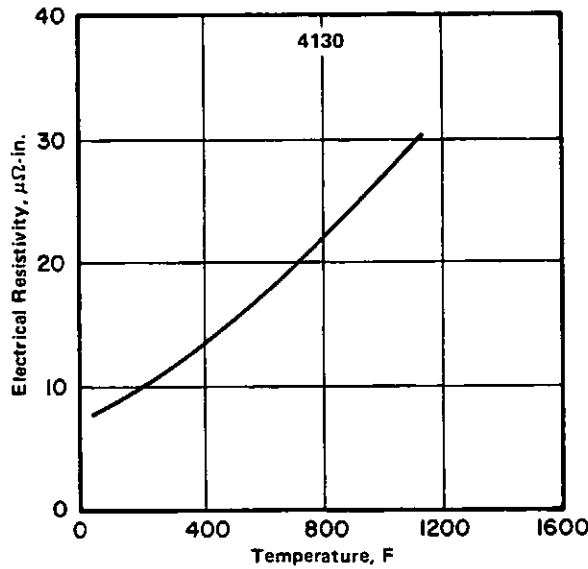
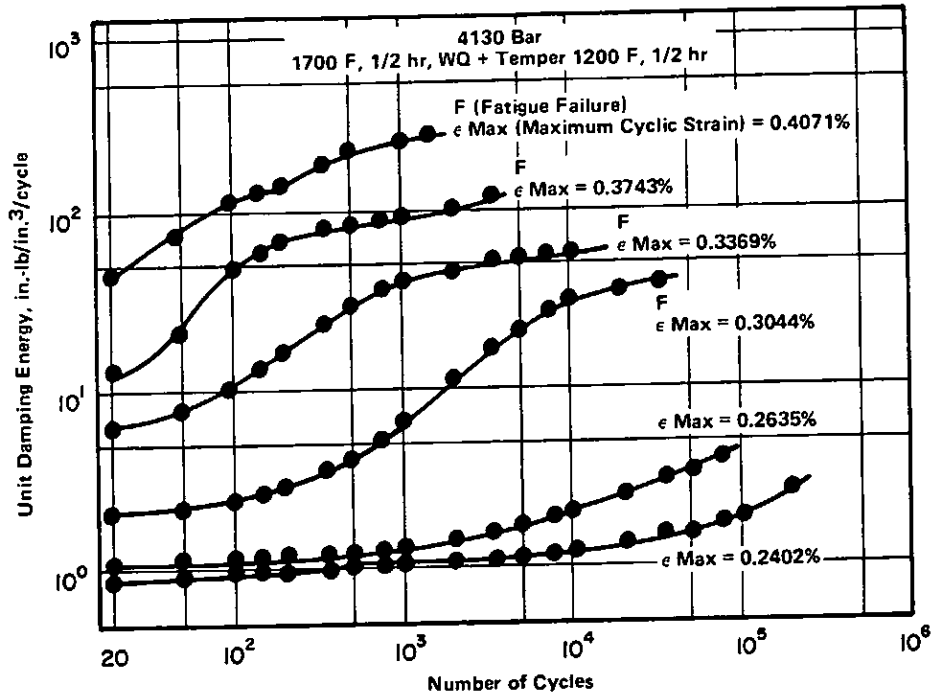


FIGURE 2.0221. ELECTRICAL RESISTIVITY AS A FUNCTION OF TEMPERATURE (36)



Fe
0.3 C
0.95 Cr
0.2 Mo

4130

FIGURE 2.0251. EFFECT OF NUMBER OF CYCLES IN REVERSED BENDING AND MAXIMUM CYCLIC TENSILE STRAIN (ϵ MAX) ON DAMPING CAPACITY (24)

4130							
Alloy Form	Condition	Orientation	F _{ty} , ksi	Applied Stress, ksi	No. Specimens	Days to Failure	
2-1/2 inch Diam Bar	A(a)	T	155	116	3	-(c)	
			155	155	3	-	
		L	154	116	3	-	
	B(b)	T	L	154	154	3	-
				161	121	3	-
		T	L	161	161	3	-
				148	111	3	-
		L	L	148	148	3	-
				148	148	3	-
0.060 inch Sheet	A	T	142	107	3	-	
			142	142	3	-	
		L	147	110	3	-	
	B	T	L	147	147	3	-
				171	124	3	-
		T	L	171	171	3	109,134, 144
				171	171	3	-
		L	L	171	128	3	-
				171	171	3	28,56

- (a) 1575 F, WQ + temper 900 F.
- (b) 1575 F, WQ + temper 800 F.
- (c) No cracking in maximum exposure time of 180 days.

TABLE 2.032. STRESS CORROSION OF CADMIUM-PLATED BAR AND SHEET SUBJECTED TO ALTERNATE IMMERSIONS (10 MINUTES IN AND 50 MINUTES OUT) IN 3.5 PERCENT SOLUTION OF SODIUM CHLORIDE HAVING pH OF 6.8 TO 7.2 (21)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Alloy	4130	
Form	0.035 inch Sheet	
Condition	Normalized and Tempered to Various Yield Strengths	
Specimen Orientation	L	
Temperature, F	Chloride Concentration(a), percent	Yield Strength Break Point(b), ksi
78	0	113
	1	110
	5	110
	20	118
215	5	140
	20	142
425	5	140
	20	138

(a) Modified NACE Solution, which is deaerated aqueous solution of 5% NaCl, 0.5% acetic acid, saturated with a continuous flow of H₂S.

(b) "Break Point" is the highest value of RT yield strength which survived the environment when loaded to approximately its yield strength in a two-point bend test (ASTM G39-73). The yield strengths were varied in the range 100 to 150 ksi by using different tempering times and temperatures.

TABLE 2.034. EFFECT OF TEMPERATURE AND CHLORIDE CONCENTRATION ON "BREAK POINT" OF SHEET IN TWO-POINT BEND TEST (47)

Alloy		4130						
Source	Form	Condition	F _{ty} , ksi (Min)	F _{tu} , ksi (Min)	e (2 in.), percent (Min)	RA, percent (Min)	Hardness	
							Min	Max
AMS 5336D (1)	Casting, investment	Normalized or normalized and tempered Plus: 1600 F > 30 min, OQ, temper at 825 F > 1 hr, air cool, 2nd temper 825 F > 1 hr, air cool	125(a)	150(a)	5(a)	10(a)	30 HRC	38 HRC(b)
AMS 6350F (3)	Sheet, strip Plate <0.250 in. sheet, strip, plate	Annealed Hot rolled or annealed 1600 F, OQ + 900 F, 30 min	-	125	-	-	26 HRC	98 HRB 25 HRC
AMS 6351C (4)	Sheet, strip, plate <0.250 in. sheet, strip, plate	Spheroidized 1600 F, OQ + 900 F, 30 min	-	125	-	-	26 HRC	85 HRB
AMS 6356C (5)	Sheet and strip Plate	Cold finished and annealed or hot rolled and annealed Hot rolled, annealed if necessary	-	-	-	-	-	95 HRB
AMS 6360G (6)	Tube: Up to 0.500 in. OD <0.188 in. wall >0.188 in. wall 0.500 in. and over OD <0.188 in. wall >0.188 in. wall	Cold drawn and stress relieved or normalized	75 70	95 90	10 10	-	-	-
AMS 6361B (7)	Tube	1550 F, Q + temper	100	125	12(c)	-	-	-
AMS 6362C (8)	Tube	1550 F, Q + temper	135	150	10(c)	-	-	-
AMS 6770J (9)	Bars 0.500 in. diam and under Over 0.500 in. diam Over 0.500 in. diam	Cold finished Hot finished Cold finished	130 (Max)	-	-	-	-	229 BHN 241 BHN
AMS 6371F (10)	Tube	Cold finished Hot finished	-	-	-	-	-	25 HRC 99 HRB
ASTM A487 (29)	Castings	Normalized + tempered Quenched + tempered	60 85	90 105	20 16	35 35	-	-

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

- (a) From separately cast specimens. For properties from casting, see AMS 2360.
- (b) From casting or (a).
- (c) Elongation values pertain to full tube; strip samples 5 percent lower.

TABLE 3.01. SPECIFIED MECHANICAL PROPERTIES

Alloy		4130			
Form		Bar			
Bar Diameter	Condition	F _{ty} , ksi	F _{tu} , ksi	e (2 in.), percent	RA, percent
1 inch	Hot Rolled:				
	As hot rolled	70	116	22	53
	Normalized 1600 F	65	108	27	57
	Annealed 1550 F	60	88	30	65
	Cold Drawn:				
As cold drawn	105	122	16	45	
Annealed 1550 F	82	98	20	53	
5/8 inch	Cold Drawn: Normalized 1550 F, water quenched, tempered 800 F, 1 hr	164.8(a)	186.1	11.8	59.3

(a) Proportional limit.

TABLE 3.0212. TENSILE PROPERTIES OF ROUND BAR IN VARIOUS HEAT-TREATED CONDITIONS (30,44)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Alloy		4130			
Form		Plate			
Thickness, in.	Condition	Specimen Orientation	F _{ty} , ksi	F _{tu} , ksi	Elongation (in 2 In.), percent
0.50	Normalized 1600 F	L	65.2	94.8	30.8
		T	64.4	94.4	26.2
	Normalized + temper 1100 F	L	64.4	94.0	31.8
		T	63.2	93.8	26.2
	Oil quench + temper 1000 F	L	113.8	130.8	18.8
		T	128.4	143.2	13.8
	Oil quench + temper 1200 F	L	97.4	113.2	24.8
	T	94.4	110.0	20.2	
>1.00	Normalized 1600 F, 2 hr, oil quench, temper 1175 F, 2 hr, air cool	T	92	119	21

TABLE 3.0213. TENSILE PROPERTIES OF PLATE IN VARIOUS HEAT-TREATED CONDITIONS (26,69)

Alloy		4130		
Form		3/8 inch Plate		
Specimen Orientation		Longitudinal		
Condition	F _{ty} , ksi	F _{tu} , ksi	Elongation (in 0.5 In.), percent	
Normalized 1625 F, then				
Tempered to Rc 26	118	129	17.0	
Tempered to Rc 34	143	150	14.5	
Tempered to Rc 44	194	206	10.0	

TABLE 3.0214. TENSILE PROPERTIES OF 3/8 INCH PLATE HEAT TREATED TO VARIOUS HARDNESSES (49)

Alloy		4130			
Form		1-inch Diameter Extruded Bar			
Composition		C-0.31, Mn-0.58, Si-0.23, P-0.009, S-0.014, Cr-0.99, Ni-0.95, Mo-0.21			
Condition	F _{ty} , ksi	F _{tu} , ksi	e (2 in.), percent	RA, percent	
1550 F, 1 hr + Slow Cool	46.5	85.2	27.0	56.3	
1575 F, 1 hr + OQ +					
Tempered 850 F, 2 hr	145.5	167.0	14.0	55.0	
Tempered 1200 F, 2 hr	64.5	82.5	31.0	72.6	
Composition		C-0.30, Mn-0.55, Si-0.35, P-0.005, S-0.005, Cr-0.98, Ni-0.11, Mo-0.20			
1550 F, 1 hr + Slow Cool	48.0	83.5	27.5	52.6	
1575 F, 1 hr + OQ +					
Tempered 850 F, 2 hr	160.5	174.0	14.0	51.1	
Tempered 1200 F, 2 hr	90.0	110.8	23.5	70.2	

TABLE 3.0215. TENSILE PROPERTIES OF EXTRUDED ROUND BAR IN VARIOUS HEAT-TREATED CONDITIONS (45)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Alloy	4130				
Form	3/4 inch x 3-1/8 inch Extruded Flat				
Composition	C-0.30, Mn-0.50, Si-0.27, P-0.016, S-0.016, Ni-0.11, Cr-0.99, Mo-0.18				
Condition	Specimen Orient.	F _{ty} , ksi	F _{tu} , ksi	e (1 in.), percent	RA, percent
1650 F, 1 hr + Air Cool	L	84.2	115.8	23.5	56.4
	T	81.8	111.0	21.2	54.2
1575 F, 1 hr + OQ + Tempered 850 F, 2 hr	L	157.6	164.9	16.5	57.8
	T	154.0	159.2	11.5	49.6
Tempered 1200 F, 2 hr	L	86.1	103.2	23.5	73.0
	T	84.3	100.2	23.5	56.4
Composition	C-0.31, Mn-0.58, Si-0.23, P-0.009, S-0.014, Cr-0.99, Ni-0.95, Mo-0.21				
1550 F, 1 hr + Slow Cool	L	41.3	76.0	30.0	55.9
	T	40.6	73.6	25.5	52.4
1575 F, 1 hr + OQ + Tempered 850 F, 2 hr	L	155.5	165.5	15.5	58.8
	T	153.0	162.0	12.0	42.8
Tempered 1200 F, 2 hr	L	53.6	77.3	36.5	73.2
	T	58.7	74.8	29.5	61.4
Composition	C-0.30, Mn-0.55, Si-0.35, P-0.005, S-0.005, Cr-0.98, Ni-0.11, Mo-0.20				
1550 F, 1 hr + Slow Cool	L	42.6	77.3	30.0	55.1
	T	41.0	75.4	28.0	43.8
1575 F, 1 hr + OQ + Tempered 850 F, 2 hr	L	160.0	167.5	16.0	59.4
	T	153.6	163.0	14.5	41.0
Tempered 1200 F, 2 hr	L	86.6	104.0	25.0	71.6
	T	84.4	101.8	20.0	59.0

TABLE 3.0216. TENSILE PROPERTIES OF EXTRUDED FLAT BAR IN VARIOUS HEAT-TREATED CONDITIONS (45)

Alloy	4130			
Form	Castings			
Condition	F _{ty} , ksi	F _{tu} , ksi	e (2 in.), percent	RA, percent
Normalized 1650 F and Tempered 1100 F Tempered 1200 F Tempered 1250 F	95.0	115.0	16.0	35.0
	61.5	93.5	23.0	37.3
	56.5	88.0	22.0	38.0
1575 F, WQ and Tempered 950 F Tempered 1100 F Tempered 1175 F	148.0	176.5	10.0	24.6
	120.0	140.0	17.0	38.5
	95.0	115.0	19.5	42.0

TABLE 3.0217. TENSILE PROPERTIES OF CASTINGS IN VARIOUS HEAT-TREATED CONDITIONS (30)

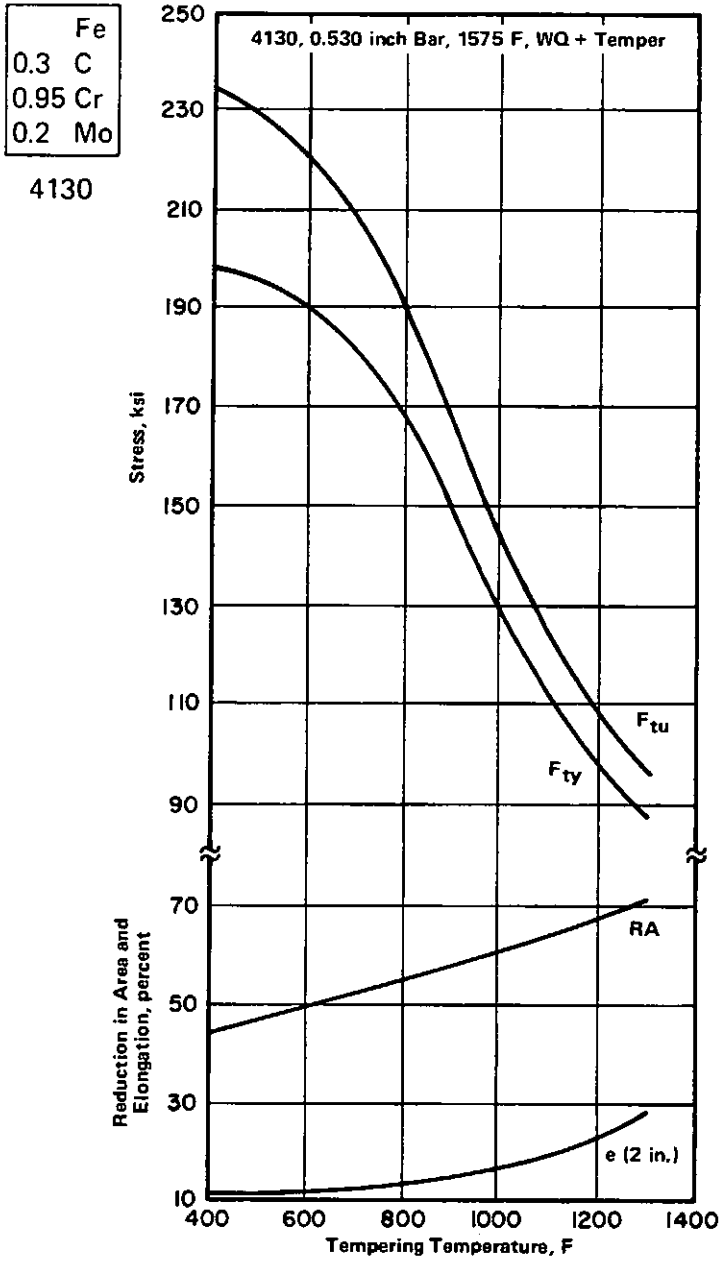


FIGURE 3.0218. EFFECT OF TEMPERING TEMPERATURE ON ROOM TEMPERATURE TENSILE PROPERTIES OF 0.530 INCH BAR (16)

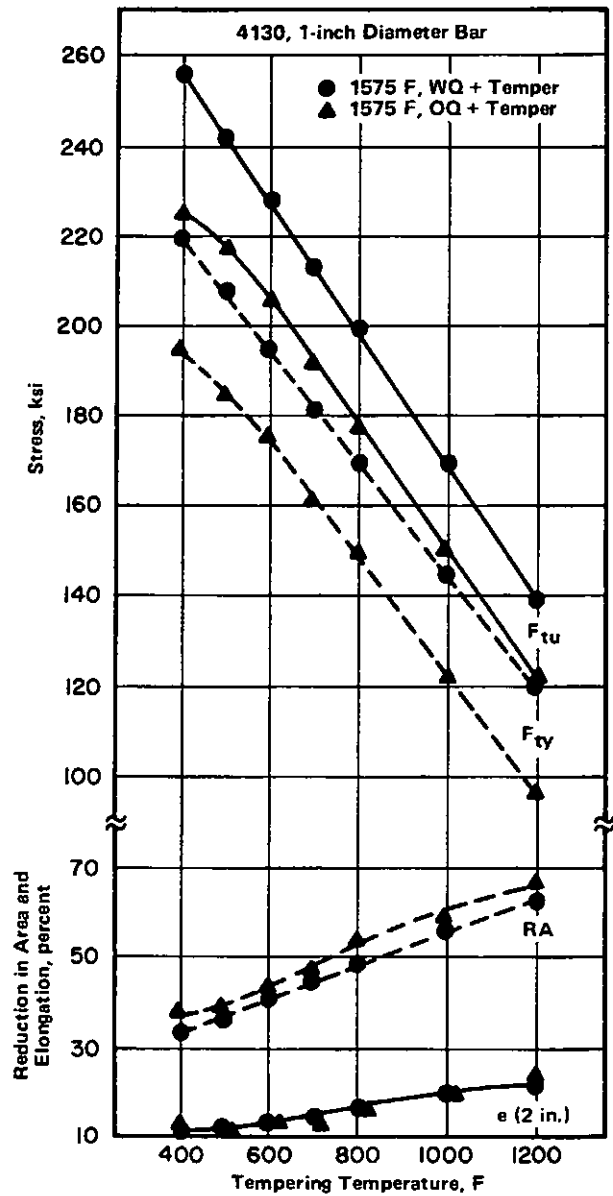


FIGURE 3.0219. EFFECT OF TEMPERING TEMPERATURE ON TENSILE PROPERTIES OF OIL-QUENCHED AND WATER-QUENCHED 1 INCH BAR (30)

Alloy	4130					
Form	Sheet					
Condition	Specimen Orient.	Thickness, in.	F _{ty} , ksi	F _{tu} , ksi	e (1 in.), percent	RA, percent
Austenitized 1575 F, 1 hr, WQ, Tempered 700 F, 30 min, Air Cool	T	0.030	167.8	183.4	—	7.1
		0.050	171.6	197.6	3.5	7.5
		0.063	169.5	193.0	3.8	11.2
		0.087	183.4	205.2	3.0	6.5
		0.125	176.4	197.9	4.5	16.5
		0.250	173.7	195.3	7.5	36.8
Austenitized 1575 F, 1/2 hr, WQ, Tempered 500 F, 30 min, Air Cool	T	0.030	185.2	221.4	2.5	7.1
		0.050	184.6	219.8	3.0	10.7
		0.063	178.4	226.0	5.2	27.8
		0.087	200.3	233.2	3.5	20.1
		0.125	191.2	228.1	6.0	23.2
		0.250	185.4	221.5	—	36.9
Austenitized 1650 F, 1 hr, OQ, Tempered 800 F, 1 hr	L	0.125	175(a)	—	—	—

Fe
0.3 C
0.95 Cr
0.2 Mo
4130

(a) Proportional limit = 163 ksi.

TABLE 3.02110. EFFECT OF SHEET THICKNESS AND TEMPERING TEMPERATURE ON TENSILE PROPERTIES (59,68)

Alloy	4130			
Form	5/8 inch Thick Flat Bar			
Austenitizing Temperature, F, and Quench Medium	F _{ty} , ksi	F _{tu} , ksi	e (4D), percent	RA, percent
1600, Oil	201	284	11.4	32.9
2200, Oil	205	276	5.6	9.9
2200, Ice-Brine	215	249	1.6	6.6

TABLE 3.02111. EFFECT OF AUSTENITIZING TEMPERATURE AND QUENCH MEDIUM ON TENSILE PROPERTIES OF AS-QUENCHED MATERIAL (50)

Alloy	4130					
Form	3/8-inch Thick Rolled Flat Bar					
Specimen Orient.	Longitudinal					
Austenitizing Temperature, Time	Quench Medium	Temper Temperature, F	F _{ty} , ksi	F _{tu} , ksi	e (1.4 in.), percent	RA, percent
2200 F, 1 hr	IBLN(a)	As quenched	167	245	—(b)	—
		392	186	257	—(b)	—
		536	184	225	7.6	21.0
		662	180	209	9.8	37.0
2200 F, 1 hr	Water	As quenched	167	278	5.2	14.0
		392	210	258	9.7	—
		536	182	228	10.3	38.7
		662	180	210	9.8	25.2
2200 F, 1 hr	Oil	As quenched	161	276	5.6	9.8
		392	176	262	7.5	18.1
		536	194	246	8.0	25.5
		662	182	222	7.7	30.6
2200 F, 1 hr, Cooled to 1600 F, 1/2 hr	IBLN	As quenched	170	279	6.6	12.0
		392	195	261	—(b)	—
		536	184	230	7.8	25.9
		662	200	210	9.3	32.4

(a) Ice-brine followed by liquid nitrogen storage.
(b) Quench cracks visible.

TABLE 3.02112. EFFECT OF QUENCH MEDIUM AND TEMPERING TEMPERATURE ON TENSILE PROPERTIES OF ROLLED BAR (51)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Alloy	4130		
Form	0.125 inch Sheet		
Condition	Austenitized 1575 F, 15 min, Quenched, Tempered 1050 F, 1 hr		
Quench Medium	F _{ty} , ksi	F _{tu} , ksi	Elongation, percent
Oil	146.0	156.4	14.0
Nitrogen Gas	147.0	162.0	13.0

TABLE 3.02113. EFFECT OF QUENCH MEDIUM ON TENSILE PROPERTIES OF SHEET (52)

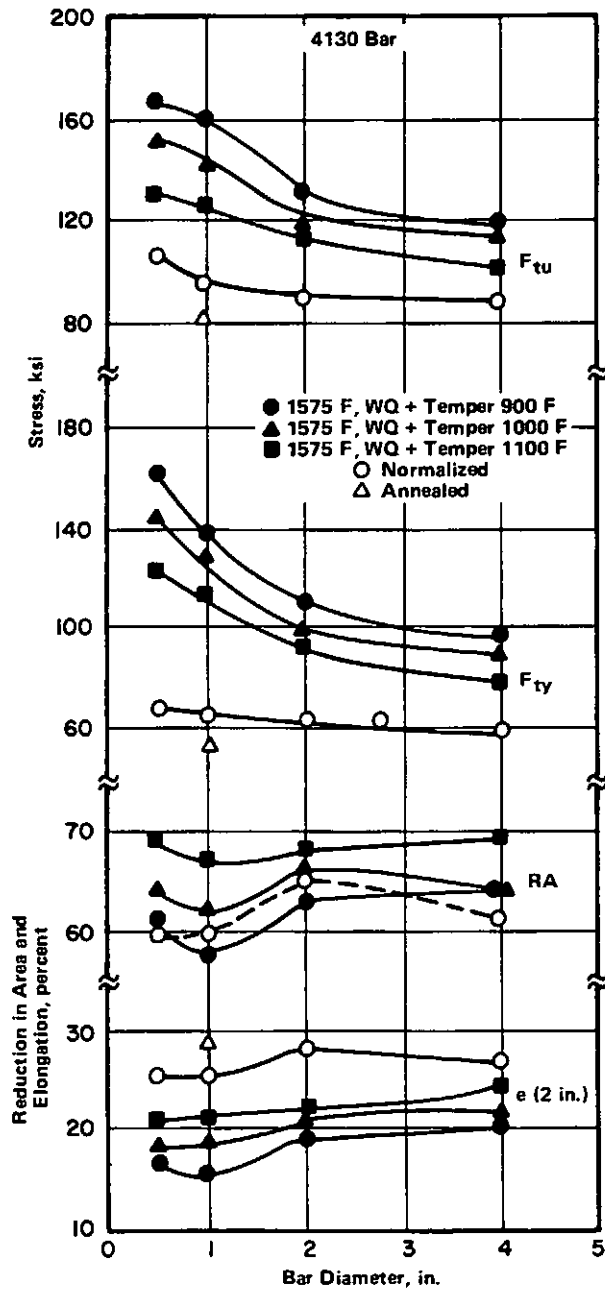


FIGURE 3.02114. EFFECT OF BAR DIAMETER ON TENSILE PROPERTIES IN VARIOUS HEAT-TREATED CONDITIONS (16)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

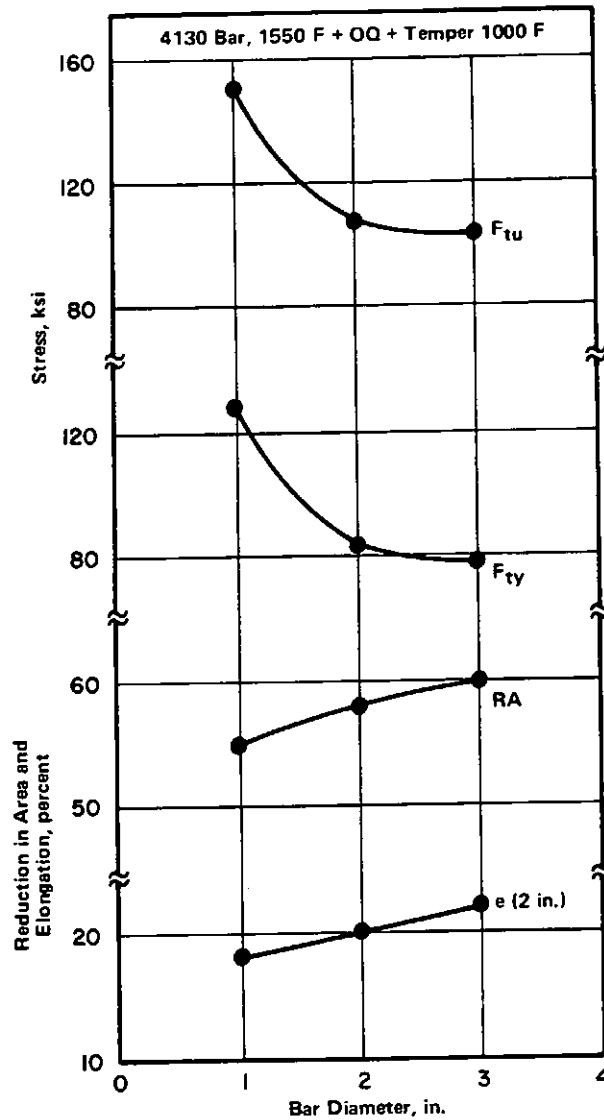


FIGURE 3.02115. TENSILE PROPERTIES OF QUENCHED AND TEMPERED ROUND BAR (74)

Alloy		4130				
Form		1/4 inch Plate				
Orientation		Longitudinal				
Condition	Treatment	Effective Strain	F_{ty} , ksi	F_{tu} , ksi	e (1 in.), percent	RA, percent
Annealed	None	0	66.5	97.1	29.5	60.2
	Explosively formed	0.068	90.7	98	26.5	43.2
	Cold rolled	0.066	68.1	96.7	26.0	43.3
Heat Treated(a)	None	0	149	155	10.5	39.1
	Explosively formed	0.068	148	155	12.3	41.3
	Cold rolled	0.066	140	149	12.5	35.7

(a) Austenitized 1600 F, 40 min, oil quenched, tempered 1000 F, 55 min.

TABLE 3.02116. EFFECT OF EXPLOSIVE FORMING ON TENSILE PROPERTIES OF PLATE (54)

Fe 0.3 C 0.95 Cr 0.2 Mo 4130	Alloy	4130						
	Form	Powder Metallurgy Forgings Die Press Process 1.125 x 4.225 x 3.000 inch Isostatic Press Process 0.565 x 4.000 x 1.300 inch High Energy Rate Forming (HERF) 2.5 inch Diam x 5.5 inch Long Water Atomized Powder, -100 mesh						
Condition	Austenitized 1575 F, RT, Water Quench, Tempered 800 F, 1 hr							
Process	Compacting		Sinter	Repress (Forging)		F _{ty} , ksi	F _{tu} , ksi	e (1 in.), percent
	Temperature, F	Pressure, ksi	Temperature, F(a)	Temperature, F	Pressure, ksi			
Die Press	RT	100	2050	2200	150	132	136	1.4
Isostatic Press	RT	50	1650	2200	150	(b)	153	0.5
HERF	-	-	-	2200	275	136	160	1.4

(a) For 1 hour.
 (b) Did not have sufficient ductility.

TABLE 3.02117. EFFECT OF FABRICATION PROCESS ON TENSILE PROPERTIES OF POWDER METALLURGY FORGING (46)

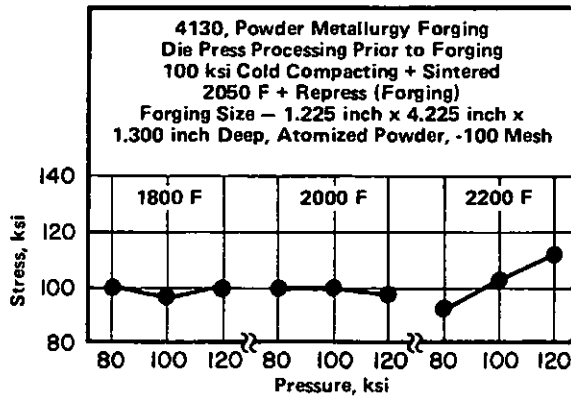


FIGURE 3.02118. EFFECT OF REPRESS (FORGING) PRESSURE AND TEMPERATURE ON ULTIMATE TENSILE STRENGTH OF POWDER METALLURGY FORGING (46)

Alloy	4130				
Form	0.030 inch Sheet				
Condition	F _{ty} , ksi	F _{tu} , ksi	e (1 in.), percent	RA, percent	Prop. Limit, ksi
As received	80.5	106.2	13.2	47	69.9
HT 1800 F, 1 hr	34.8	66.2	23.8	57	29.0
Mn plated	81.5	106.3	12.8	39	77.6
HT 1800 F, 1 hr	37.2	75.3	20.2	34	33.4

TABLE 3.02119. EFFECT OF Mn PLATING AND ELEVATED TEMPERATURE EXPOSURE ON TENSILE PROPERTIES OF SHEET (48)

FERROUS ALLOYS

Alloy	4130	
Condition	Quenched and Tempered	
	F _{tu} , ksi(a)	
125	150	180
925 F	775 F	575 F

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

(a) If the material is exposed for more than 10 hours to temperatures exceeding those listed (unstressed), a permanent reduction in strength is likely to occur. The temperatures are approximately 125 F below the tempering temperature used to achieve the various strength levels.

TABLE 3.02120. TEMPERATURE LIMITS FOR PROLONGED UNSTRESSED EXPOSURE WHICH PREVENT DEGRADATION OF TENSILE STRENGTH (63)

Alloy	4130
Form	3/4 x 3-1/2 inch Extruded Bar
Condition	As Extruded
Specimen Orientation	Keyhole Charpy Impact, ft-lb
L	68
T	36

TABLE 3.0231. IMPACT PROPERTIES OF EXTRUDED BAR (45)

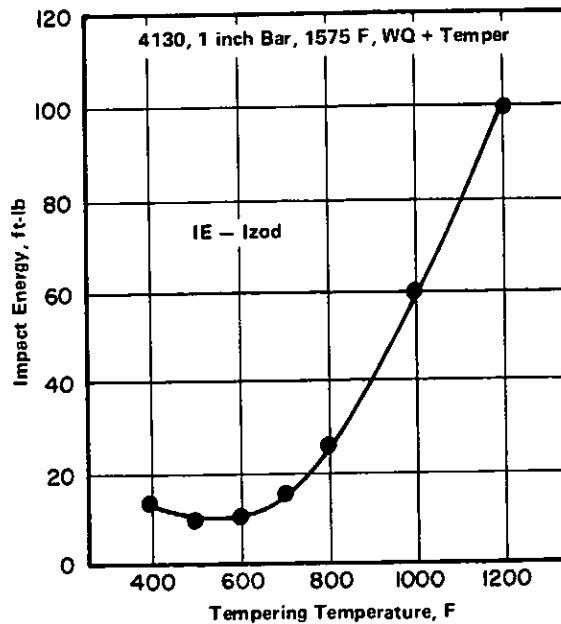


FIGURE 3.0232. EFFECT OF TEMPERING TEMPERATURE ON ROOM TEMPERATURE IMPACT STRENGTH (30)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Alloy	4130	
Form	Wrought	
Condition	Izod Impact, ft-lb	F _{tu} , ksi
Normalized	63.7	97.0
Annealed	45.5	81.2

TABLE 3.0233. IMPACT RESISTANCE OF NORMALIZED AND ANNEALED MATERIAL (31)

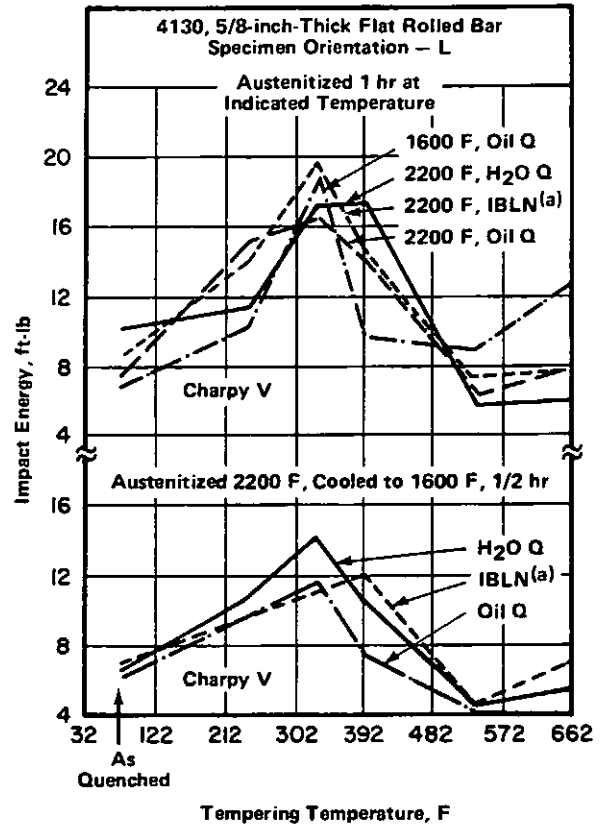


FIGURE 3.0234. EFFECT OF QUENCH MEDIUM AND TEMPERING TEMPERATURE ON IMPACT PROPERTIES OF ROLLED BAR (51)

(a) Ice-brine followed by liquid N₂ storage.

Alloy	4130								
Form	Powder Metallurgy Forgings Die Press Process - 1.125 inch x 4.225 inch x 1.300 inch Deep Isostatic Press Process - 0.565 inch x 4.000 inch x 1.300 inch Deep High Energy Rate Forming (HERF) - 2.5 inch Diameter x 5.5 inch Long Water Atomized Powder - 100 mesh								
Condition	Austenitized 1575 F, RT, H ₂ O Quench, Tempered 800 F, 1 hr								
Process	Compacting		Sinter	Repress (Forging)		Charpy-V Notched Impact Energy, ft-lb			
	Temperature, F	Pressure, ksi	Temperature, F(a)	Temperature, F	Pressure, ksi	-40 F	77 F	210 F	450 F
Die Press	RT	100	2050	2200	150	4.5	3.2	7.3	5.9
Isostatic Press	RT	50	1650	2200	150	1.0	2.3	2.1	6.2
HERF	-	-	-	2200	275	1.7	1.3	1.2	1.3

(a) For 1 hour.

TABLE 3.0235. EFFECT OF FABRICATION PROCESS AND TEMPERATURE ON IMPACT PROPERTIES OF POWDER METALLURGY FORGING (46)

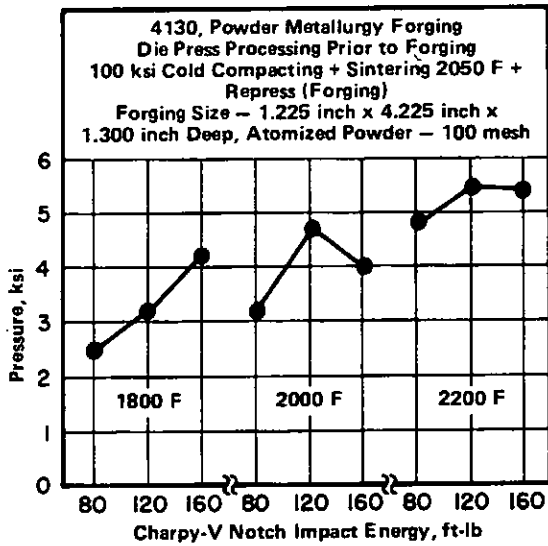


FIGURE 3.0236. EFFECT OF REPRESS (FORGING) PRESSURE AND TEMPERATURE ON IMPACT PROPERTIES OF POWDER METALLURGY FORGING (46)

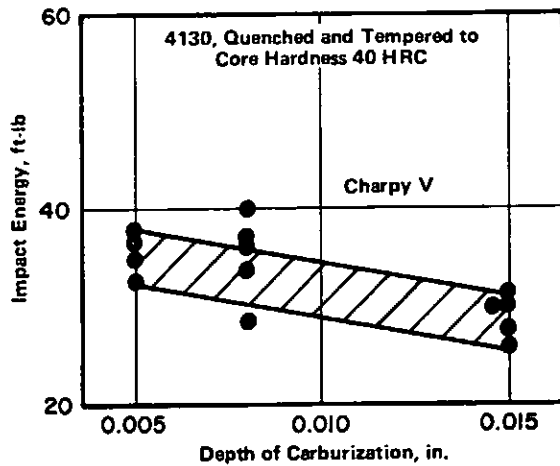


FIGURE 3.0237. EFFECT OF CARBURIZATION ON IMPACT PROPERTIES (74)

Fe
0.3 C
0.95 Cr
0.2 Mo
4130

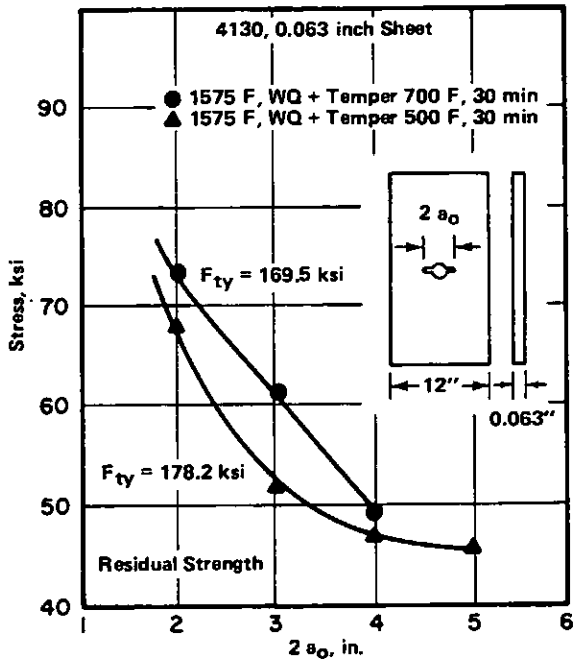


FIGURE 3.02711. EFFECT OF INITIAL CRACK LENGTH ($2 a_0$) ON RESIDUAL STRENGTH (MAXIMUM LOAD DIVIDED BY GROSS CROSS-SECTIONAL AREA) OF SHEET AFTER TWO DIFFERENT TEMPERING TREATMENTS (17)

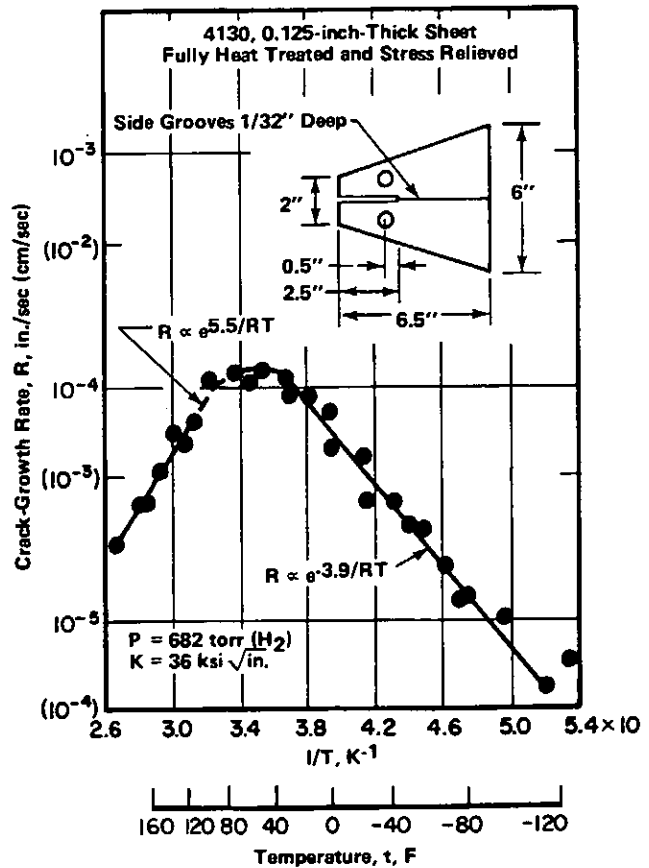


FIGURE 3.02712. EFFECT OF TEMPERATURE ON SLOW-CRACK-GROWTH RATE IN GASEOUS HYDROGEN (66)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

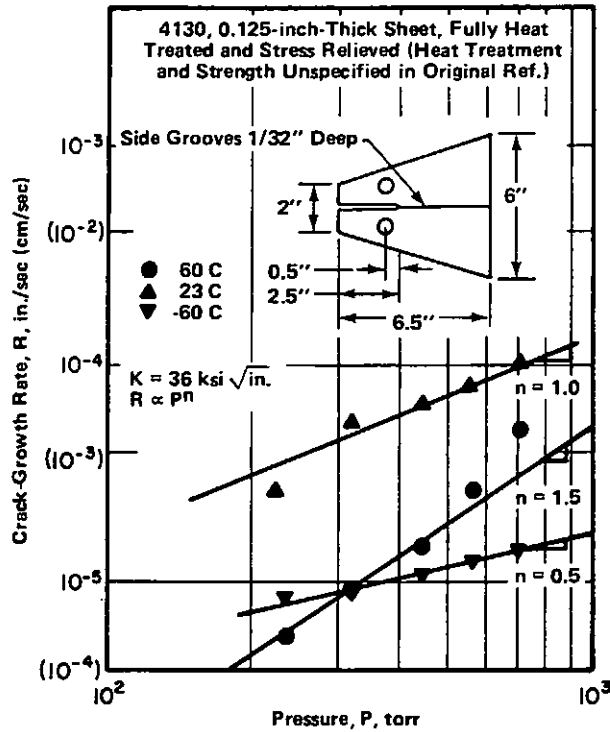


FIGURE 3.02713. EFFECT OF HYDROGEN PRESSURE ON SLOW-CRACK-GROWTH RATE (66)

Alloy	4130		
	Form	Sheet	
		Sheet Thickness, in.	K _c , ksi√in.
Condition		Blunt	Sharp
Austenitized 1575 F, 1 hr, WQ, Temper 700 F, 30 min, Air Cool	0.030	136	148
	0.050	140	173
	0.063	158	182
	0.087	126	134
	0.125	148	153
Austenitized 1575 F, 1/2 hr, WQ, Temper 500 F, 30 min, Air Cool	0.030	-	158
	0.050	149	152
	0.063	141	129
	0.087	122	125
	0.125	165	146
	0.250	124	142

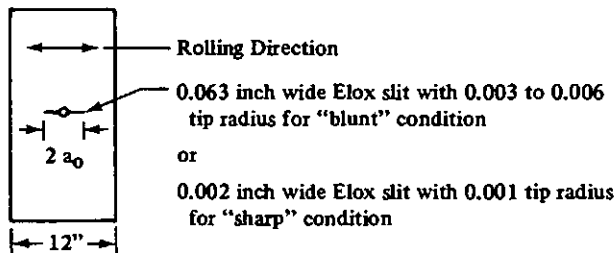


TABLE 3.02722. EFFECT OF SHEET THICKNESS ON FRACTURE-TOUGHNESS PARAMETER K_c (68)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

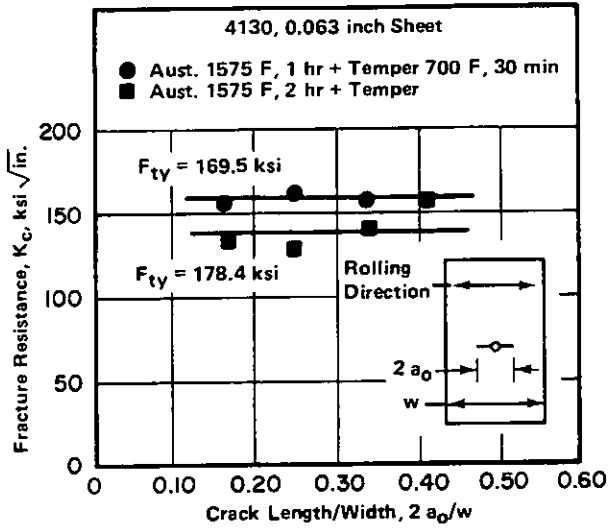


FIGURE 3.02723. EFFECT OF YIELD STRENGTH ON FRACTURE RESISTANCE OF SHEET (71)

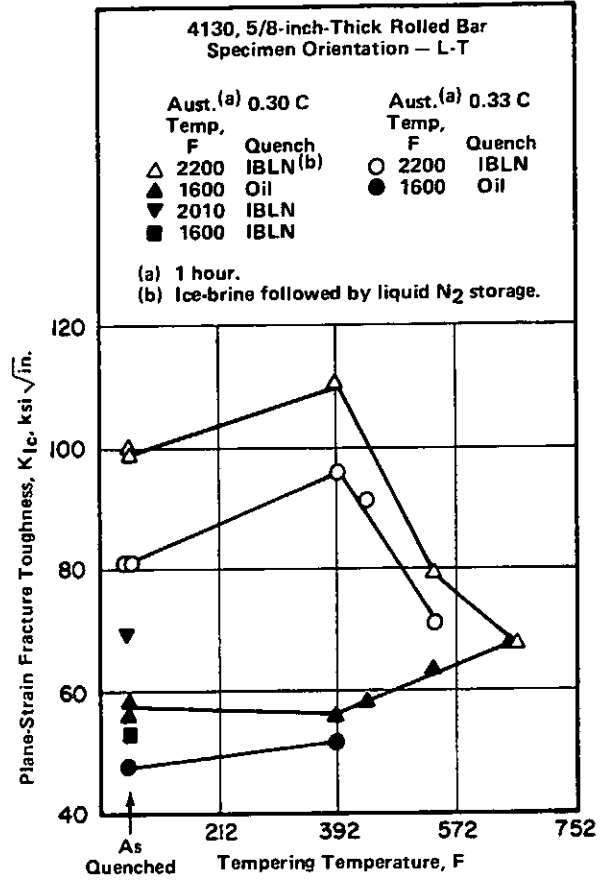


FIGURE 3.02724. EFFECT OF TEMPERING TEMPERATURE, CARBON CONTENT, AND QUENCHING MEDIUM ON PLANE-STRAIN FRACTURE TOUGHNESS OF FLAT ROLLED BAR (51)

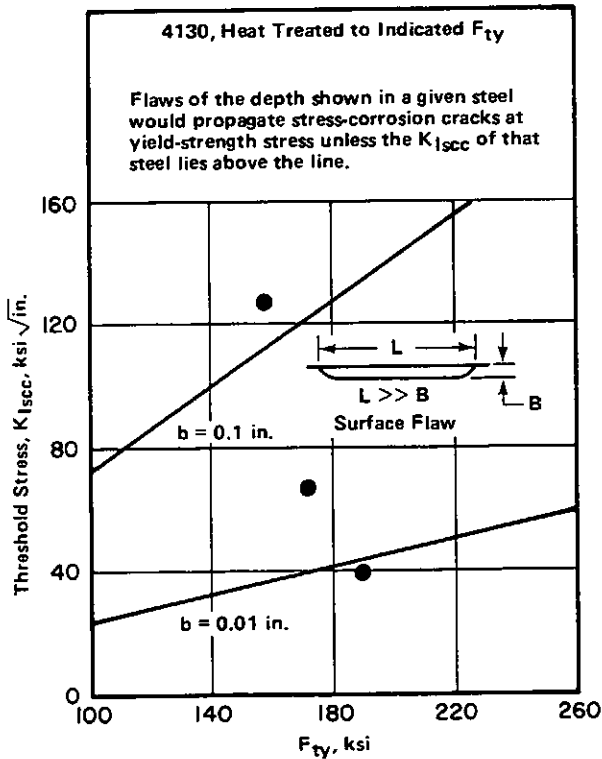


FIGURE 3.02725. EFFECT OF YIELD STRENGTH AND FLAW DEPTH ON THRESHOLD STRESS INTENSITY IN STATIC LOADING (58)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

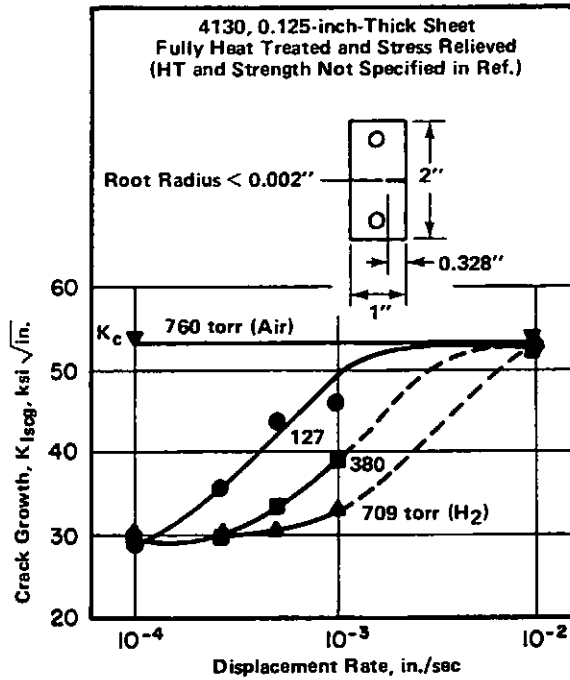


FIGURE 3.02726. EFFECT OF HYDROGEN PRESSURE AND DISPLACEMENT RATE ON INITIATION OF SLOW CRACK GROWTH, K_{Iscg} (EMBRITTLEMENT) (66)

Alloy	4130	
Form	>1 inch Plate	
Condition	1600 F, 2 hr, OQ + 1175 F, 2 hr + AC	
Air, $K_Q^{(a)}$, ksi $\sqrt{\text{in.}}$	Hydrogen Pressure ^(b) , psi	Arrest Stress Intensity, ksi $\sqrt{\text{in.}}$
114	3,000	80
	6,000	62
	9,000	41
	10,000	29
	14,000	47

(a) 10 percent secent.
(b) Temperature = 55 F.

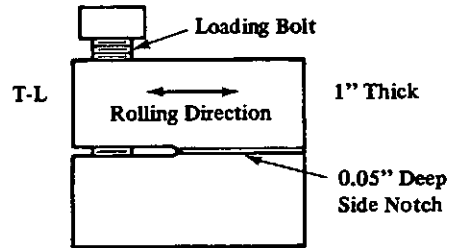
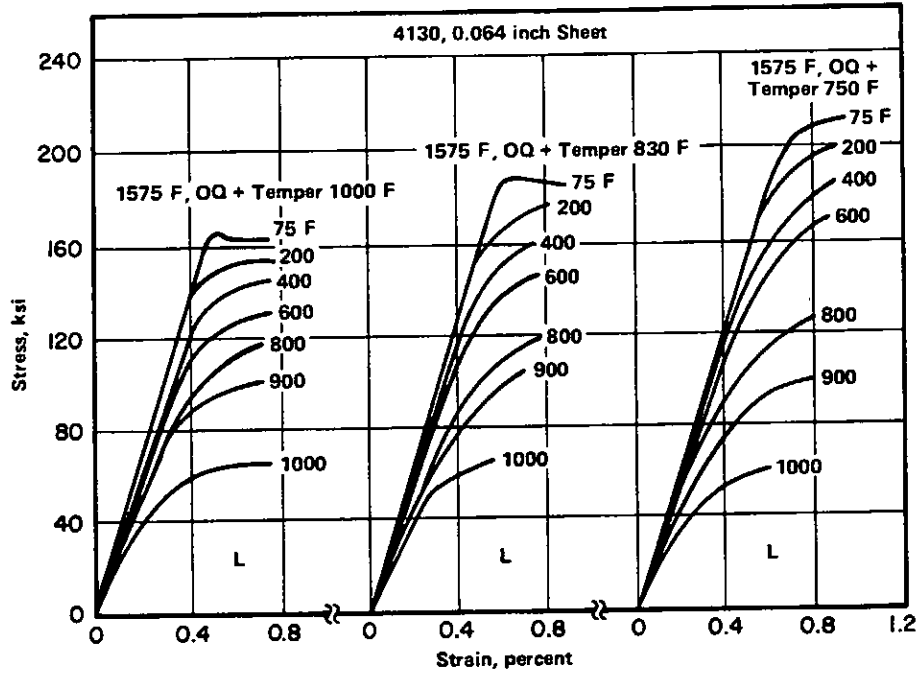


TABLE 3.02727. EFFECT OF HYDROGEN PRESSURE ON THE ARREST OF CRACK PROPAGATION (69)



Fe
0.3 C
0.95 Cr
0.2 Mo

4130

FIGURE 3.0311. TENSILE STRESS-STRAIN CURVES FOR SHEET AT TEMPERATURES UP TO 1000 F AFTER THREE DIFFERENT TEMPERING TREATMENTS: CURVES APPLY TO SPECIMENS HELD FOR 0.5 UP TO 100 HOURS AT TEST TEMPERATURE (14)

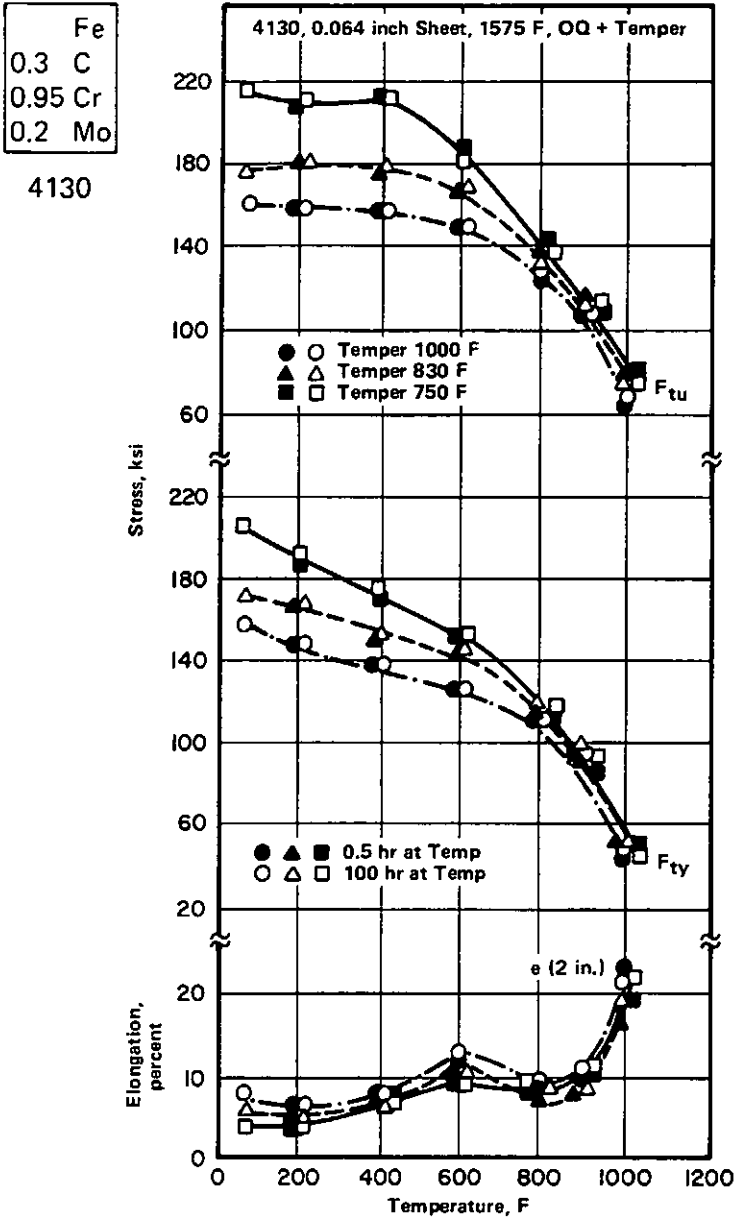


FIGURE 3.0312. EFFECT OF TEMPERATURE ON TENSILE PROPERTIES OF SHEET AFTER THREE DIFFERENT TEMPERING TREATMENTS AND TWO DIFFERENT EXPOSURE TIMES (14)

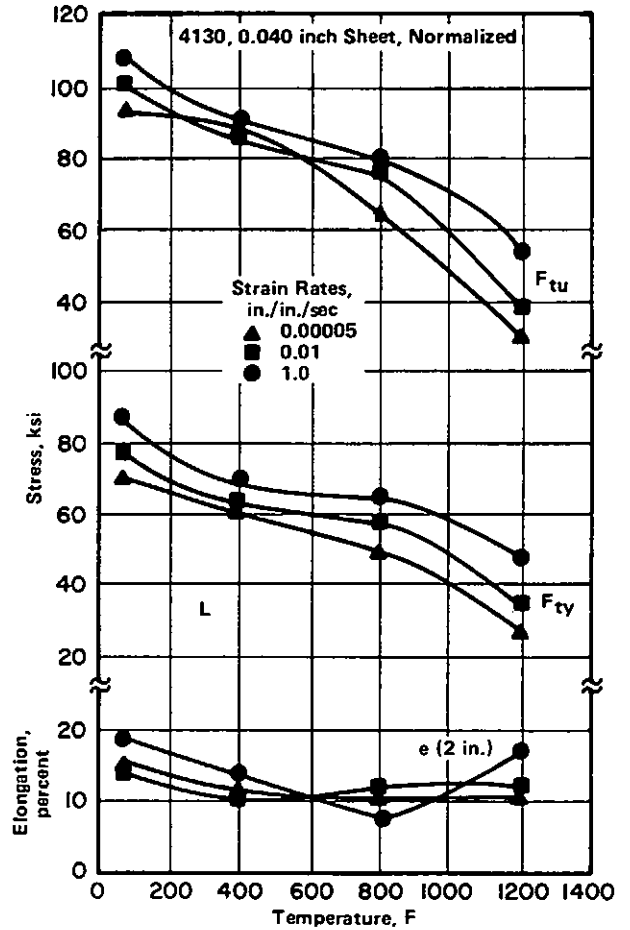


FIGURE 3.0313. EFFECT OF TEMPERATURE AND STRAIN RATE ON TENSILE PROPERTIES OF NORMALIZED SHEET AFTER RAPID HEATING; DATA PERTAIN TO SPECIMENS HELD FOR 10 SECONDS UP TO 1/2 HOUR AT TEST TEMPERATURES (15)

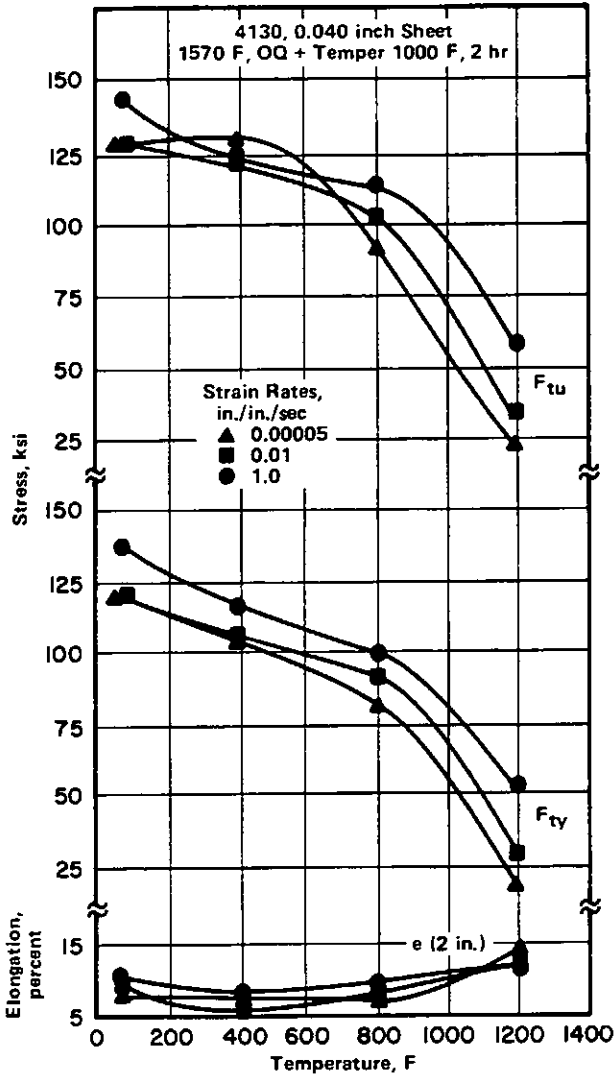


FIGURE 3.0314. EFFECT OF TEMPERATURE AND STRAIN RATE ON TENSILE PROPERTIES OF HEAT TREATED SHEET AFTER RAPID HEATING; DATA PERTAIN TO SPECIMENS HELD FOR 10 SECONDS UP TO 1/2 HOUR AT TEST TEMPERATURE (15)

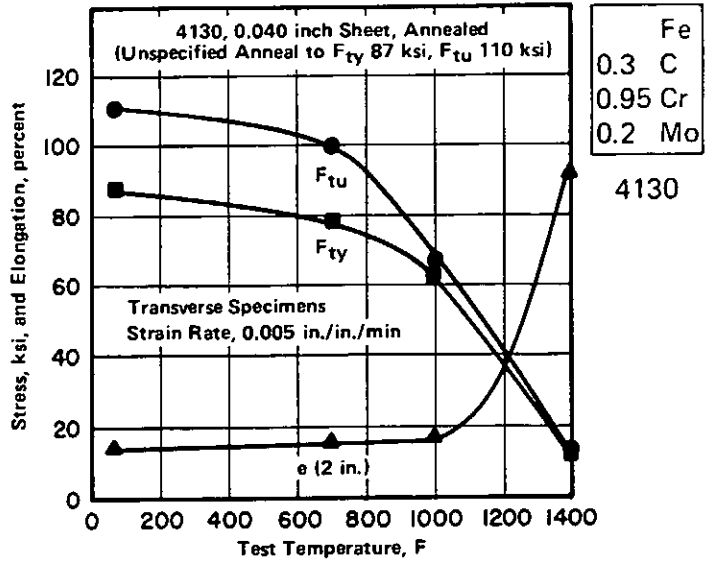


FIGURE 3.0315. EFFECT OF TEMPERATURE ON TENSILE PROPERTIES OF SHEET AFTER 5-MINUTE SOAK AT TEMPERATURE (53)

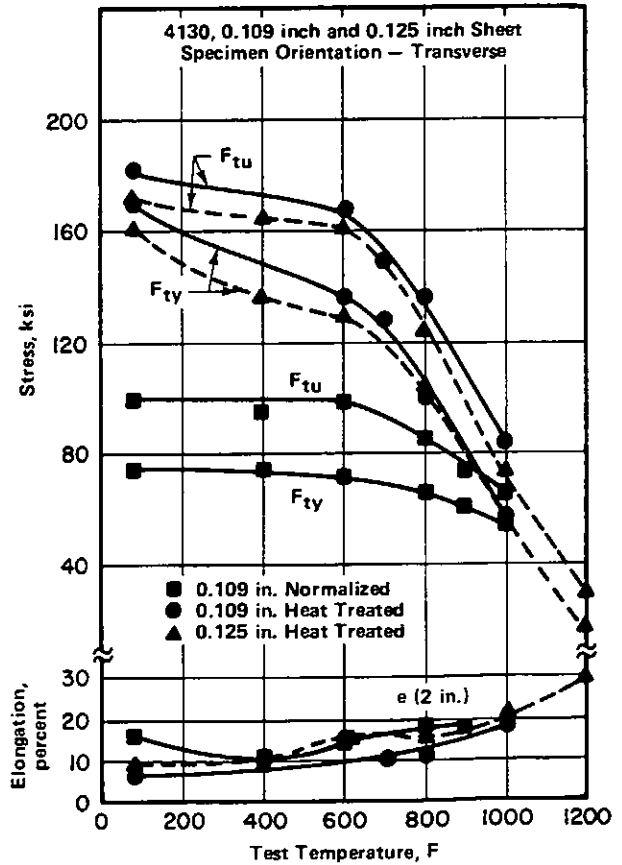


FIGURE 3.0316. EFFECT OF SHORT-TIME EXPOSURE TO ELEVATED TEMPERATURE ON TENSILE PROPERTIES OF SHEET (55)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

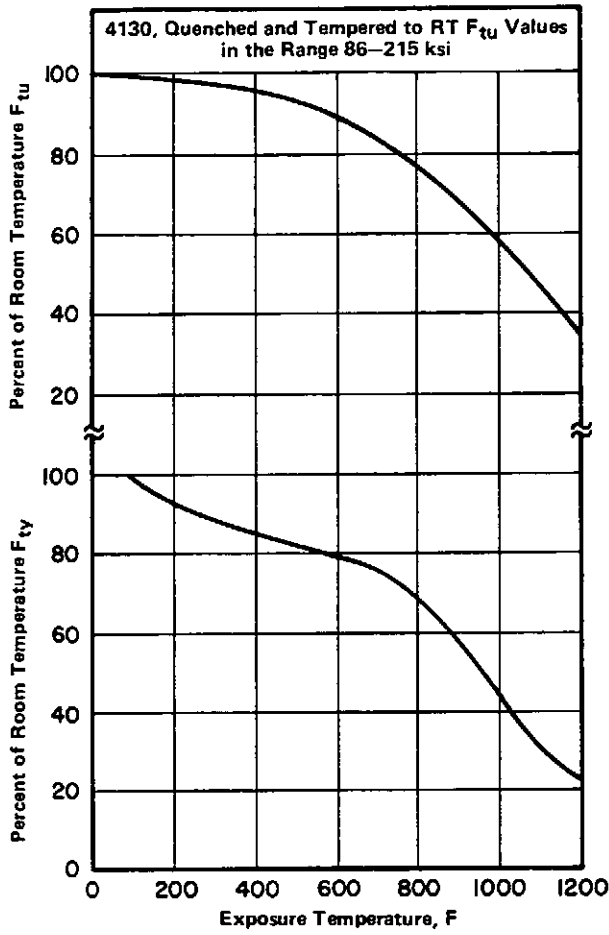


FIGURE 3.0317. EFFECT OF PRIOR EXPOSURE TO TEST TEMPERATURE UP TO 10 HOURS ON TENSILE PROPERTIES (63)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

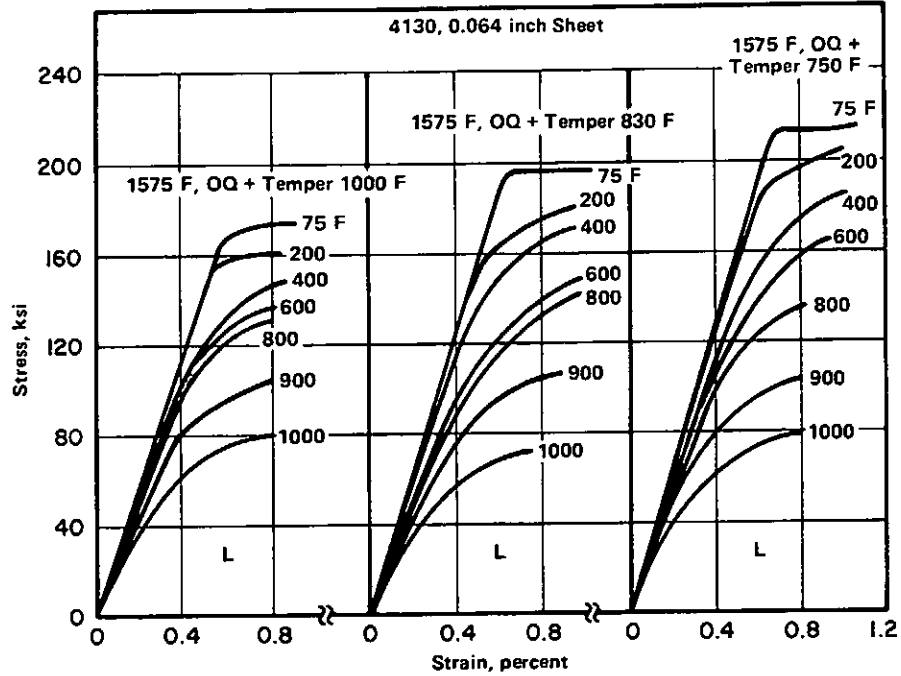


FIGURE 3.0321. COMPRESSIVE STRESS-STRAIN CURVES FOR SHEET AT TEMPERATURES UP TO 1000 F AFTER THREE DIFFERENT TEMPERING TREATMENTS; CURVES APPLY TO SPECIMENS HELD FOR 0.5 UP TO 100 HOURS AT TEST TEMPERATURES (14)

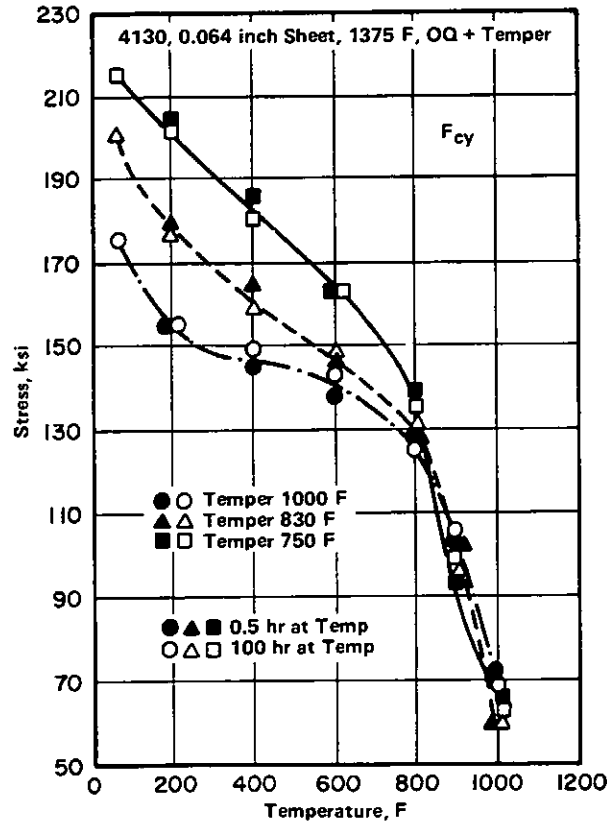


FIGURE 3.0322. EFFECT OF TEMPERATURE ON COMPRESSIVE YIELD STRENGTH OF SHEET AFTER THREE DIFFERENT TEMPERING TREATMENTS AND TWO DIFFERENT EXPOSURE TIMES (14)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

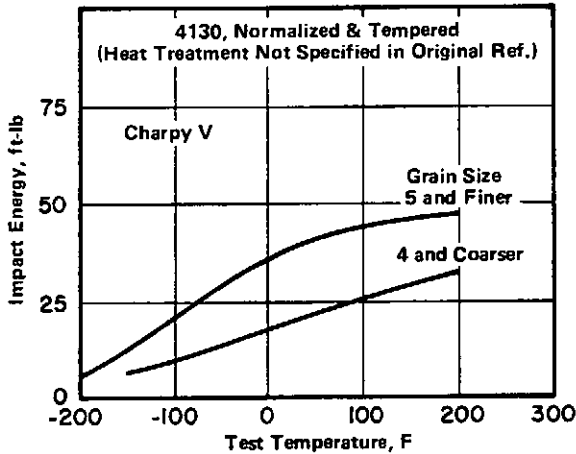


FIGURE 3.0331. EFFECT OF GRAIN SIZE AND TEMPERATURE ON IMPACT PROPERTIES (33)

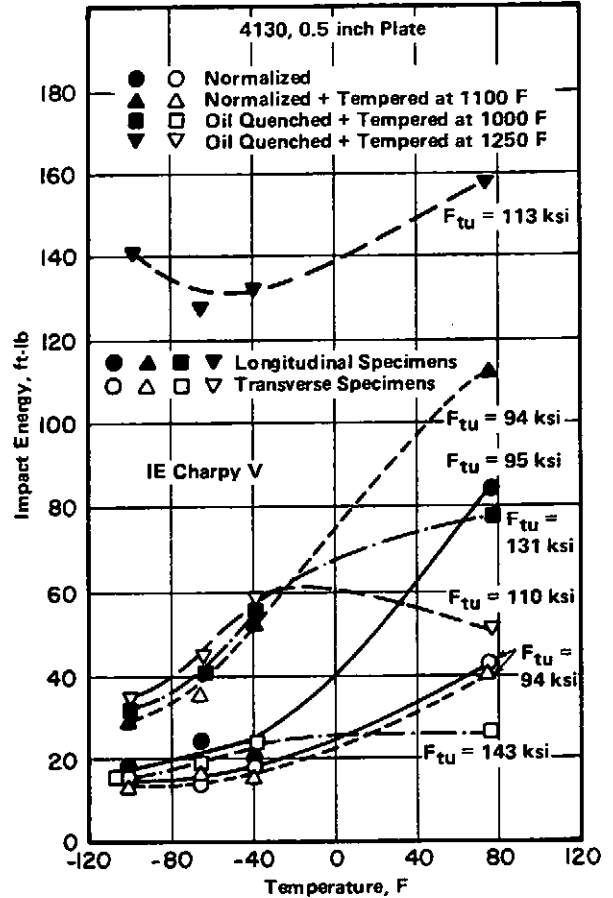


FIGURE 3.0332. EFFECT OF LOW TEMPERATURES ON IMPACT PROPERTIES IN SEVERAL HEAT-TREATED CONDITIONS (26)

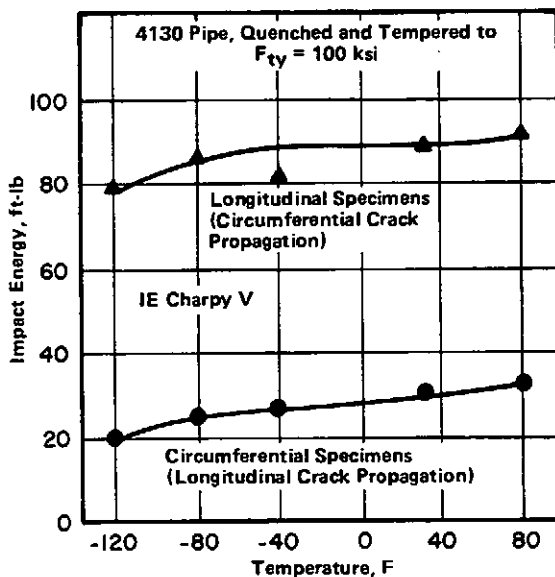


FIGURE 3.0333. EFFECT OF LOW TEMPERATURES ON IMPACT PROPERTIES OF PIPE (42)

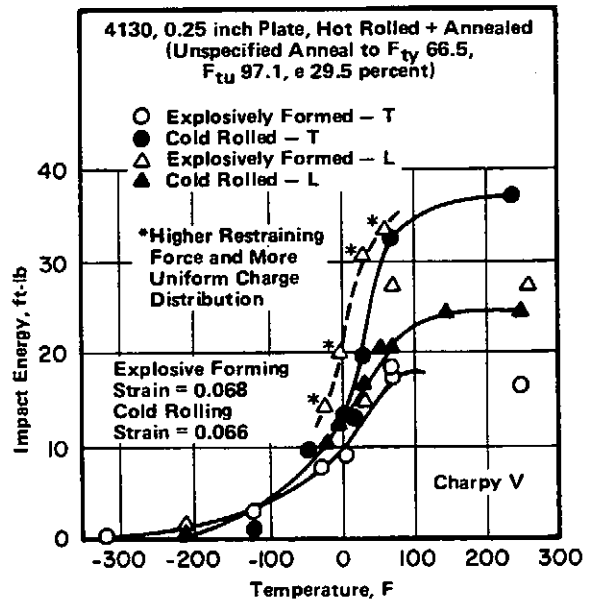


FIGURE 3.0334. EFFECT OF TEMPERATURE ON IMPACT PROPERTIES OF EXPLOSIVELY FORMED AND ROLLED AND ANNEALED PLATE (54)

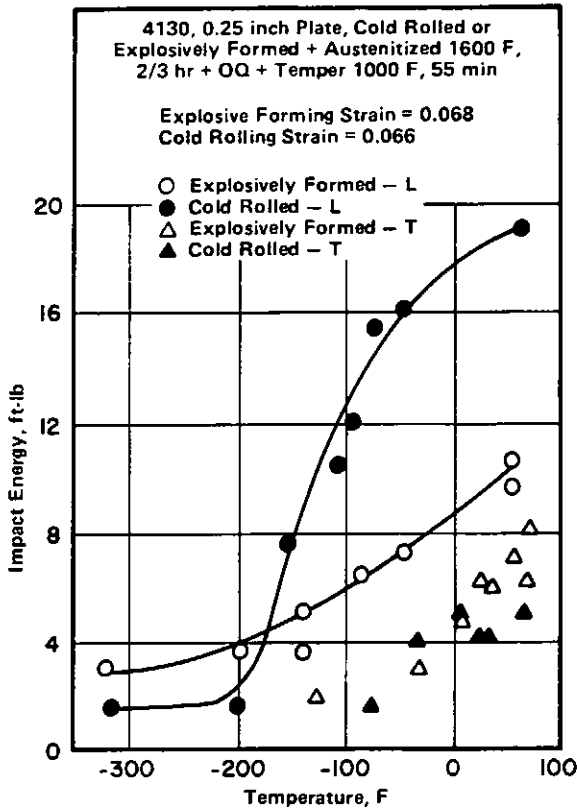


FIGURE 3.0335. EFFECT OF TEMPERATURE ON IMPACT PROPERTIES OF EXPLOSIVELY FORMED AND COLD-ROLLED HEAT-TREATED PLATE (54)

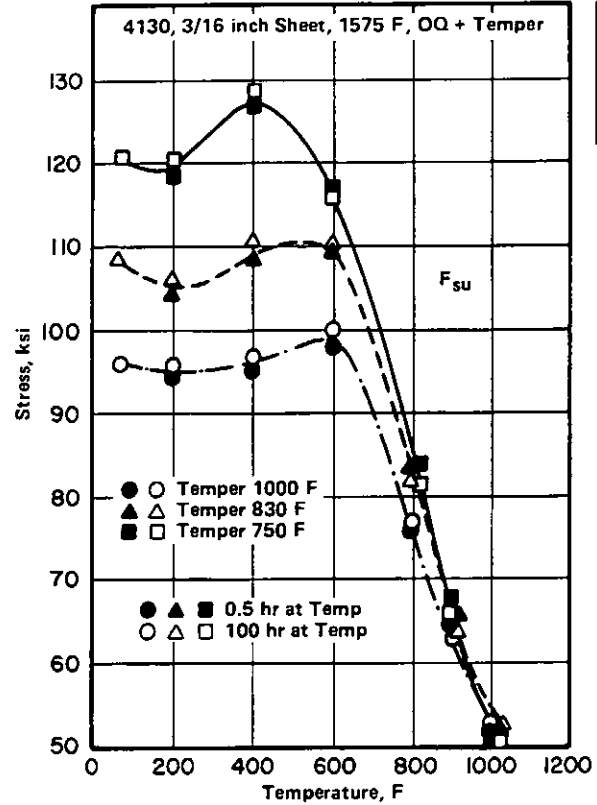


FIGURE 3.0351. EFFECT OF TEMPERATURE ON SHEAR STRENGTH OF SHEET AFTER THREE DIFFERENT TEMPERING TREATMENTS AND TWO DIFFERENT EXPOSURE TIMES (14)

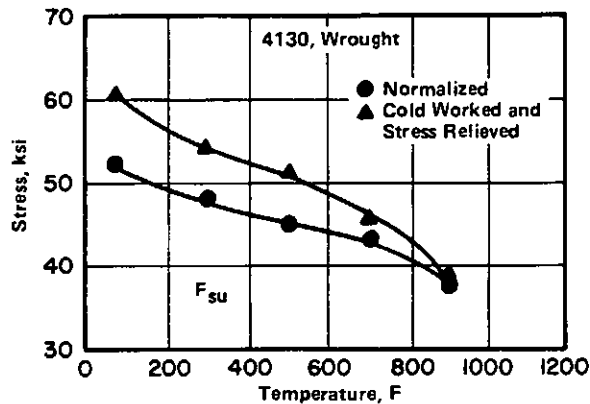


FIGURE 3.0352. EFFECT OF TEMPERATURE ON SHEAR STRENGTH OF NORMALIZED AND OF COLD-WORKED-AND-STRESS-RELIEVED TYPE 4130 (27)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

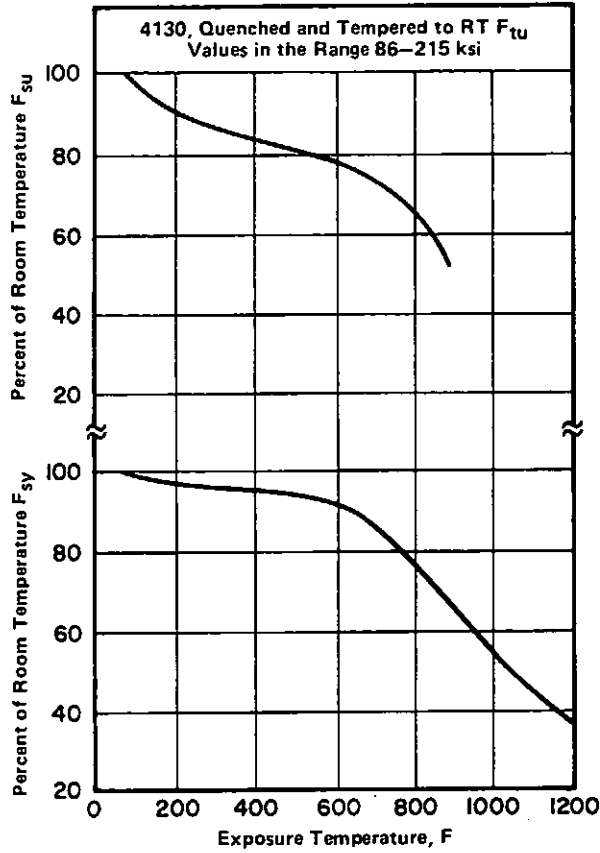


FIGURE 3.0353. EFFECT OF PRIOR TEST TEMPERATURE EXPOSURE UP TO 10 HOURS ON SHEAR PROPERTIES (63)

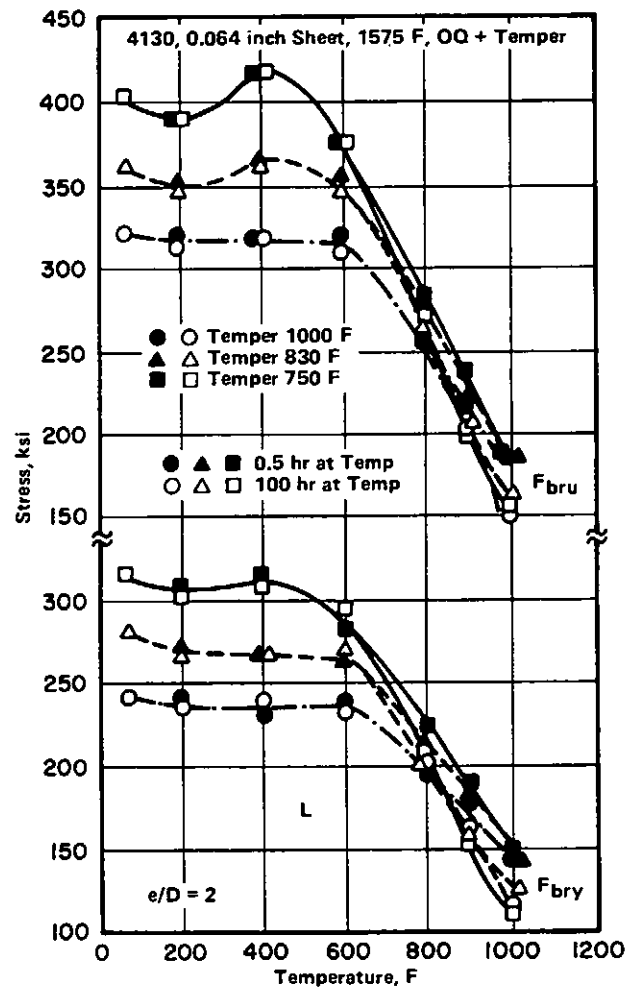


FIGURE 3.0361. EFFECT OF TEMPERATURE ON BEARING STRENGTH ($e/D = 2$) OF SHEET AFTER THREE DIFFERENT TEMPERING TREATMENTS AND TWO DIFFERENT EXPOSURE TIMES (14)

Fe
0.3 C
0.95 Cr
0.2 Mo
4130

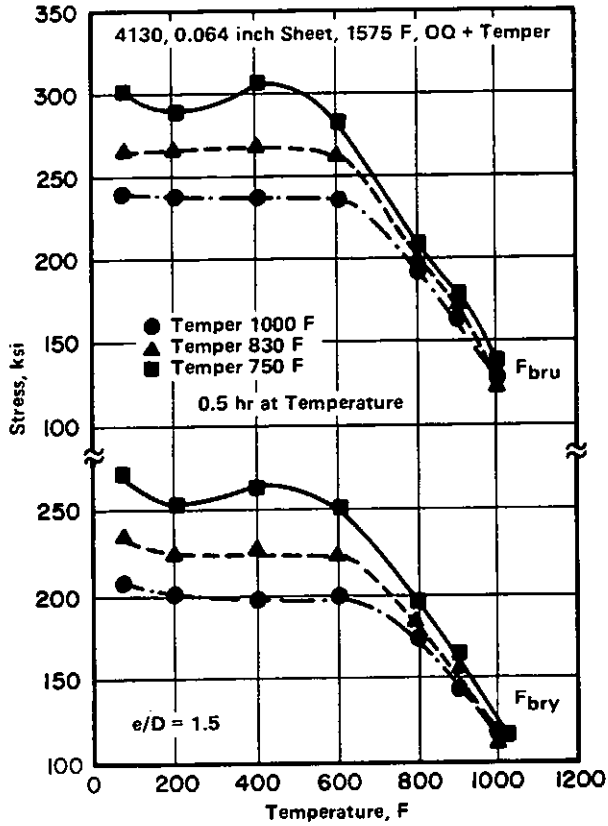


FIGURE 3.0362. EFFECT OF TEMPERATURE ON BEARING STRENGTH ($e/D = 1.5$) OF SHEET AFTER THREE DIFFERENT TEMPERING TREATMENTS (14)

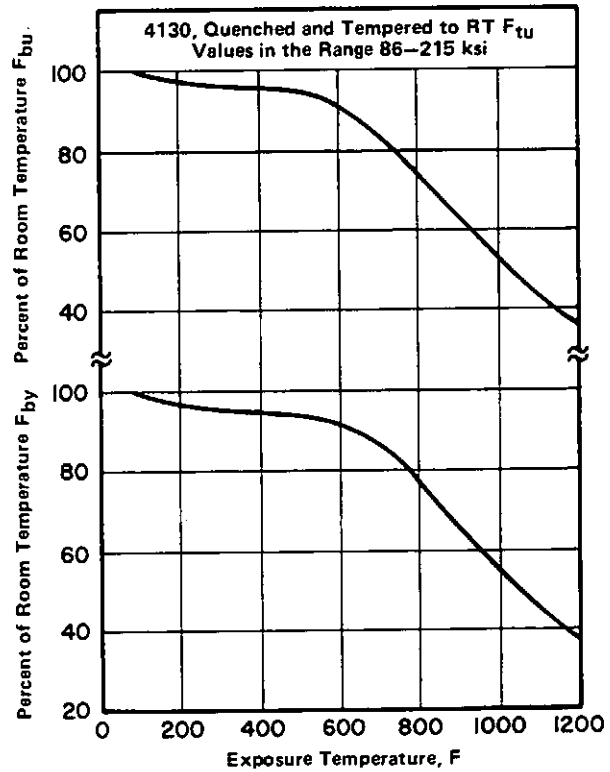


FIGURE 3.0363. EFFECT OF PRIOR TEST TEMPERATURE EXPOSURE UP TO 10 HOURS ON BEARING PROPERTIES (63)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

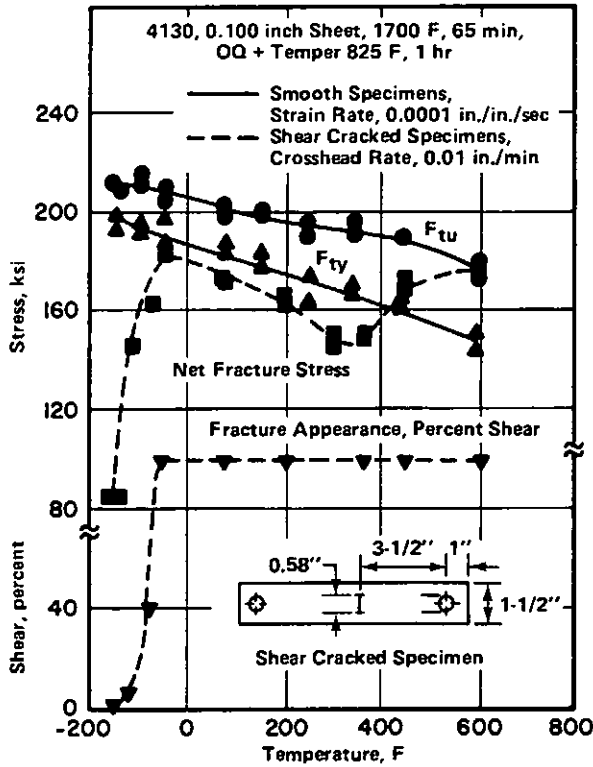


FIGURE 3.03711. COMPARISON OF TENSILE ULTIMATE (F_{tu}) AND YIELD (F_{ty}) STRENGTHS WITH CRACK STRENGTH (NET FRACTURE STRESS) AND FRACTURE APPEARANCE OF SHEAR-CRACKED SHEET SPECIMENS QUENCHED AND TEMPERED AT 825 F (38)

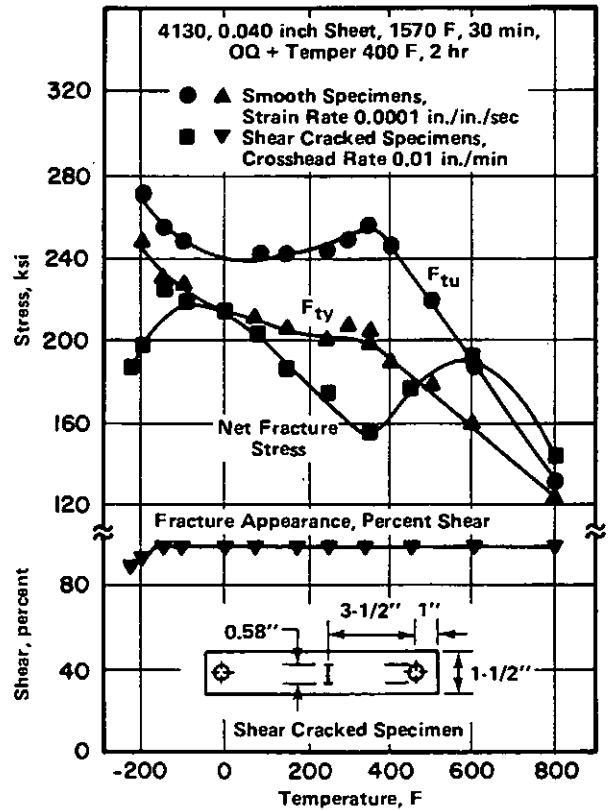


FIGURE 3.03712. COMPARISON OF TENSILE ULTIMATE (F_{tu}) AND YIELD (F_{ty}) STRENGTHS WITH CRACK STRENGTH (NET FRACTURE STRESS) AND FRACTURE APPEARANCE OF SHEAR-CRACKED SHEET SPECIMENS QUENCHED AND TEMPERED AT 400 F (38)

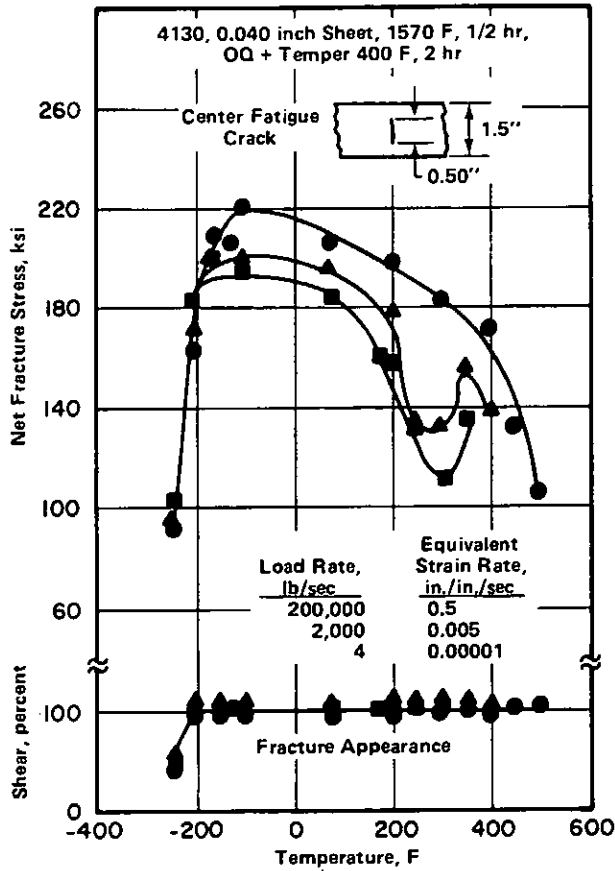


FIGURE 3.03713. EFFECT OF TEMPERATURE AND LOADING RATE ON CRACK STRENGTH (NET FRACTURE STRESS) AND FRACTURE APPEARANCE OF FATIGUE-CRACKED SHEET SPECIMENS QUENCHED AND TEMPERED AT 400 F (39)

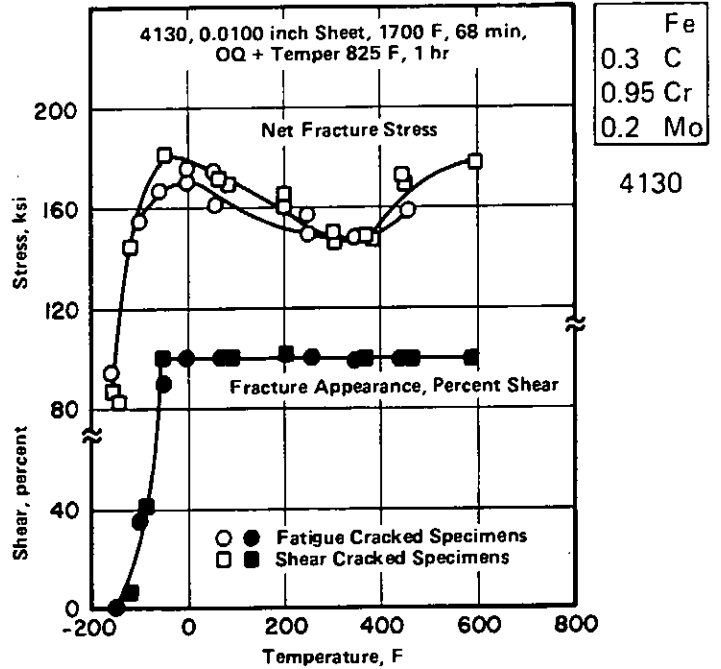


FIGURE 3.03714. EFFECT OF TEMPERATURE ON CRACK STRENGTH (NET FRACTURE STRESS) AND FRACTURE APPEARANCE OF BOTH FATIGUE-CRACKED AND SHEAR-CRACKED SHEET SPECIMENS QUENCHED AND TEMPERED AT 825 F (38)

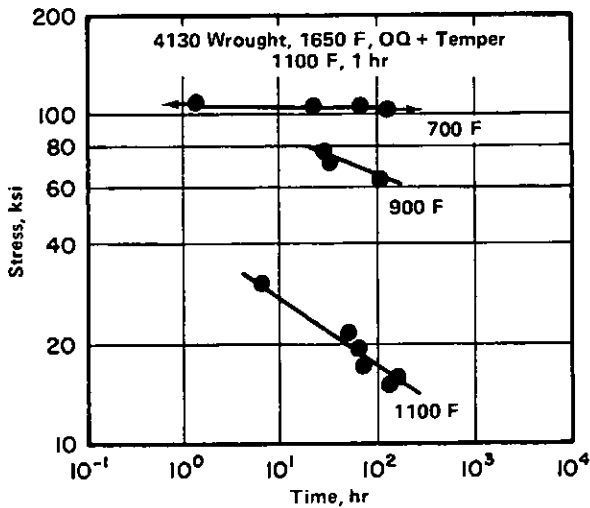


FIGURE 3.041. CREEP-RUPTURE CURVES FROM 700 TO 1100 F (40)

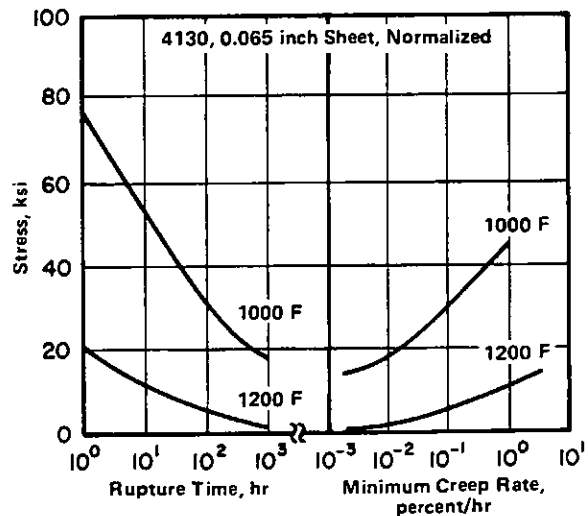


FIGURE 3.042. CREEP-RUPTURE AND MINIMUM-CREEP-RATE CHARACTERISTICS OF NORMALIZED SHEET AT 1000 AND 1200 F (18)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

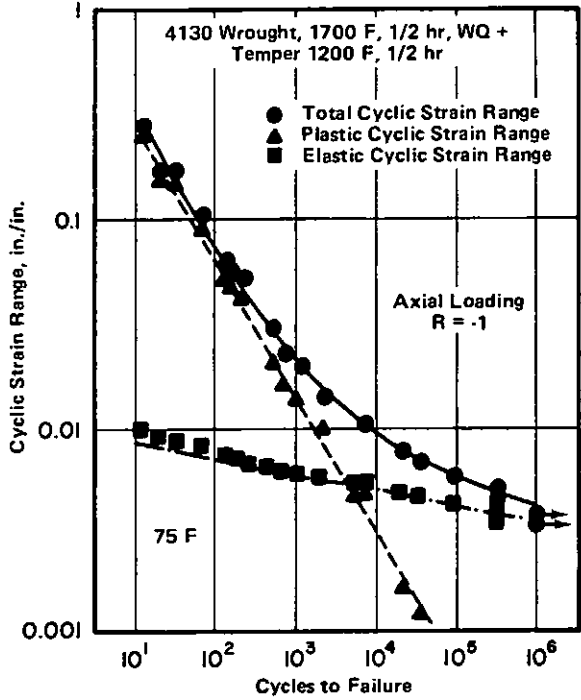


FIGURE 3.051. FATIGUE LIFE AS A FUNCTION OF CYCLIC STRAIN FOR TYPE 4130 TEMPERED AT 1200 F (22)

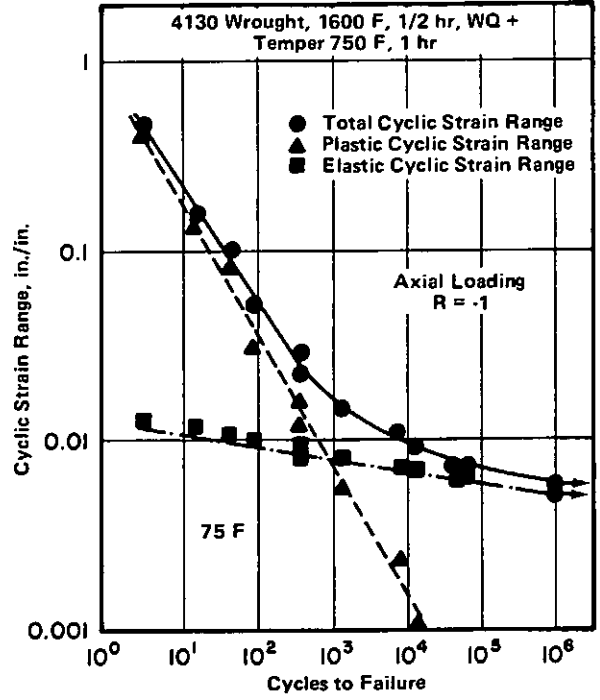


FIGURE 3.052. FATIGUE LIFE AS A FUNCTION OF CYCLIC STRAIN FOR TYPE 4130 TEMPERED AT 750 F (22)

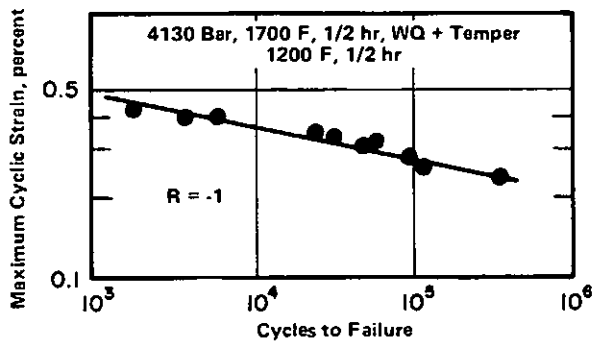


FIGURE 3.053. EFFECT OF MAXIMUM CYCLIC TENSILE STRAIN ON FATIGUE LIFE IN REVERSED BENDING (24)

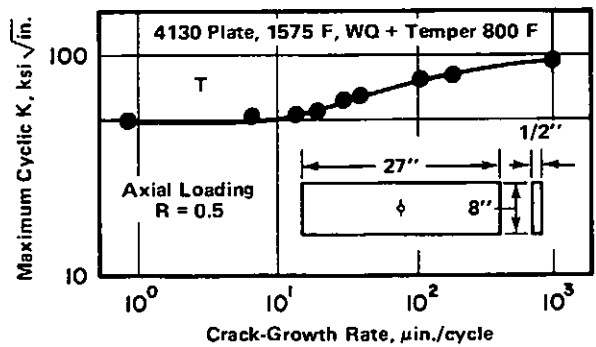


FIGURE 3.054. FATIGUE-CRACK-GROWTH RATE AS A FUNCTION OF MAXIMUM CYCLIC STRESS INTENSITY FACTOR (K) FOR CENTER-CRACKED SPECIMENS (23)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

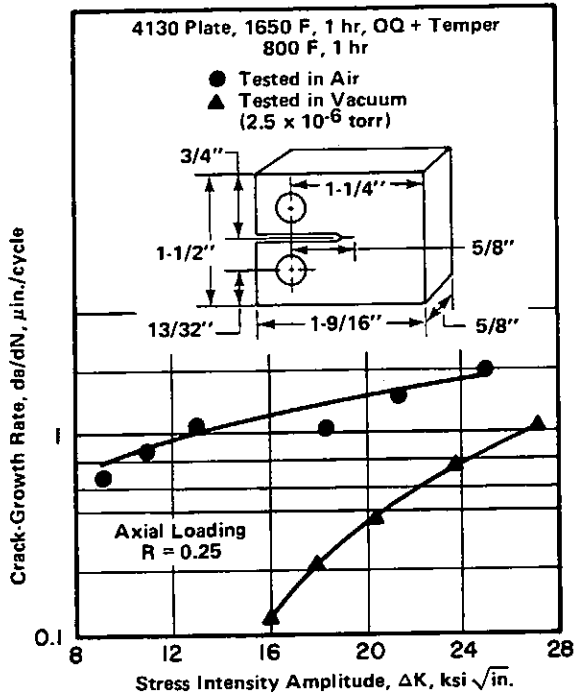


FIGURE 3.055. FATIGUE-CRACK-GROWTH RATE IN AIR AND VACUUM AS A FUNCTION OF STRESS INTENSITY AMPLITUDE (ΔK) FOR COMPACT TENSION SPECIMEN (25)

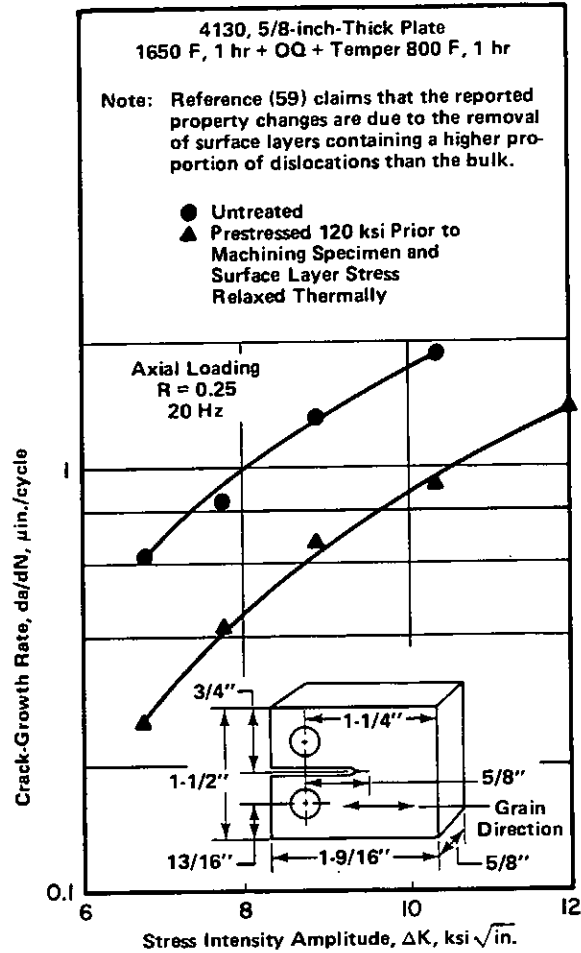


FIGURE 3.056. EFFECT OF PRESTRESSING AND REMOVAL OF SURFACE-LAYER STRESS BY THERMAL RELAXATION ON FATIGUE CRACK GROWTH RATE OF PLATE (59)

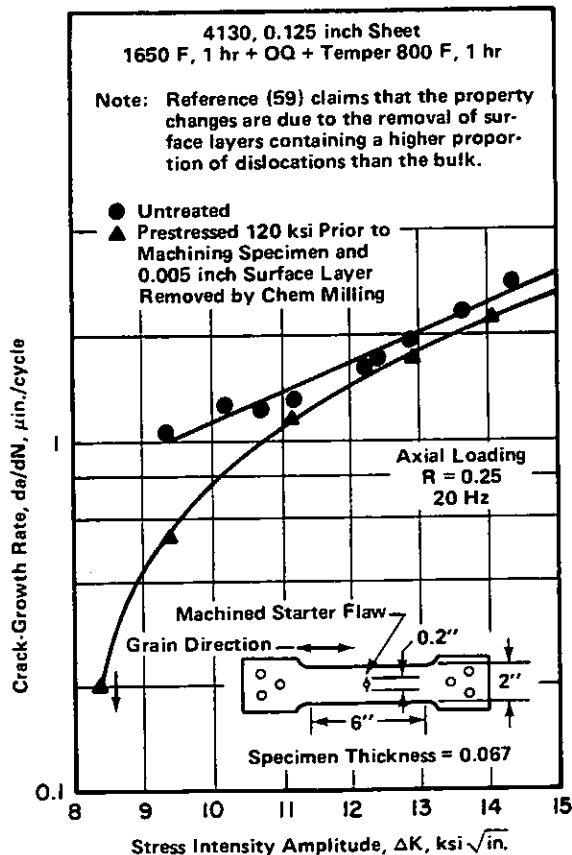


FIGURE 3.057. EFFECT OF PRESTRESSING AND CHEM-MILLING ON FATIGUE CRACK GROWTH RATE OF SHEET (59)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

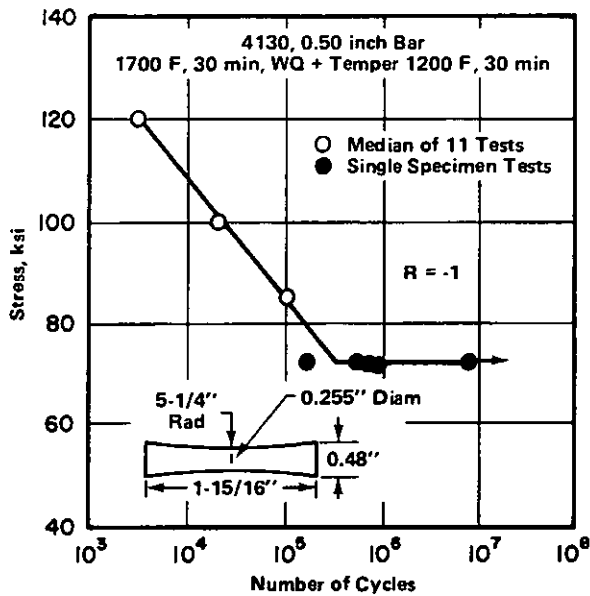


FIGURE 3.058. S-N CURVE AT ROOM TEMPERATURE IN ROTATING BENDING (41)

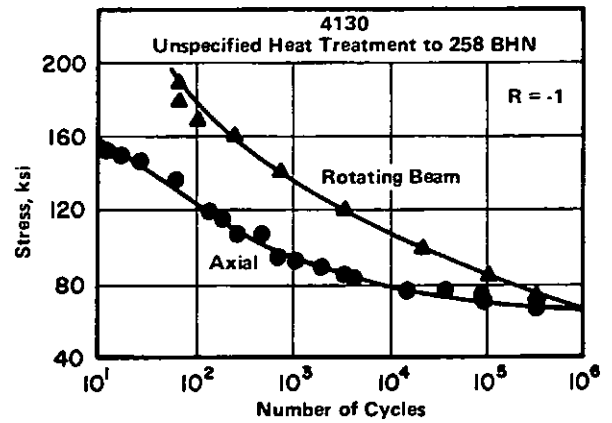


FIGURE 3.059. COMPARISON OF AXIAL-LOADING AND ROTATING-BEAM-BENDING FATIGUE DATA (65)

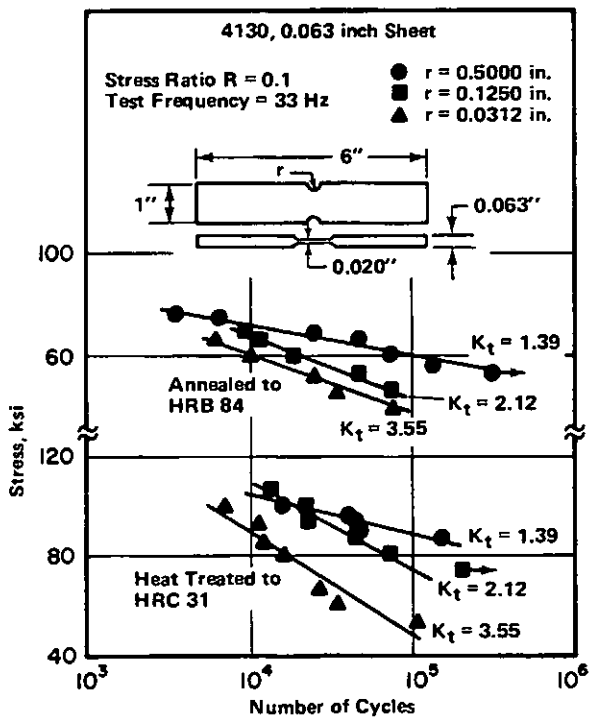


FIGURE 3.0510. AXIAL-STRESS S-N CURVES FOR EDGE-NOTCHED SPECIMENS (65)

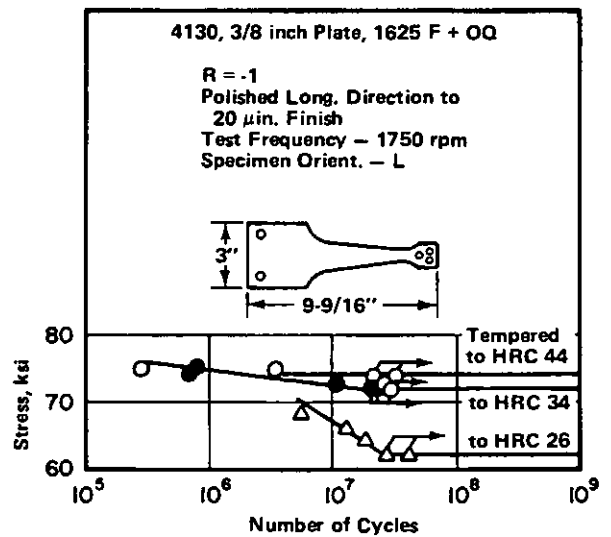


FIGURE 3.0511. EFFECT OF TEMPER TREATMENT (MEASURED BY ROCKWELL HARDNESS) ON REPEATED-BENDING FATIGUE PROPERTIES (49)

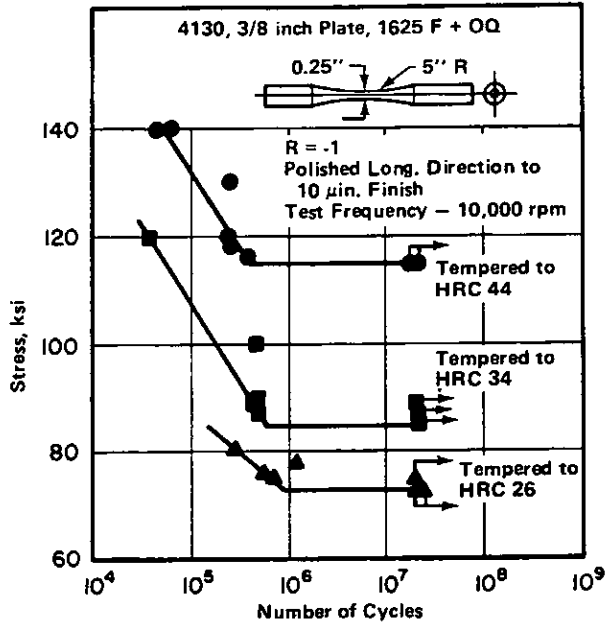


FIGURE 3.0512. EFFECT OF TEMPER TREATMENT (MEASURED BY ROCKWELL HARDNESS) ON ROTATING-BEAM FATIGUE PROPERTIES (49)

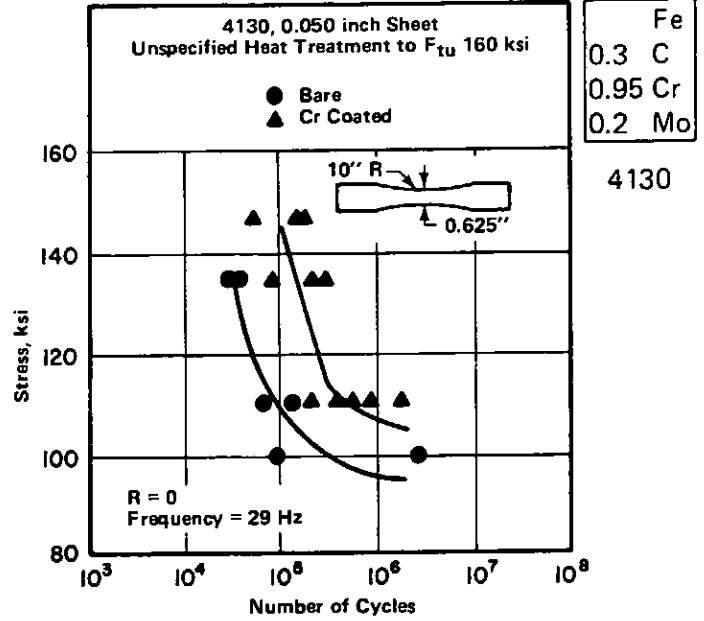


FIGURE 3.0513. EFFECT OF CHROMIUM COATING ON AXIAL-STRESS FATIGUE PROPERTIES OF SHEET (57)

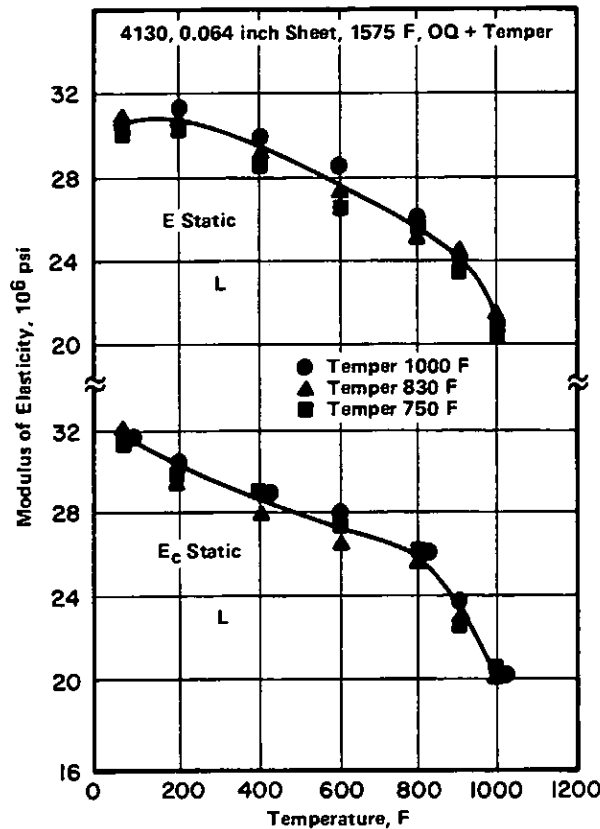


FIGURE 3.0621. EFFECT OF TEMPERATURE ON MODULUS OF ELASTICITY IN TENSION AND COMPRESSION OF SHEET AFTER THREE DIFFERENT TEMPERING TREATMENTS (14)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

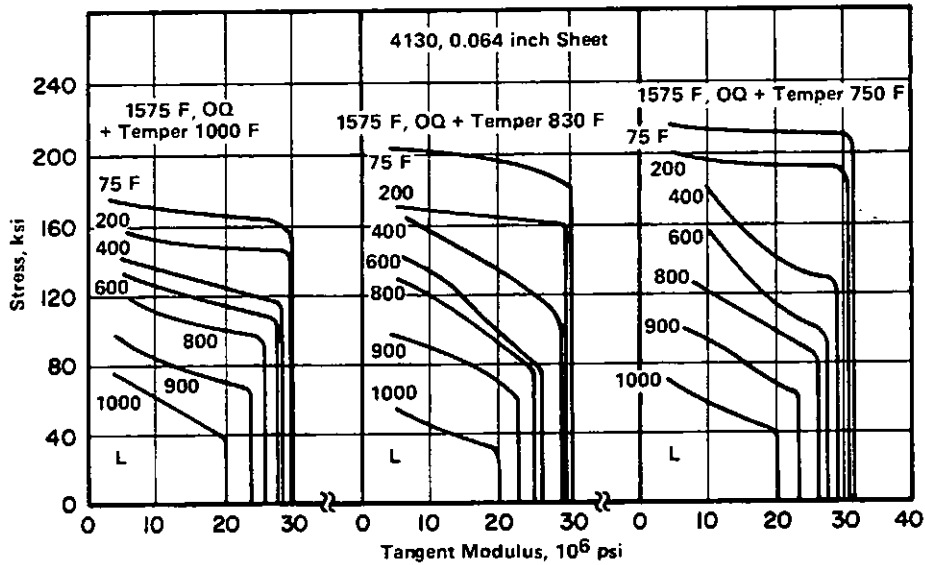


FIGURE 3.0622. COMPRESSIVE TANGENT MODULUS CURVES FOR SHEET AFTER THREE DIFFERENT TEMPERING TREATMENTS (14)

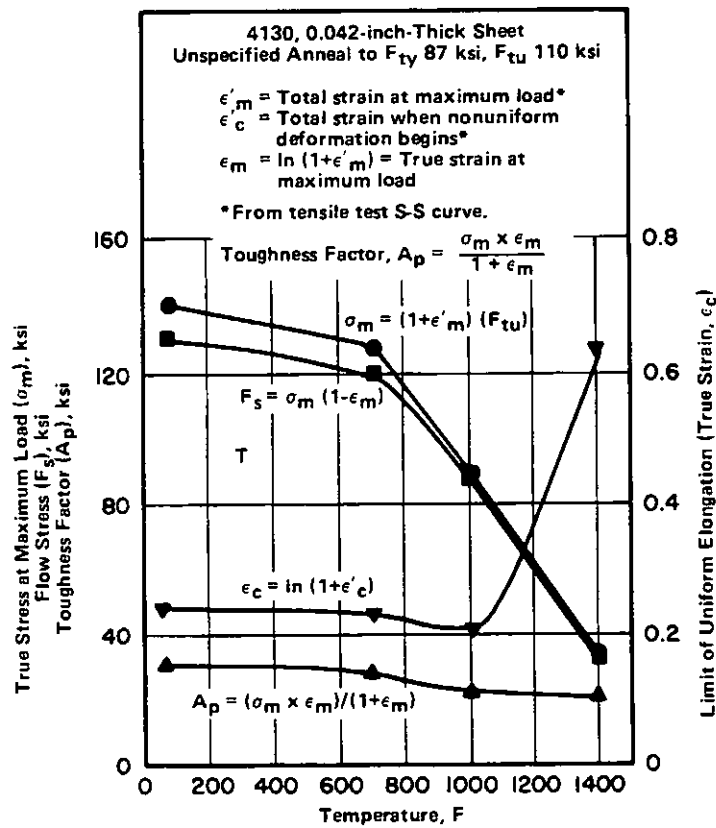


FIGURE 4.013. EFFECT OF TEMPERATURE ON FORMABILITY OF SHEET (53)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Alloy	4130			
Form	0.50-inch-Thick Plate			
Condition	OQ + 1250 F + Weld With 8016N Coated Stick Electrodes (2.5 percent Ni)			
Specimens	Orientation	F _{ty} , ksi	F _{tu} , ksi	e (2 in.), percent
Parent metal	L	97.5	113	24.8
	T	94.5	110	20.2
As welded	L	79.0	105	20.2
	T	81.0	106	21.2

TABLE 4.032. TENSILE PROPERTIES OF PARENT METAL AND OF 0.5 INCH THICK PLATE WITH BUTT WELD AT THE CENTER OF THE SPECIMEN GAGE LENGTH (26)

Alloy	4130	
Form	0.50-inch-Thick Plate	
Condition	OQ + 1250 F + Weld With 8016Q Coated Stick Electrodes (2.5 percent Ni)	
Test Temp, F	-65	
Notch Position	Charpy V, ft-lb	
In weld metal	22	
In heat-affected zone	34	
In outer edge of heat-affected zone	31	
In parent metal (longitudinal)	130	
In parent metal (transverse)	44	

TABLE 4.033. CHARPY-V NOTCH IMPACT PROPERTIES AT -65 F OF BUTT-WELDED 0.50-INCH-THICK PLATE (26)

Alloy	4130					
Form	0.090 inch Sheet					
Condition	Annealed or As Welded			1575 F, 1 hr, OQ + 900 F, 1 hr		
	F _{ty} , ksi	F _{tu} , ksi	e (2 in.), percent	F _{ty} , ksi	F _{tu} , ksi	e (2 in.), percent
Base Metal	43.6	68.6	30.4	151.6	167.5	8.9
GTA ^(a)	46.3	67.1	21.8	150.4	163.5	8.6
PGTA ^(b)	17.3	66.5	19.6	153.8	165.7	8.6
GMA ^(c)	45.4	66.4	18.2	151.9	165.5	7.8
PGMA ^(d)	48.1	69.2	16.5	153.9	164.1	7.2

- (a) Gas tungsten arc welded.
- (b) Pulsed gas tungsten arc welded.
- (c) Gas metal arc welded.
- (d) Pulsed gas metal arc welded.

TABLE 4.034. TENSILE PROPERTIES OF SHEET BUTT WELDED BY VARIOUS ARC PROCESSES (43)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

Alloy	4130		
Form	0.125 inch Sheet		
	F _{ty} , ksi	F _{tu} , ksi	e (2 in.), percent
Annealed	48.9	68.4	30.7
GTA Welded	47.8	67.0	22.9

TABLE 4.035. TENSILE PROPERTIES OF BUTT-WELDED SHEET (60)

Alloy	4130			
Form	Sheet and Plate Butt Welds – Beads Off			
Type Electrode	Weld Heat Treatment (to F _{tu} Listed)	Thickness, in.	Duplicate Tests	
			F _{tu} , ksi	Failure Location ^(a)
Arcaloy 502	150 ksi	0.185	158	H B
		0.236	150	– –
		0.375	161	H W
		0.500	152	H H
	175 ksi	0.078	165	– –
		0.100	177	– –
		0.236	176	W B
		0.245	159	– –
		0.375	178	H H
		0.500	162	W W
P & H21	150 ksi	0.235	156	B B
		0.350	155	B B
		0.492	154	W H
	175 ksi	0.077	176	– –
		0.162	180	– –
		0.235	166	W W
		0.255	164	– –
0.235	173	W W		
0.492	166	W W		
AW-4	175 ksi	0.095	175	– –

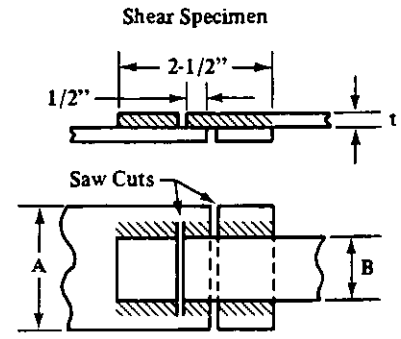
(a) H = heat-affected zone, W = weld metal, B = base metal.

TABLE 4.036. EFFECT OF ELECTRODE ALLOY AND SHEET OR PLATE THICKNESS ON TENSILE ULTIMATE STRENGTH OF BEAD-OFF BUTT WELDS (70)

Alloy	4130		
Form	Sheet and Plate Fillet Welds		
Type Electrode	Weld Heat Treatment (to F_{TU} Listed)	Thickness, in.	F_{su} , ksi
Arcaloy	150 ksi	0.076	93.3
		0.190	68.6
		0.248	64.5
		0.258	60.5
		0.385	62.5
	175 ksi	0.076	98.7
		0.190	74.3
		0.250	66.5
		0.260	62.9
		0.388	62.1
P & H21	150 ksi	0.076	98.1
		0.156	78.8
		0.246	77.5
		0.250	65.8
		0.380	55.5
	175 ksi	0.076	93.4
		0.156	83.4
		0.245	74.8
		0.250	57.1
		0.380	54.6
AW-4	150 ksi	0.076	100.8
		0.093	93.8
		0.123	87.0
		0.255	58.8
		0.258	60.8
	175 ksi	0.388	58.8
		0.500	49.2
		0.078	99.1
		0.093	94.3
		0.123	90.1
		0.255	58.6
		0.260	61.6
		0.386	53.5
		0.496	40.8

Fe
0.3 C
0.95 Cr
0.2 Mo

4130



t	A	B
1/4	2	1
3/8	2	1
1/2	3	1-1/2

TABLE 4.037. EFFECT OF ELECTRODE ALLOY AND SHEET OR PLATE THICKNESS ON SHEAR ULTIMATE STRENGTH OF LONGITUDINAL-WELD LAP JOINTS (70)

Fe
0.3 C
0.95 Cr
0.2 Mo

4130

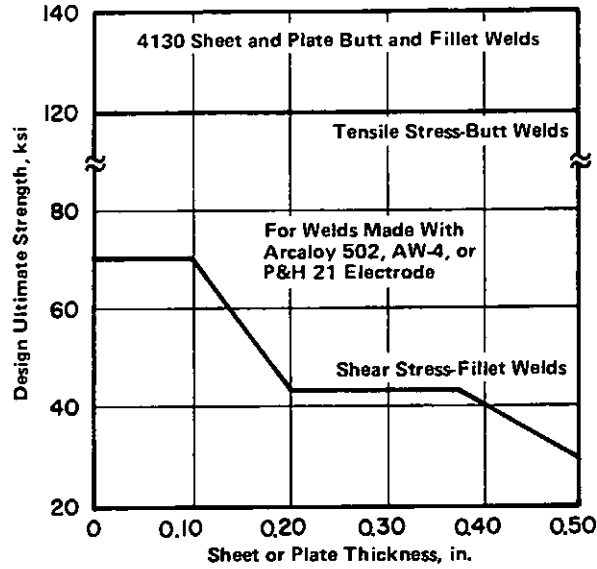


FIGURE 4.038. TENSILE AND SHEAR ULTIMATE-STRENGTH DESIGN CURVES FOR SHEET AND PLATE WELDS HEAT TREATED TO EITHER 150 OR 175 KSI (70)

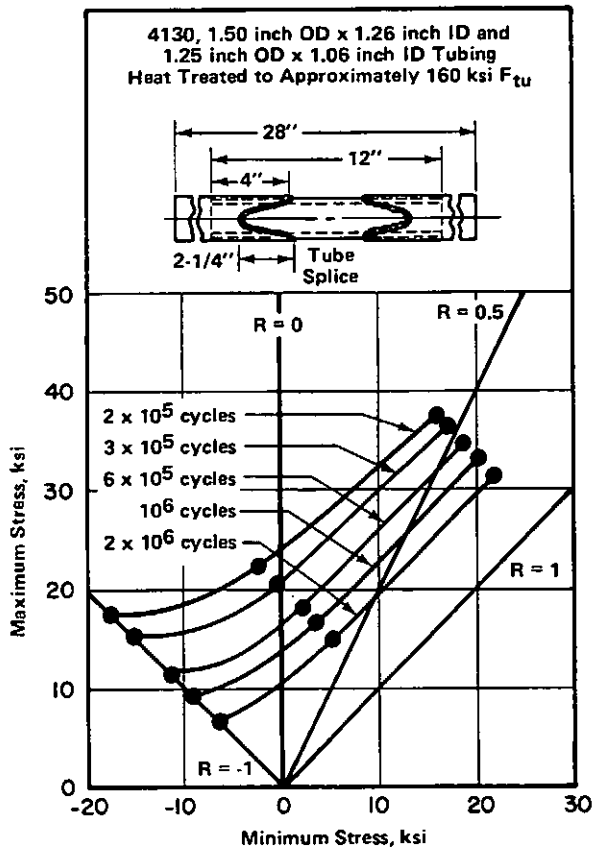


FIGURE 4.039. CONSTANT-LIFE DIAGRAM FOR AN AXIAL-LOADED TUBE SPLICE - STRESSES CALCULATED IN THE LARGE TUBE (61)

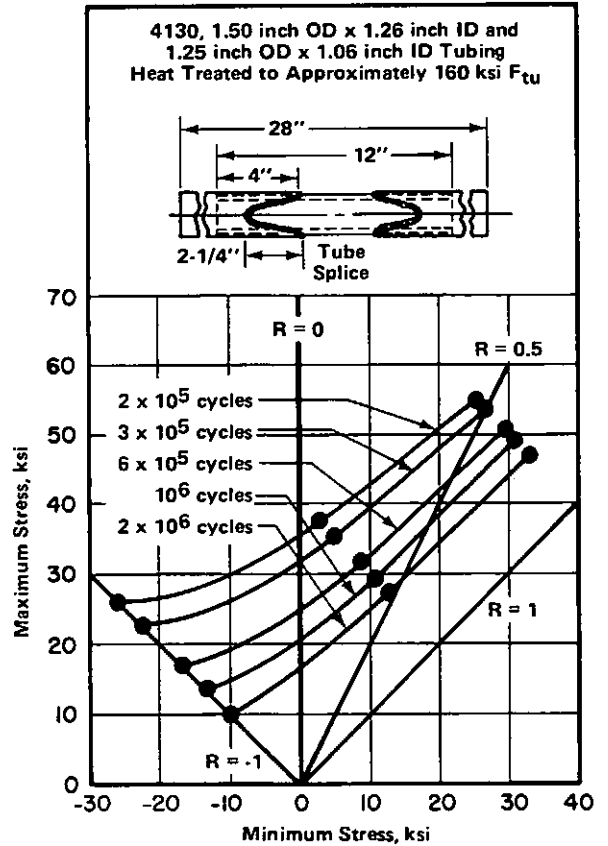


FIGURE 4.0310. CONSTANT-LIFE DIAGRAM FOR AXIAL-LOADED TUBE SPLICES - STRESSES CALCULATED IN SMALL TUBE (61)